

Figure 5. – Probability to failure as a function of Weibull stress for the case of intergranular fracture. Tests on smooth and on notched (AE) specimens.

C-Mn-Ni-Mo steels

Using this modified expression for the Weibull stress, it was found (fig. 5) that the Weibull theory accounts very well for the results obtained on different specimen geometries, including thin tubes under tension-torsion loading. The value of $m \sim 13$ for the Weibull shape factor is much smaller than the value (typically ~ 22) found for cleavage fracture, reflecting the larger scatter found for the case of intergranular fracture.

Short crack effects in fracture and fatigue

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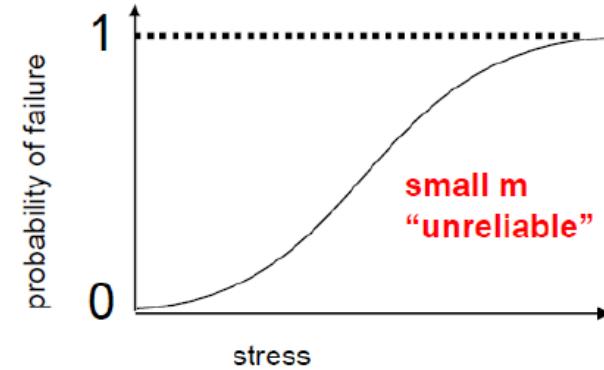
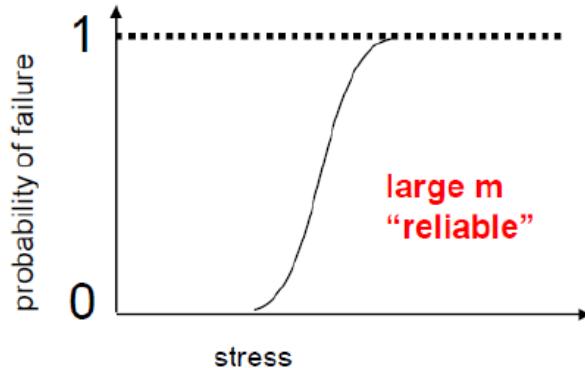
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m and reliability

- **large m:**
narrow distribution, small spread
➔ reliable material
- “tough” ceramic components:
 $m = 10-40$
- **small m:**
wide distribution, large spread
➔ unreliable material
- “bad” ceramic components:
 $m = 1-10$



3-point and 4-point bend tests

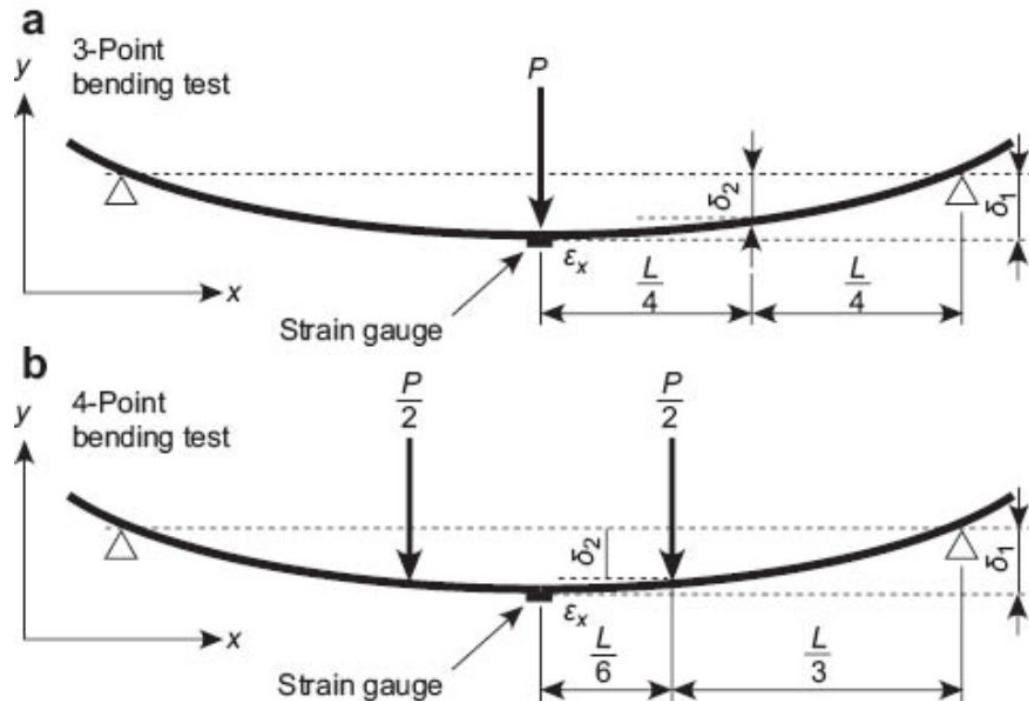
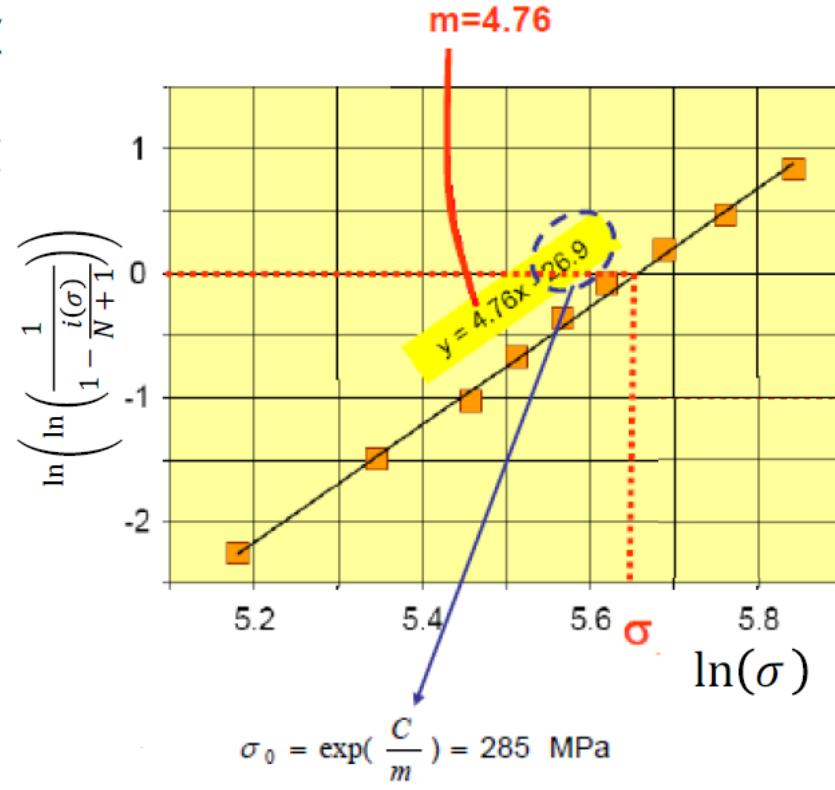


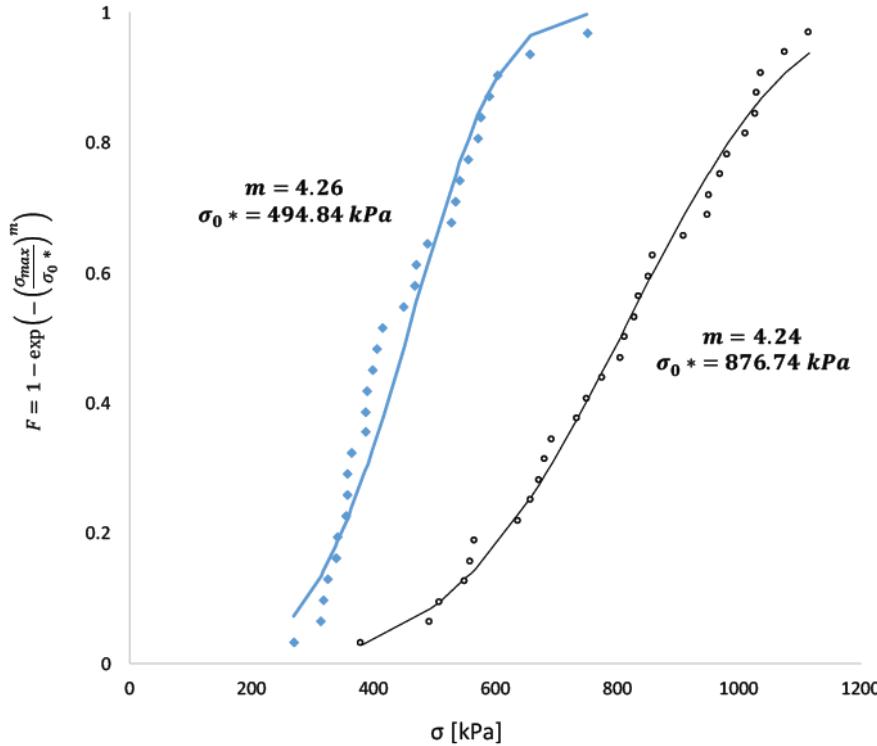
Figure 3-2. Three- and four-point bending; here the inner span L_i equals one-third of the outer span L_o , (a situation referred to as “loads at third points”).

Tensile data for ground glass with $N = 9$

Test No.	Failure strength (MPa)	Rank	Failure strength (MPa)	Failure probability $F(\sigma) = i/(N + 1)$
1	178	1	178	0.1
2	276	2	210	0.2
3	262	3	235	0.3
4	296	4	248	0.4
5	210	5	262	0.5
6	248	6	276	0.6
7	235	7	296	0.7
8	318	8	318	0.8
9	345	9	345	0.9



Bend tests of solid fat at two loading speeds (N = 30)



Fit of the two-parameter Weibull distribution (solid curves) to estimates of $F(\sigma)$ from 30 four-point bend specimens of palm kernel oil (a solid fat) tested at 0.01 mm/s (crosses) and 23 °C and at 10 mm/s and 23 ° (circles). The corresponding values of m and σ_0^* are given in each case, where $\sigma_0^* = \sigma_0/k_m^{1/m}$.