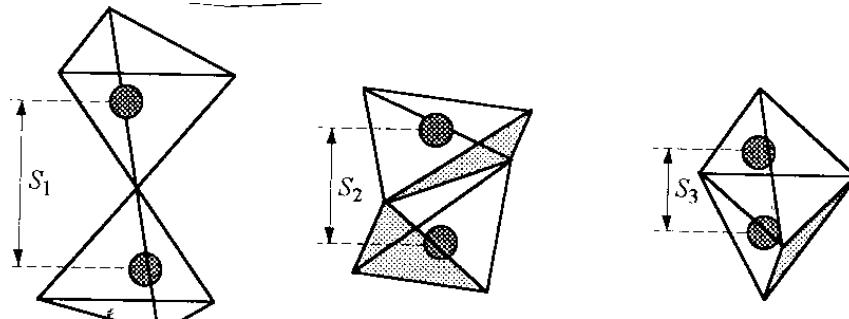


Lecture 2

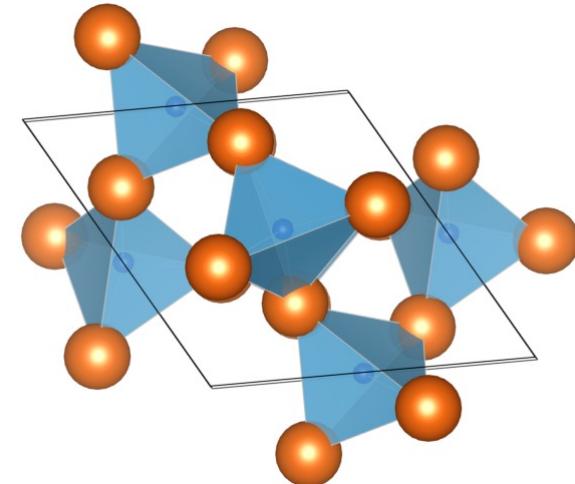
- Pauling rules, bonding and structures - continuation
- Simple crystalline structures – important examples
- Crystalline symmetry
- Macroscopic crystalline symmetry & point groups
- Point groups and real structures

Pauling rules -3, 4, 5

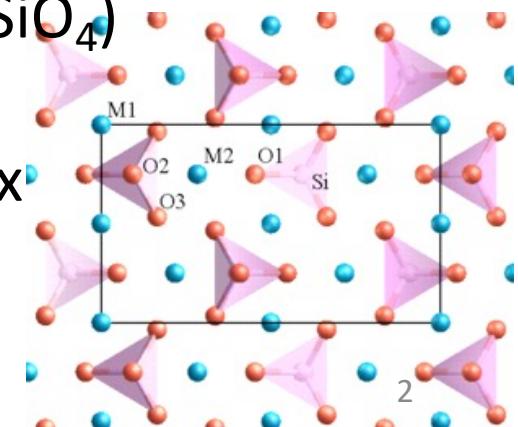
- **Rule 3:** The bond is strongest when coordination polyhedra share corners, less stable when they share edges, and least stable when they share faces.



Why?

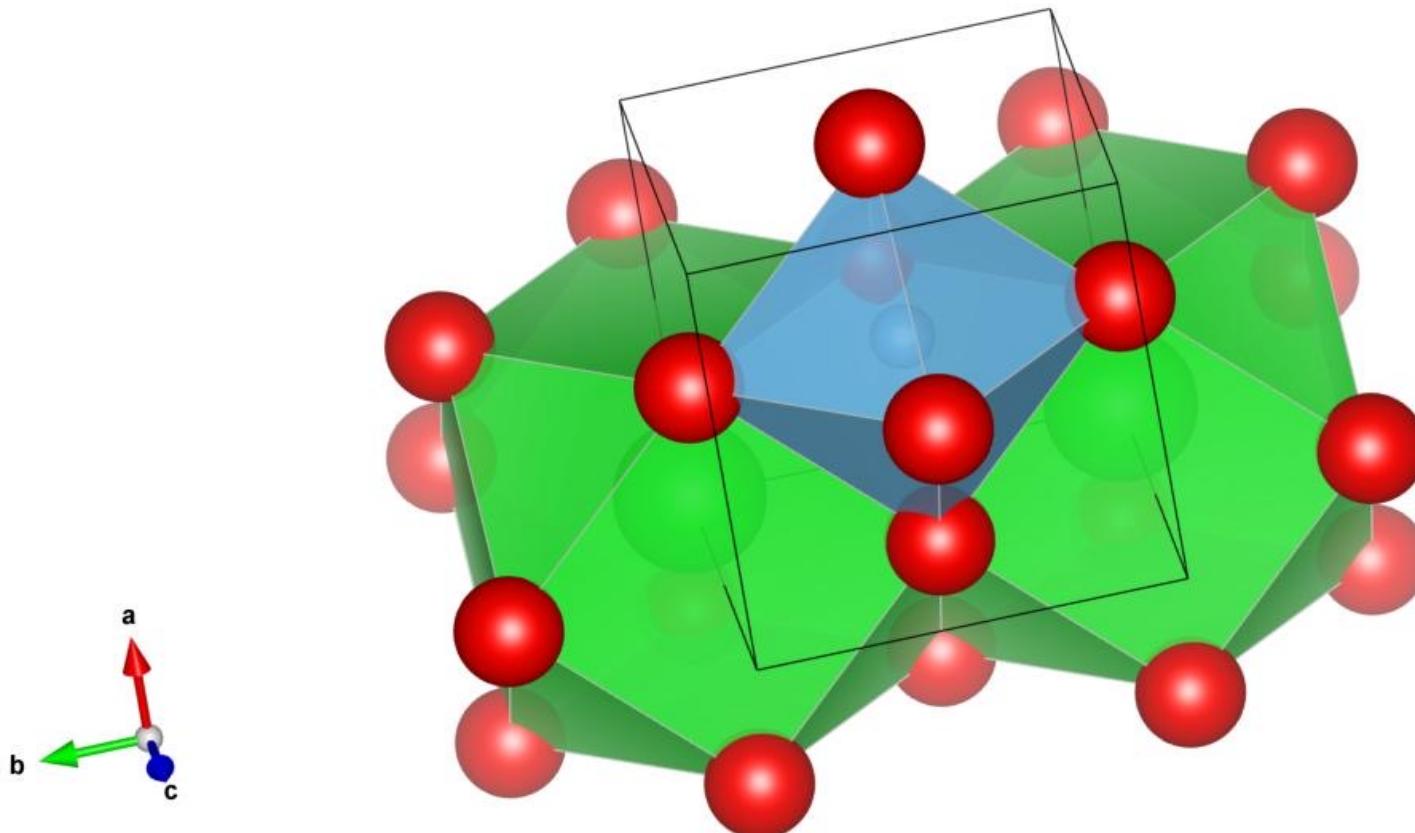


- **Rule 4:** Coordination polyhedra of small cations with large charge tend to share corners (or not share any elements: olivine $(\text{Mg},\text{Fe})_2\text{SiO}_4$)
- **Rule 5:** Simpler structures are more likely than complex (the rule of parsimony)



Polyhedra sharing – Rules 3 and 4

BaTiO_3 , perovskite structure



Ti^{+4} is small, heavily charged, and corresponding polyhedra share corners
- Compare this to Ba^{+2}

Pauling rules - MgO

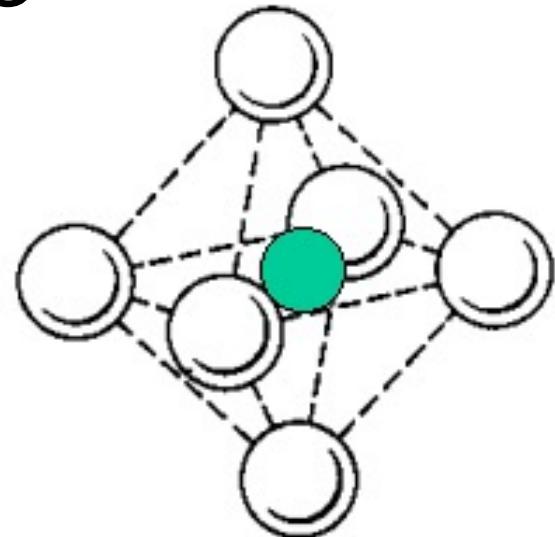
For MgO:

$$r_{Mg^{2+}} = 0.86 \text{ \AA}, r_O = 1.26 \text{ \AA}$$

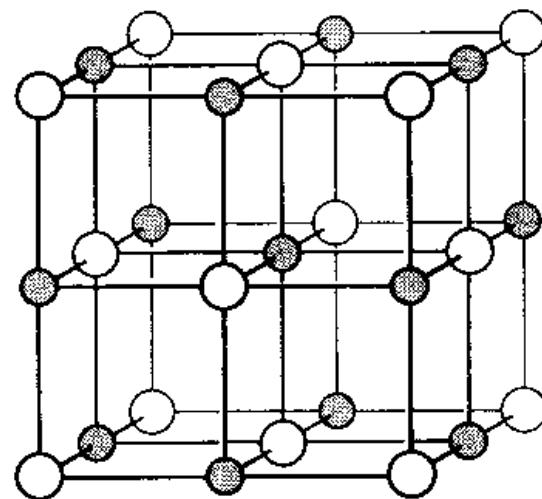
$$r_{Mg^{2+}}/r_O = 0.86/1.26 = 0.68 \rightarrow CN(Mg) = 6 \text{ for } Mg^{2+}$$

$$\text{Bonding force around } Mg^{2+}: +2 / 6 = 1/3$$

$$\text{Oxygen CN: } CN(O) \times 1/3 + (-2) = 0 \rightarrow CN(O) = 6$$



Are other Pauling rules obeyed?



- What is the structure based on rules 1 and 2?

Pauling rules – ReO_3

For ReO_3 :

$$r_{\text{Re}+6} = 0.76\text{\AA}, r_{\text{O}} = 1.26\text{\AA}$$

$$r_{\text{Re}+6}/r_{\text{O}} = 0.77/1.26 = 0.61 \rightarrow \text{CN}(\text{Re}) = 6 \text{ for Re}^{+6} \quad \text{Re}^{\text{VI}}\text{O}^{\text{II}}_3$$

Bonding force around Re^{+6} : $+6 / 6 = 1$

Oxygen $\text{CN}(\text{O})$: $\text{CN}(\text{O}) \times 1 + (-2) = 0 \rightarrow \text{CN}(\text{O}) = 2$

Which type of coordination for O?

Each O is connected to 2 Re i.e.

each O is the bridge between

two octahedra.

Are other Pauling rules satisfied?

- Corners of octahedra are shared

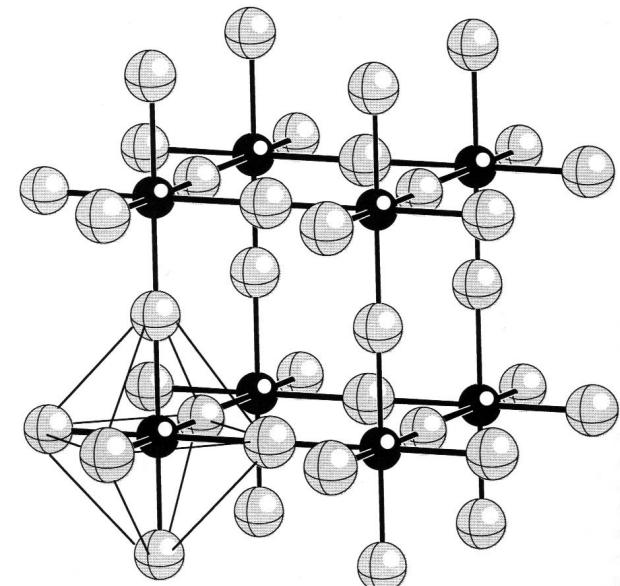
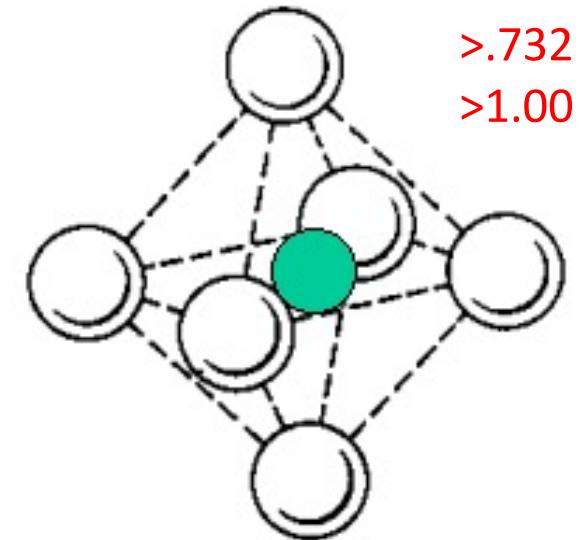
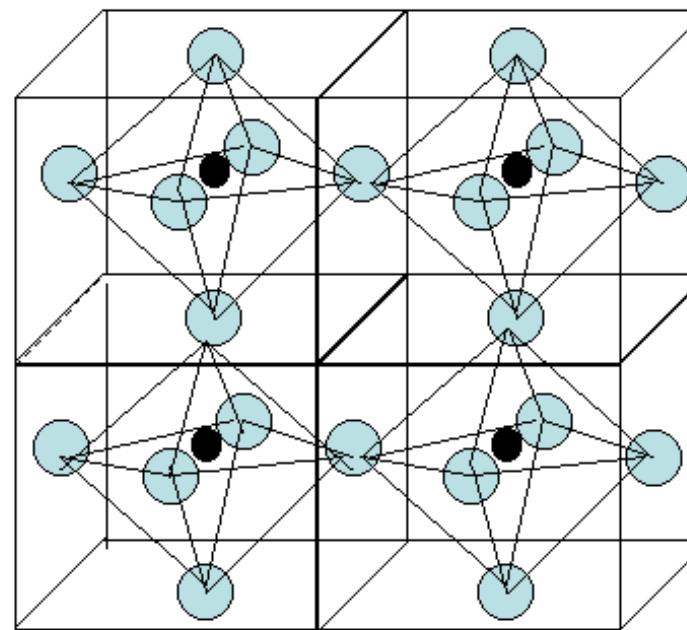


Figure 10.9. The structure of ReO_3 . The Re are the smaller black spheres. The fact that Re is octahedrally coordinated and O is in two-fold coordination was deduced from Pauling's first two rules.

R/ NC	
>.225	4
>.414	6
>.732	8
>1.00	12

Pauling rules for structures with two cations

-example of BaTiO_3

- Apply Pauling rules to determine coordination numbers (CN) for all ions in BaTiO_3 . Specify coordination numbers of the anions to each cation.
- We start by finding radii of all ions.

Recommendation: use the table composed after Shannon and Prewitt from the previous lecture,

Pauling rules for structures with two cations

-example of BaTiO₃-

We recall conditions for coordination of cation with respect to near-neighbor anions, which is given by:

$r_{\text{cation}}/r_{\text{anion}}$	CN(cation)
>.225	4
>.414	6
>.732	8
>1.00	12



$$\text{CN}(\text{Ba}) = 12$$

$$\text{CN}(\text{Ti}) = 6$$

Pauling rules for structures with two cations -example of BaTiO_3 -

We now need to find coordination number of oxygen, $\text{CN}(\text{O})$.

For this we use Pauling second rule. When there are two cations around oxygen, then the rule refers to both cations considered together:

$\text{CN}(\text{O})_{\text{Ba}} * (\text{Bond strength around Ba}) + \text{CN}(\text{O})_{\text{Ti}} * (\text{Bond strength around Ti})$
=absolute value of charge (valence) of oxygen.

We first find bond strengths for the cations:

For Ba: Bond strength around Ba=(valence of Ba)/ $\text{CN}(\text{Ba})=2/12=1/6$

For Ti: Bond strength around Ti=(valence of Ti)/ $\text{CN}(\text{Ti})=4/6=2/3$

Pauling rules for structures with two cations -example of BaTiO₃-

We get:

$\text{CN(O)}_{\text{Ba}} * (\text{Bond strength around Ba}) + \text{CN(O)}_{\text{Ti}} * (\text{Bond strength around Ti}) =$
absolute value of charge of oxygen --->

$$\text{CN(O)}_{\text{Ba}} * (1/6) + \text{CN(O)}_{\text{Ti}} * (2/3) = 2$$

This gives: $\text{CN(O)}_{\text{Ba}} = 4$, $\text{CN(O)}_{\text{Ti}} = 2$

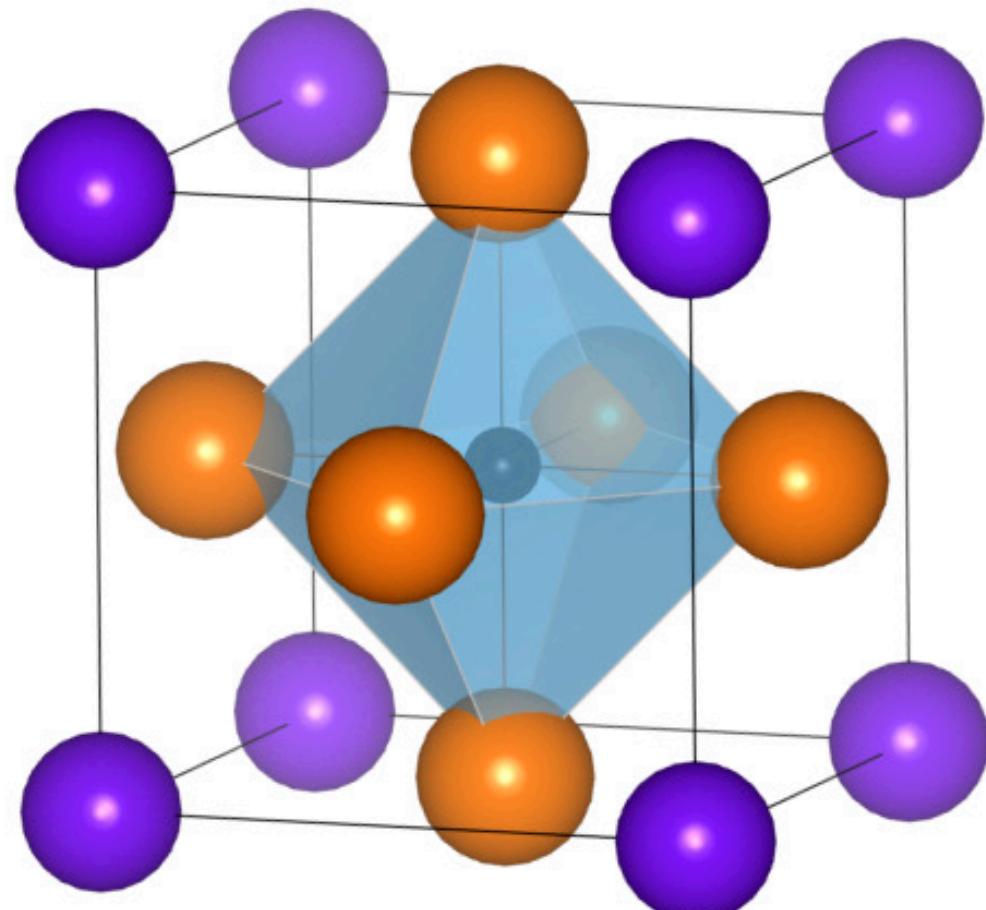
Therefore, each O anion is linked to 2 Ti ions, and 4 Ba ions. We write:



BaTiO_3 perovskite structure

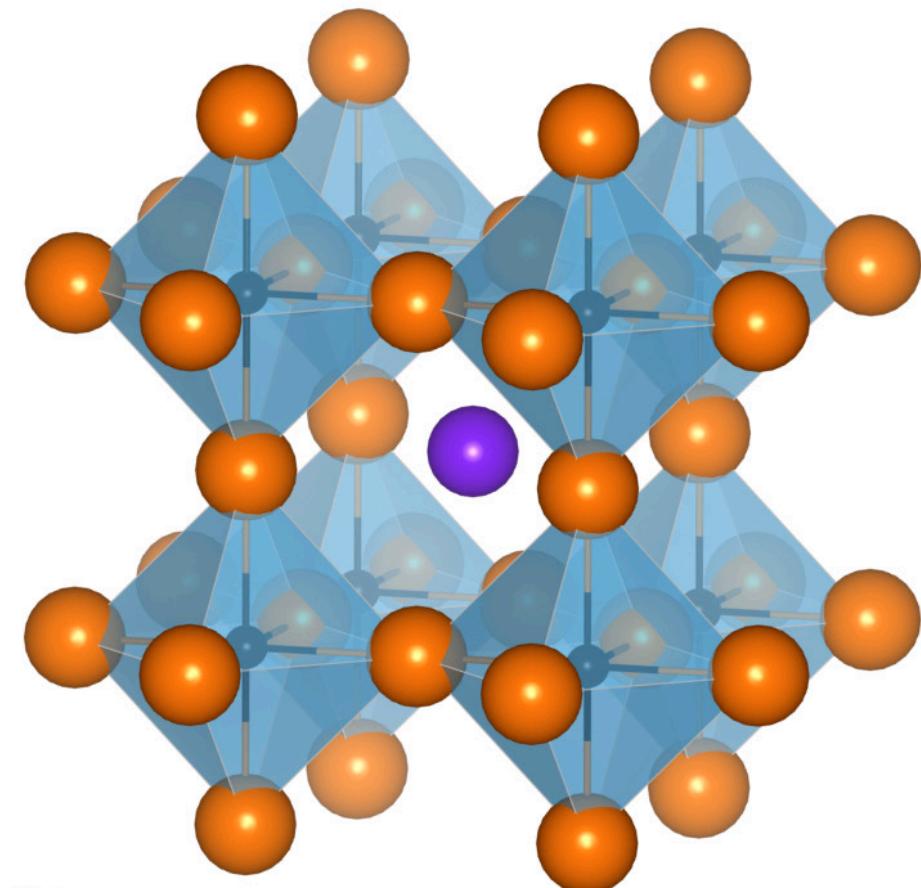
Check coordination numbers for oxygen:

$$\text{CN(O)}_{\text{Ba}} = 4, \text{CN(O)}_{\text{Ti}} = 2$$



BaTiO_3

Black:Ti; orange: O; Purple: Ba



Pauling rules -coordination number check-

- Taking $A_a^{CN(A)} B_b^{CN(B)} X_x^{CN(X)}$

where X is an anion, and A and B are cations, then:

$$aCN(A) + bCN(B) = xCN(X)$$

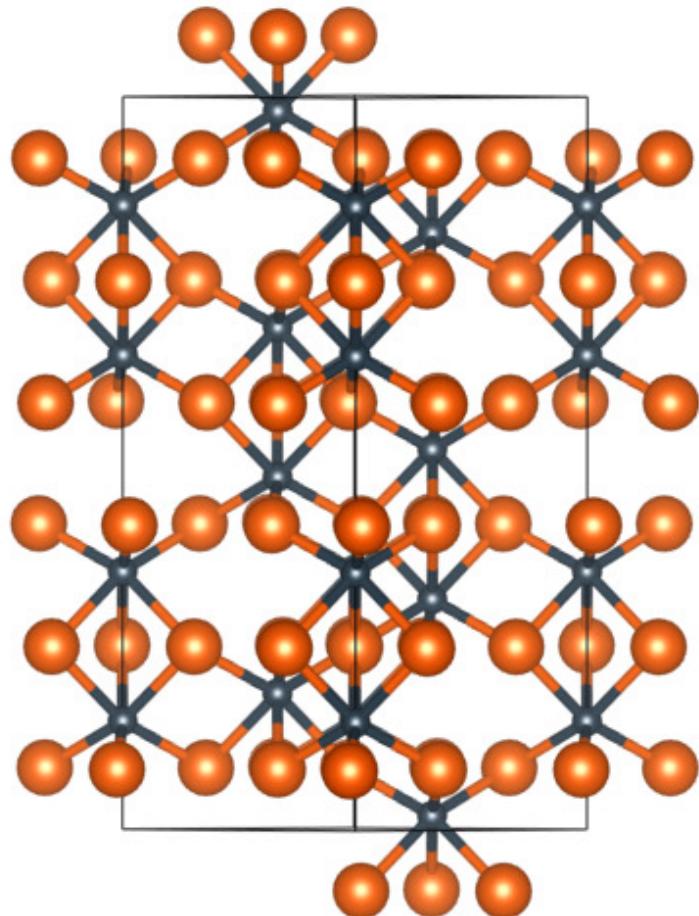
For BaTiO₃:

Check that the ratio between ions of different kinds is respected



Pauling rules – limitations - comment on sharing of polyhedra

example: Al_2O_3 (corundum, alumina)

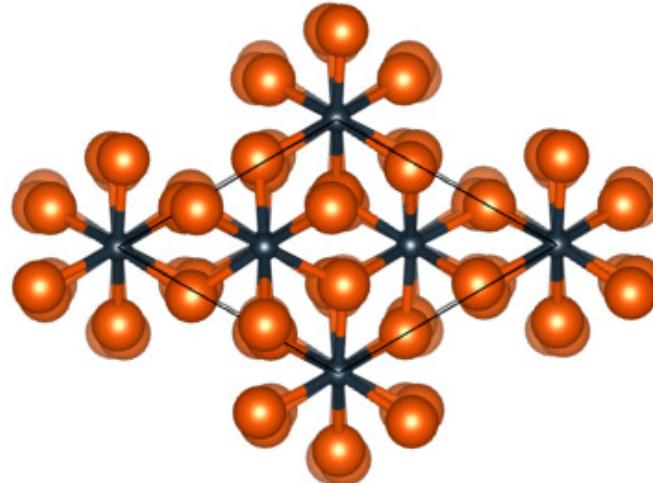


Unit cell view

Orange: O; Black: Al

Edge-shared octahedra

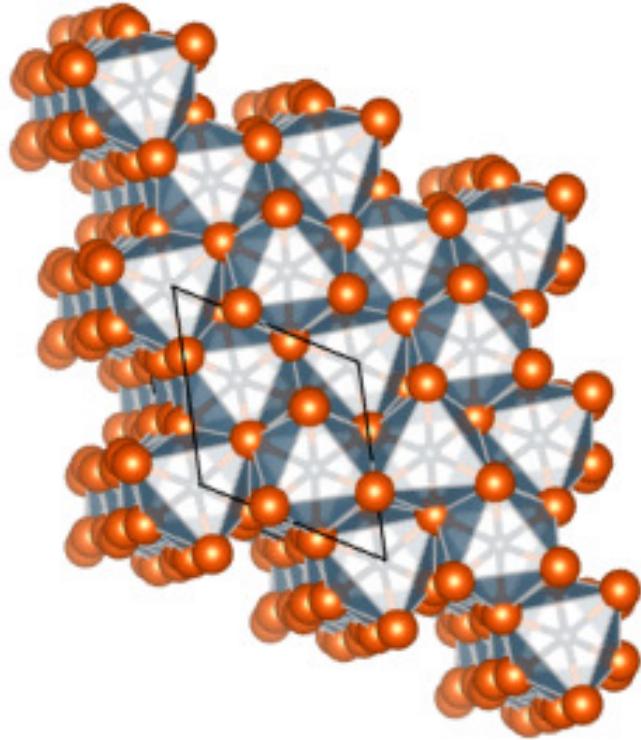
(we expect it shares corners rather than edges)



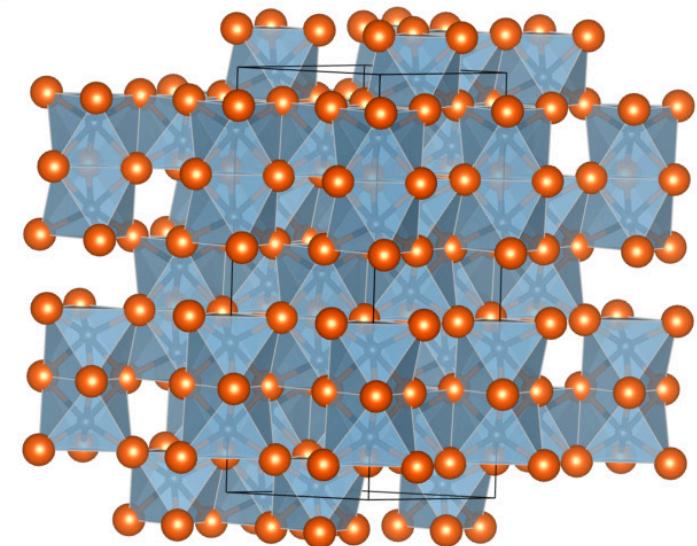
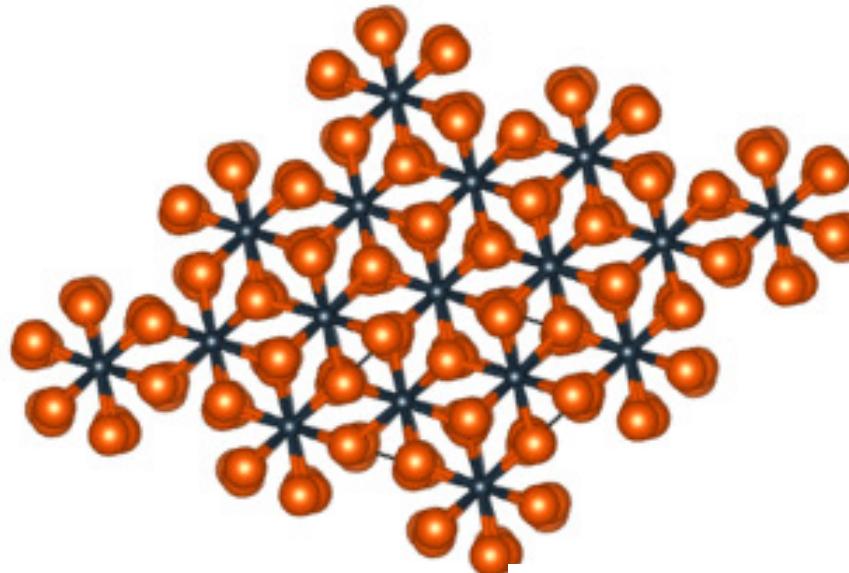
HCP, 2/3 of octahedral sites are filled, primitive trigonal, distorted tetrahedron

Pauling rules – comment on sharing of polyhedra

example: Al_2O_3 (corundum, alumina)

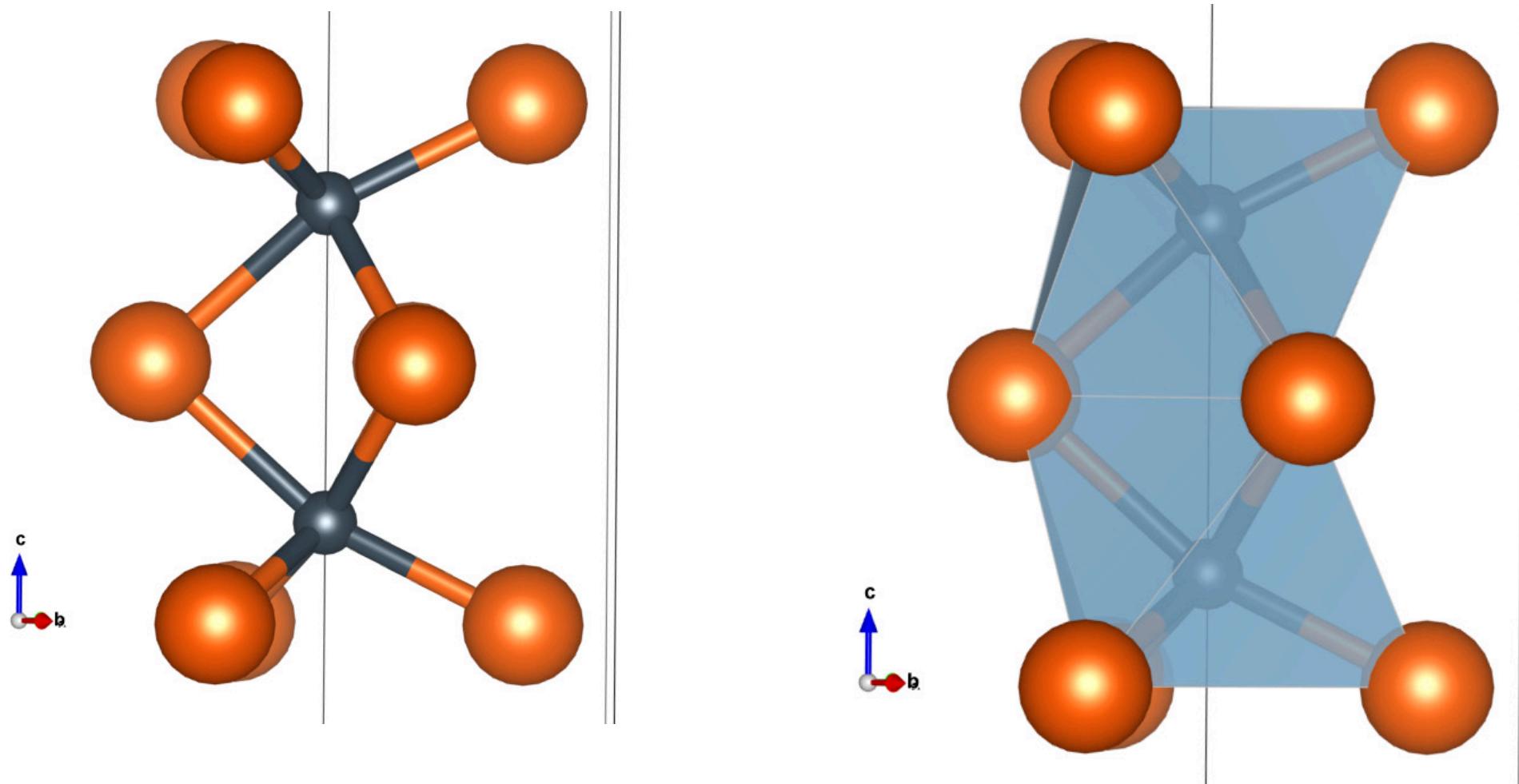


Side-sharing octahedra



Pauling rules – comment on sharing of polyhedra

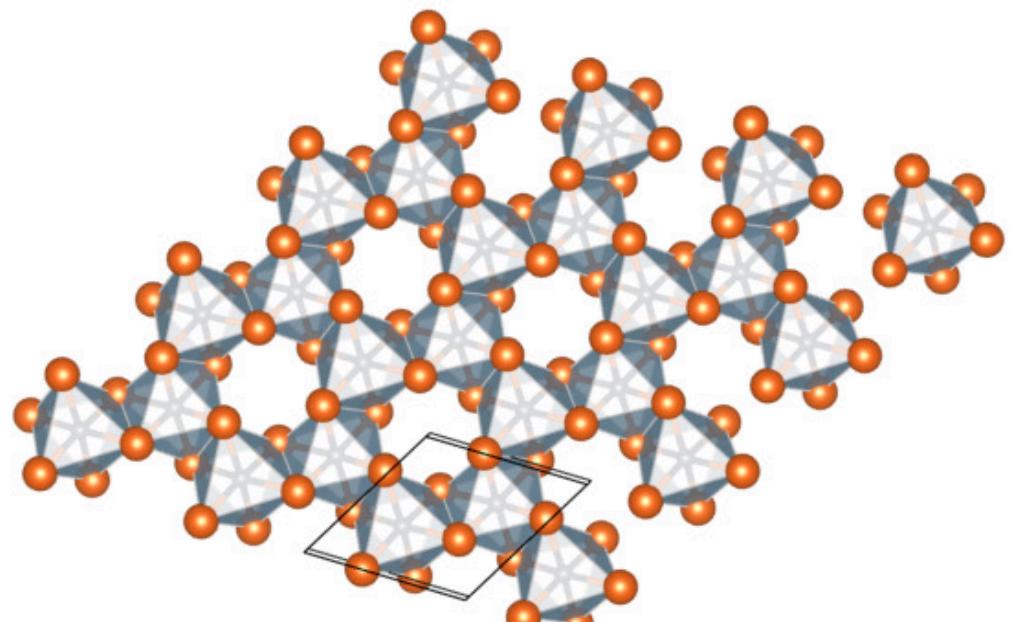
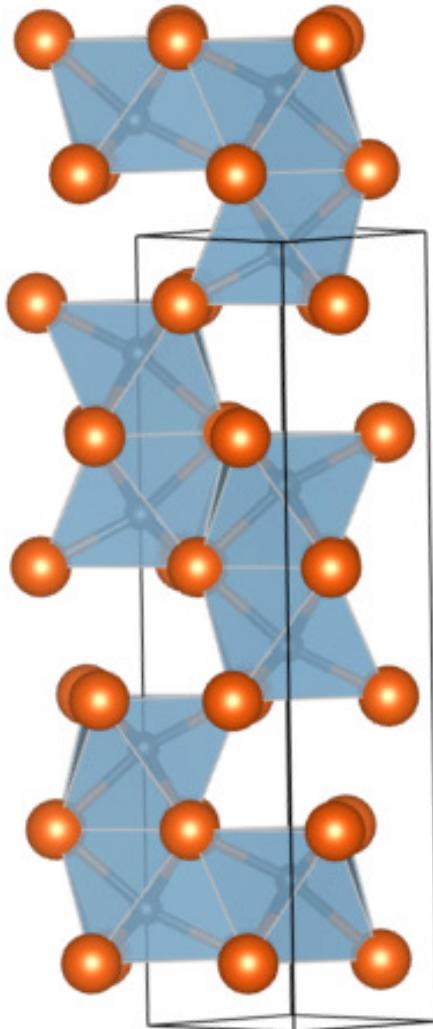
example: Al_2O_3 (corundum, alumina)



Octahedra are deformed to make distance between cations larger

Pauling rules – comment on sharing of polyhedra

example: Al_2O_3 (corundum, alumina)



there are empty spaces in the structure, which allow for the distortion of tetrahedra

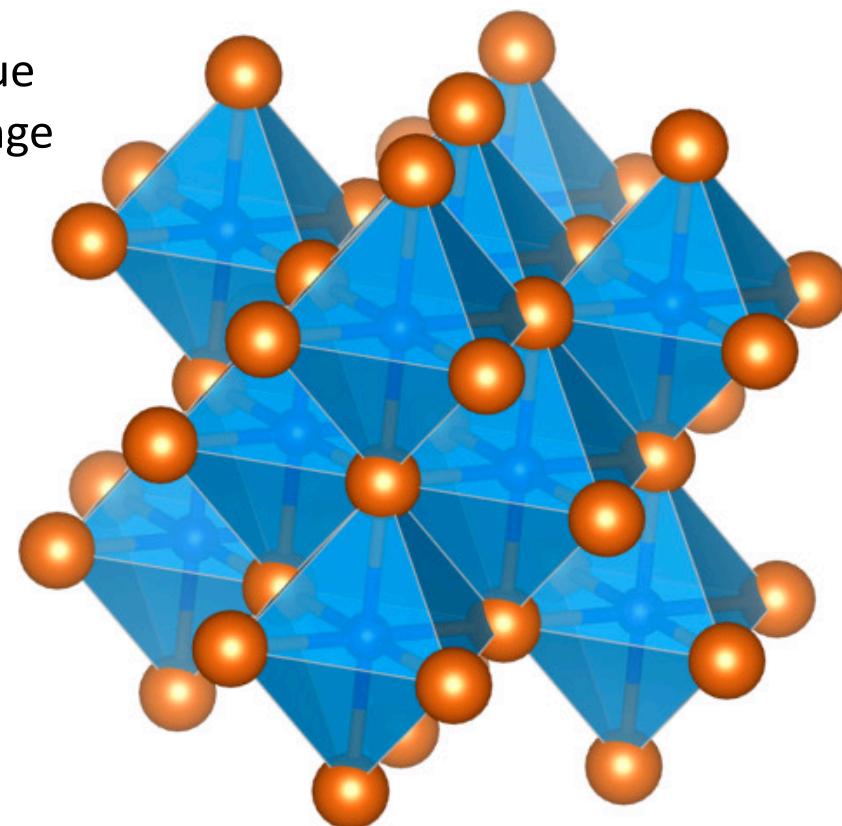
Important simple structures

- AX: MgO (rock-salt), CsCl (cesium chloride), ZnO (wurtzite), SiC (zinc blend) and its polymorphs
- AX₂: SiO₂ (quartz), ZrO₂ (fluorite), TiO₂ (rutile)...
- A₂X₃: Al₂O₃ (corundum)...
- ABX₃: BaTiO₃ (perovskite)...
- ...

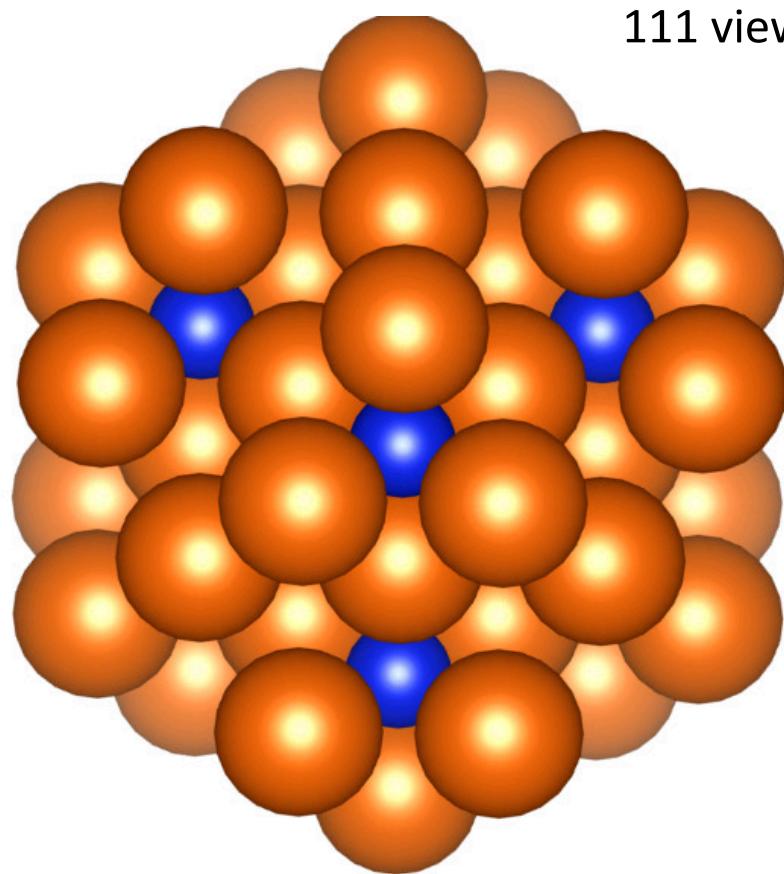
AX structures

Rock salt: LiF, KCl, MgO, NiO, CaO, TiC,... close cubic (FCC) packing-
all octahedral sites are filled, tetrahedral sites empty;
Cubic structure Fm-3m

Mg: blue
O: orange



111 view



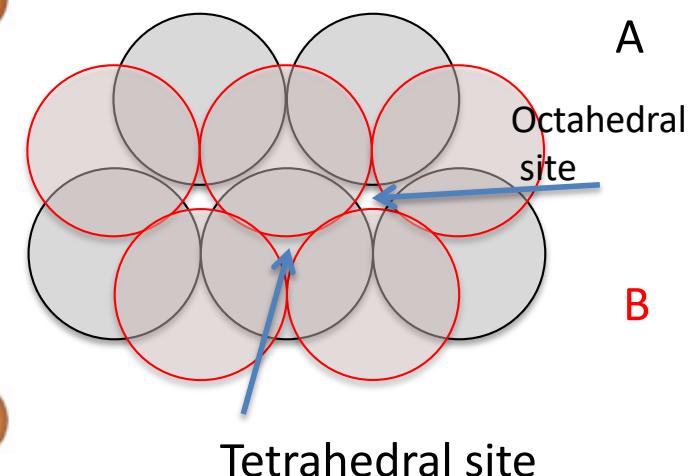
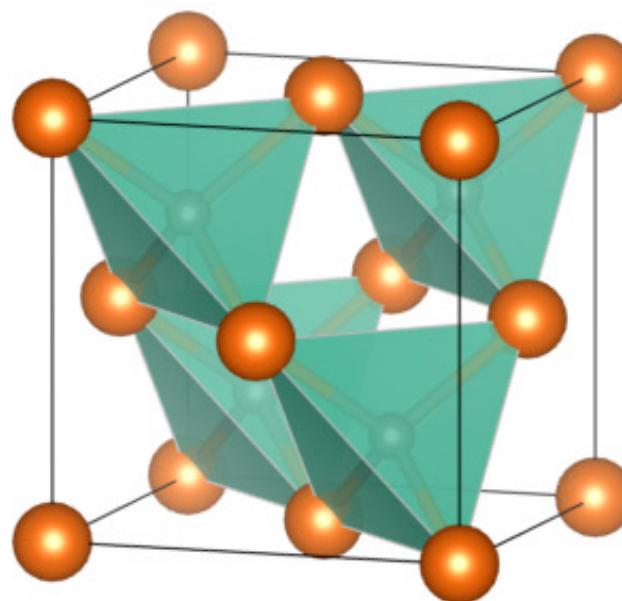
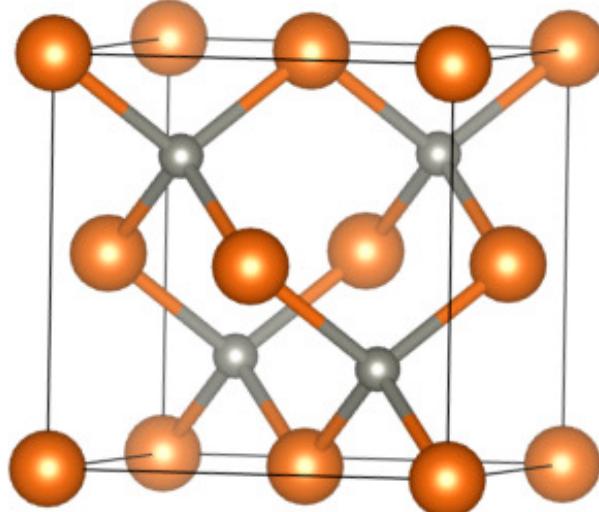
AX structures

Zincblende: ZnO, ZnS, BeO, SiC, BN, GaAs,... close cubic packing (FCC)-
Small cations in tetrahedral sites. Half of tetrahedral sites filled.

Cubic structure F-43m – it is different from the example on the previous slide

Orange: O

Gray: Zn

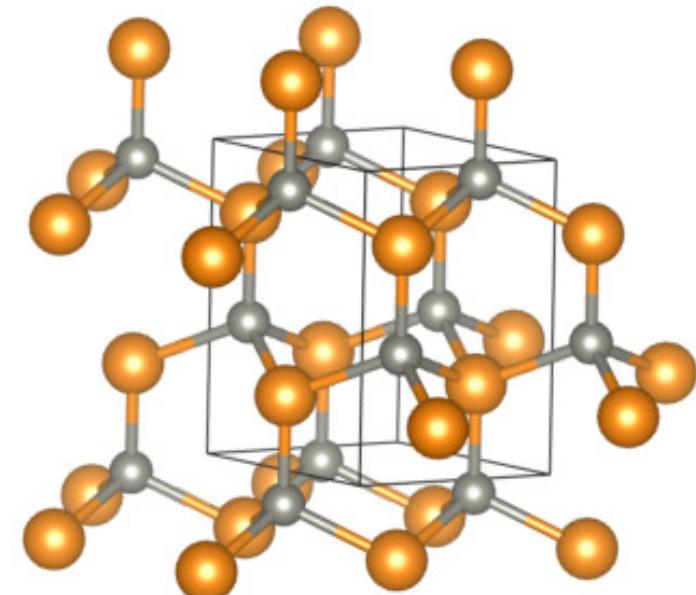
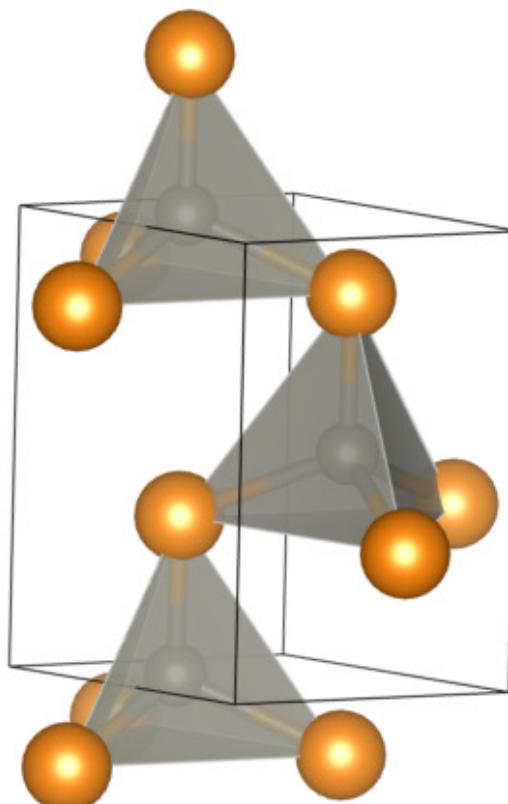
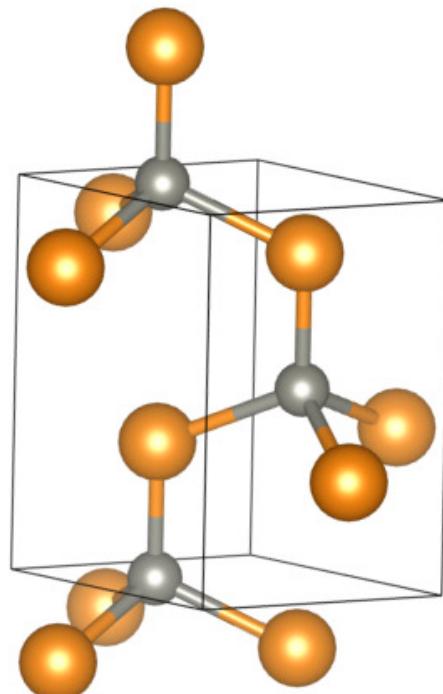


Cubic but not centrosymmetric! - Zincblende is piezoelectric: to be discussed later

AX structures

Wurtzite: ZnO, AlN, SiC,... hexagonal close packing (HCP)-
Small cations filling tetrahedral sites. Half of tetrahedral sites filled.

Hexagonal structure



Orange: O
Gray: Zn

This structure is polar: to be discussed later

Polytypes and polymorphs

- Zincblend and wurtzite are polytypes:
 - FCC or hexagonal, half of tetrahedral sites filled

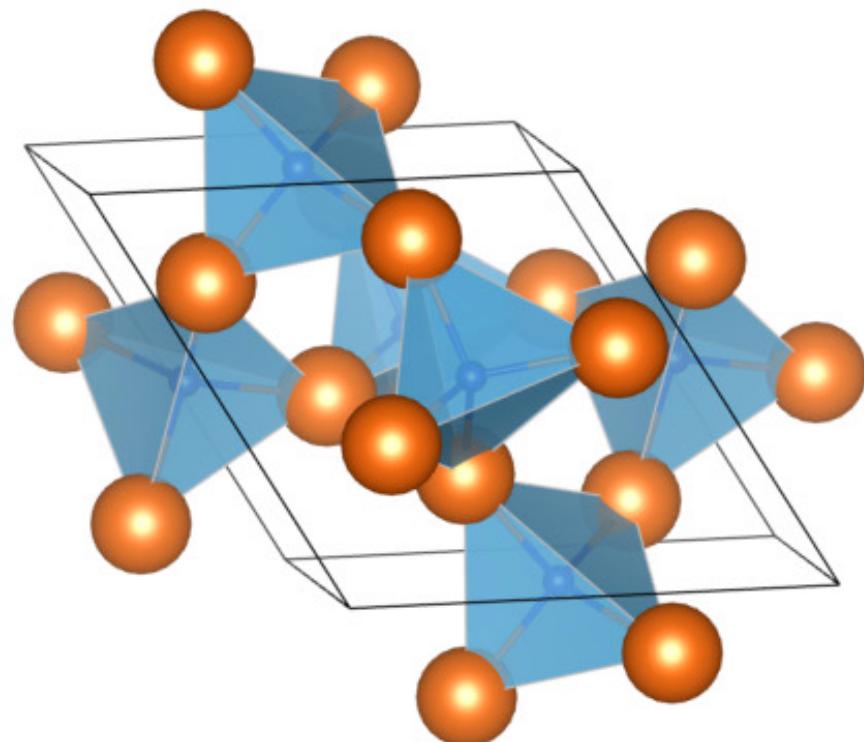
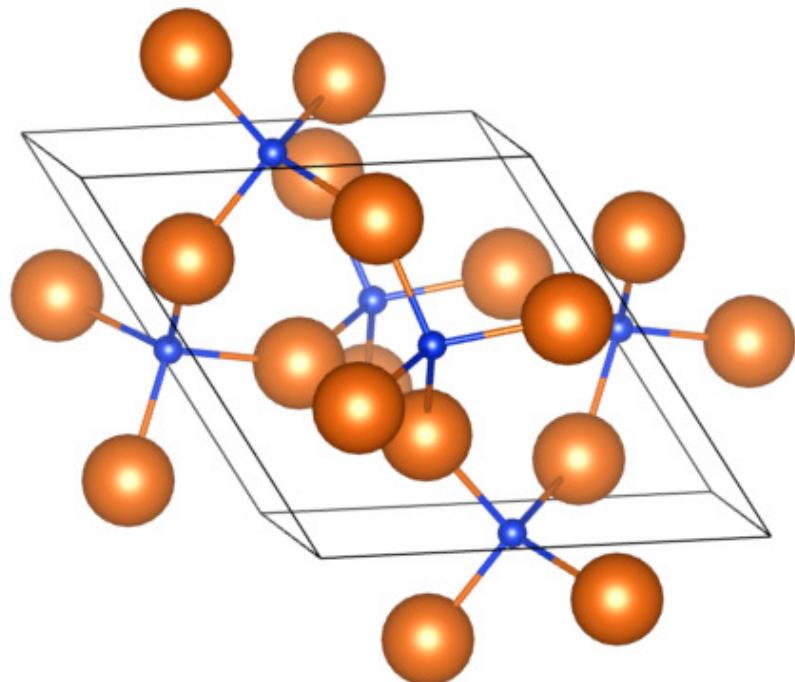
the same atoms arranged in a different packing

SiC: a large number of polytypes (dozens)

Polymorphs: compound appears in different crystalline forms. If transformation from one phase to another can take place by small displacements of atoms the transformation is called ***displacive***. If transformation requires bond braking it is called ***reconstructive***.

AX_2 structures

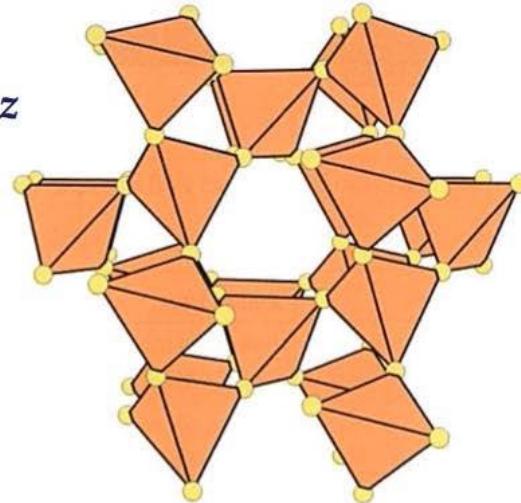
SiO_2 (silica) - Si filling half of tetrahedral sites.
Trigonal structure



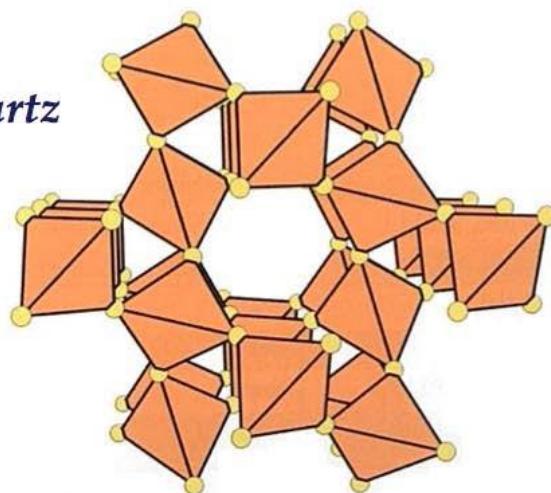
Alpha-quartz is noncentrosymmetric (piezoelectric)

Polymorphs of quartz

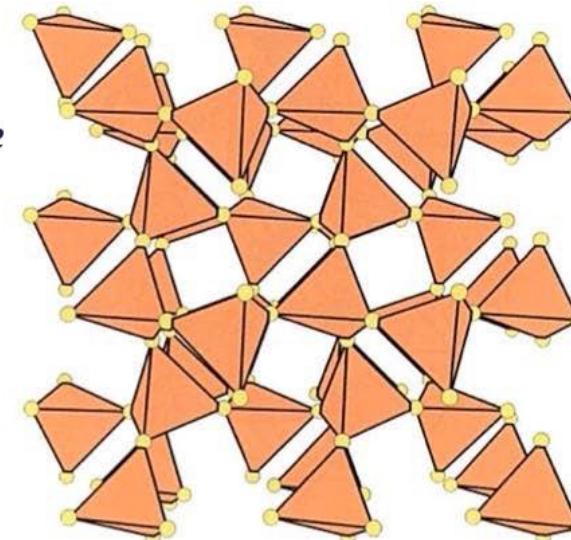
α -quartz



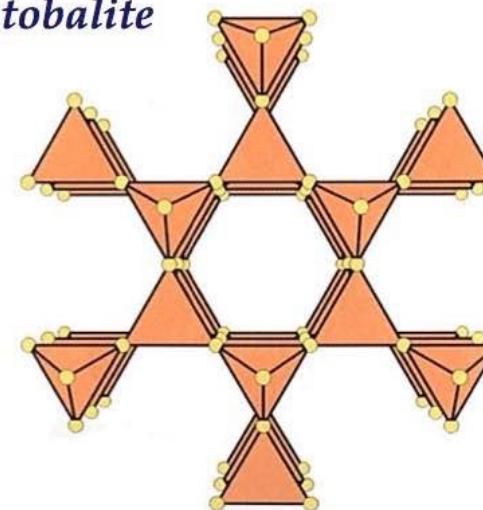
β -quartz



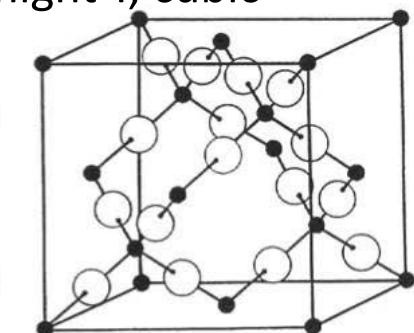
trydimite



cristobalite



hight T, cubic



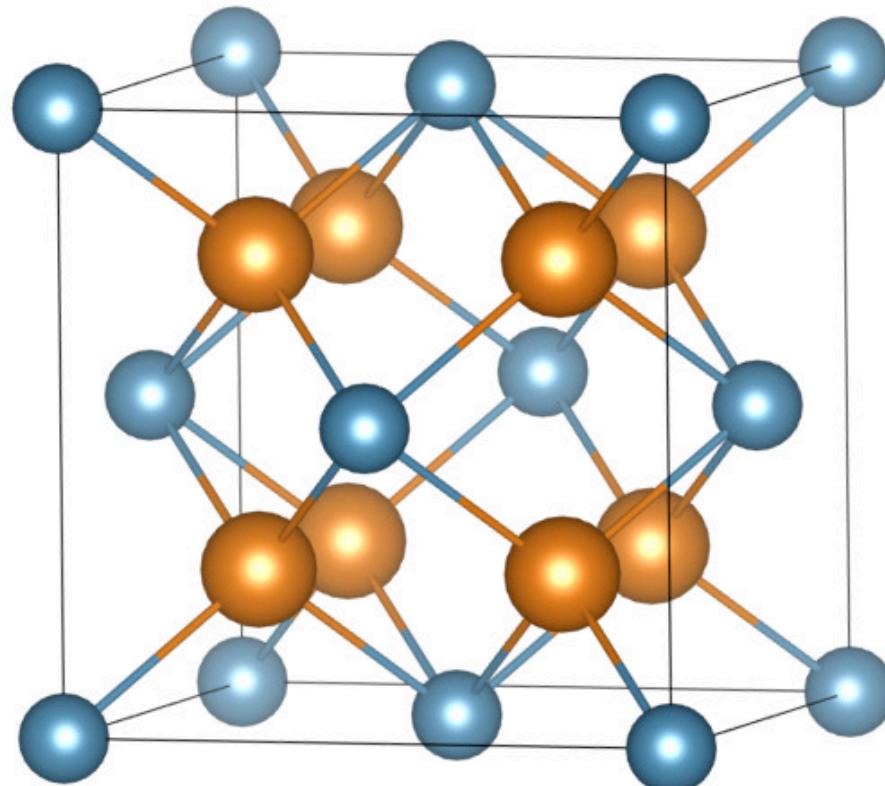
Glass-ceramic technology, W. Hoeland & G. Beall, The American Ceramic Society, 2002

AX_2 structures

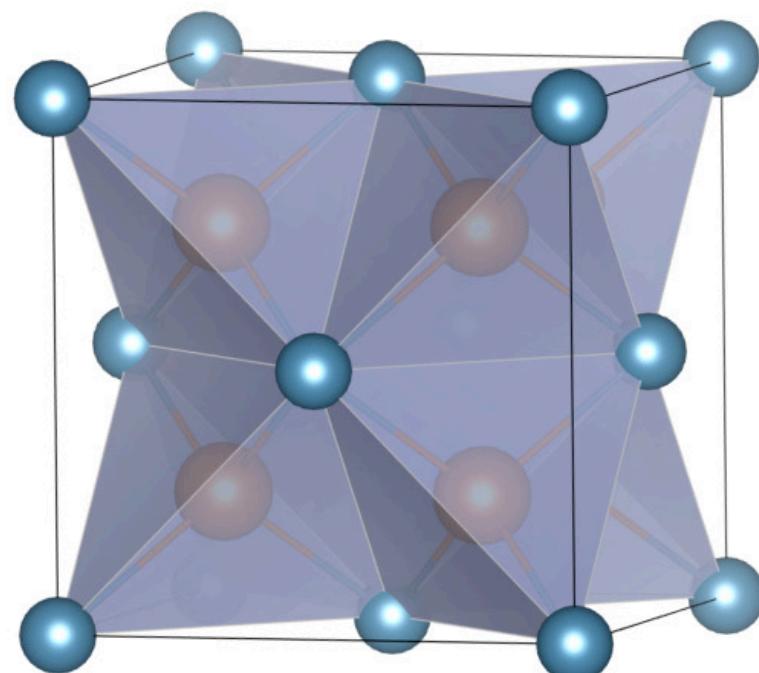
Fluorite: CaF_2 , ZrO_2 , CeO_2 ,...

8-fold coordination of cations. 4-fold coordination of anions

Cubic structure, some distortions may result in materials with new properties (e.g. Ferroelectricity/antiferroelectricity)



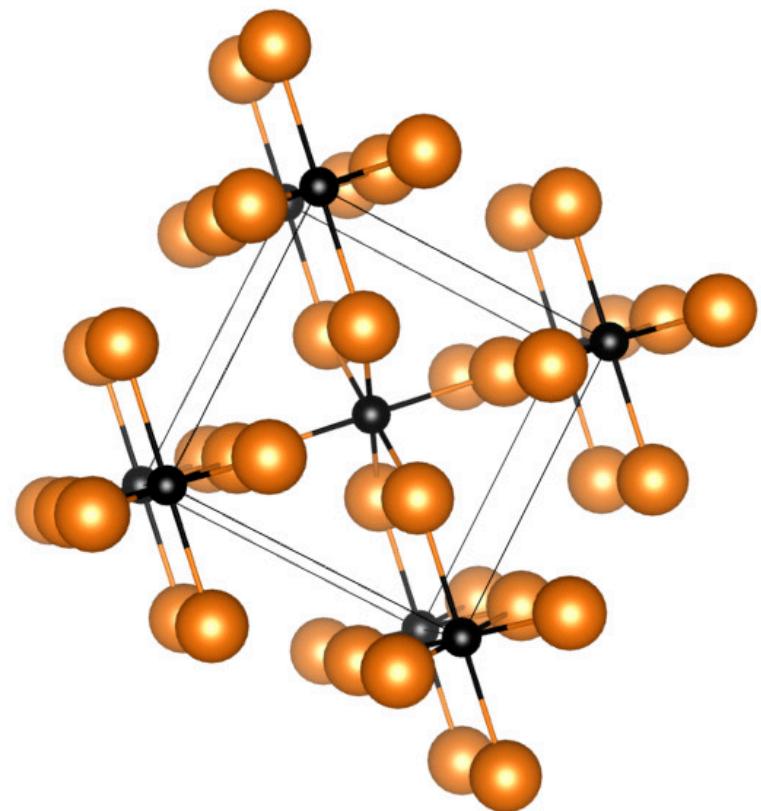
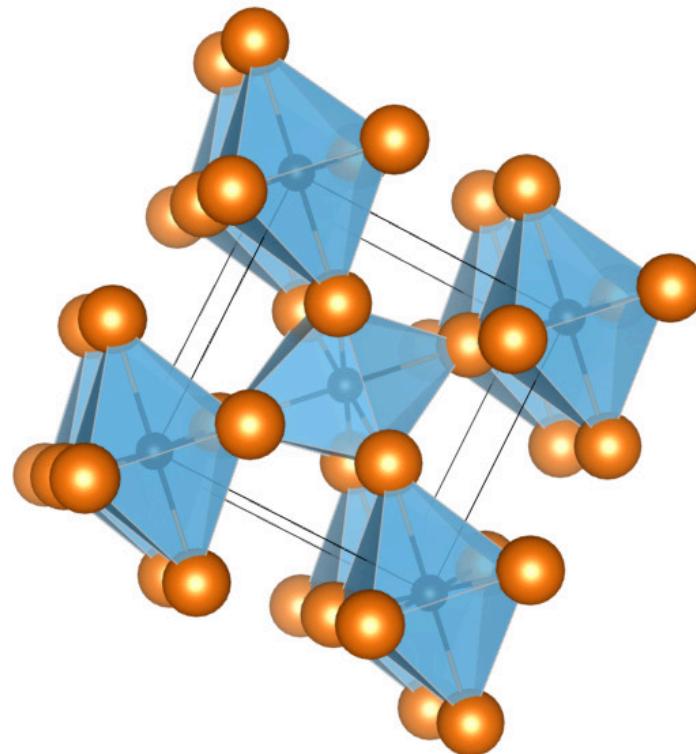
Orange: O,F Blue: Cation



AX_2 structures

Rutile: TiO_2 , SnO_2 , PbO_2 , VO_2 , NbO_2 , TeO_2 , MoO_2 , WO_2 , MnO_2 , RuO_2 , OsO_2 , IrO_2 , GeO_2

HCP packing; One half of octahedral sites filled by cations;
Tetragonal structure

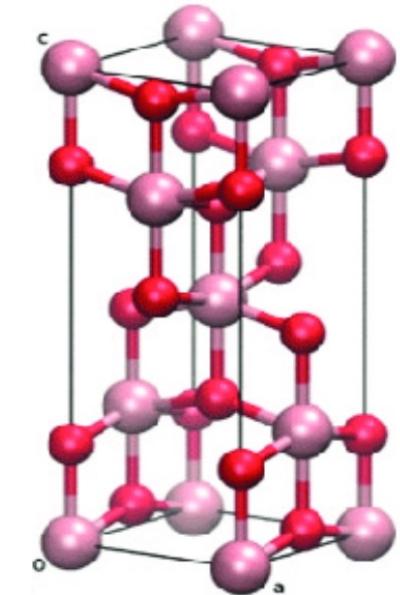
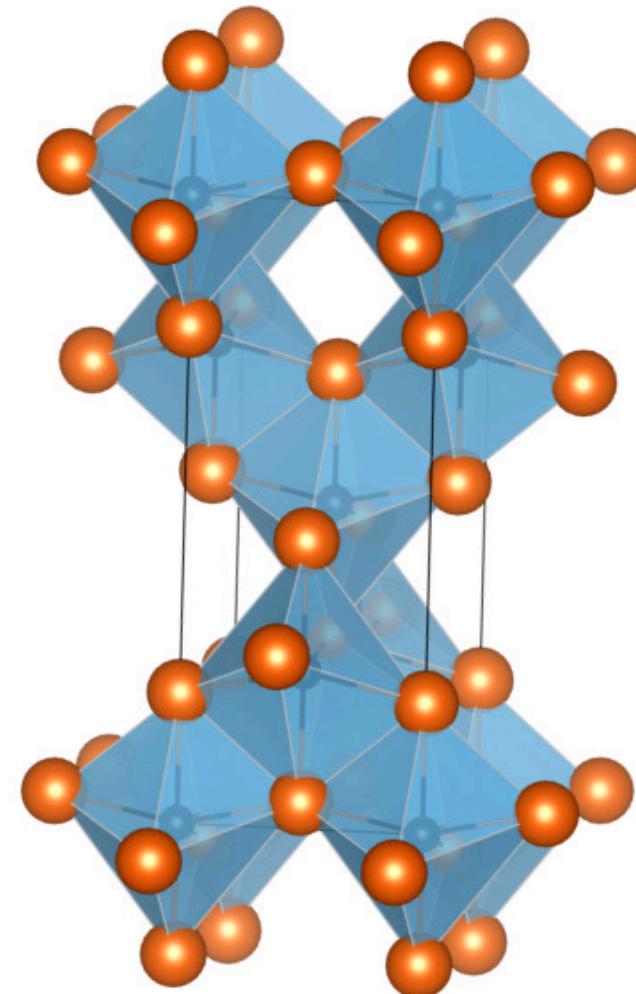
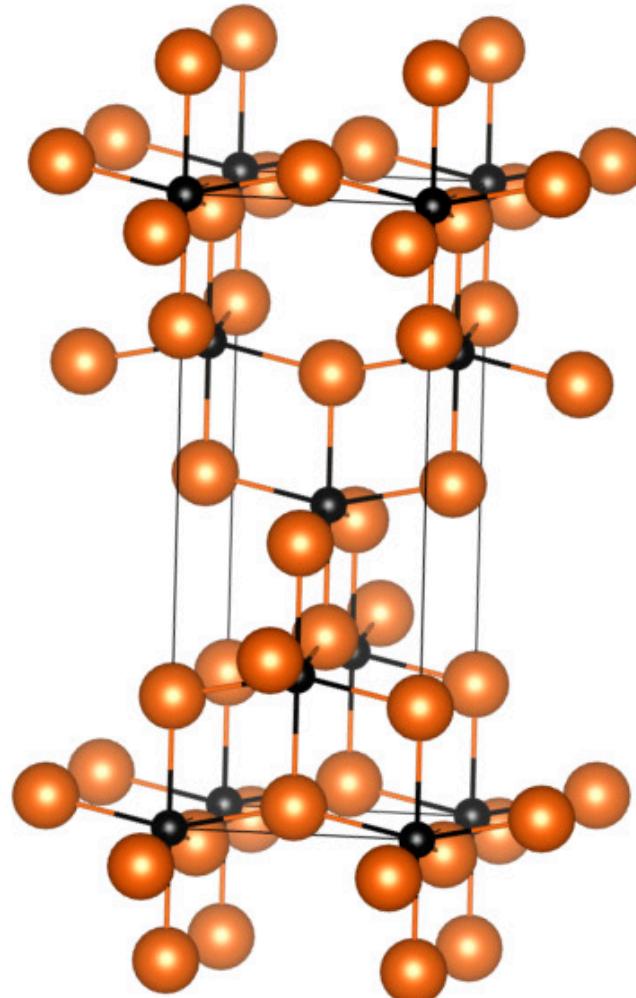


AX_2 structures

Anatase: TiO_2 polytype; A tetragonal structure;

CCP packing;

Here oxygen octahedra are sharing edges (Pauling rule is not respected – a metastable structure)

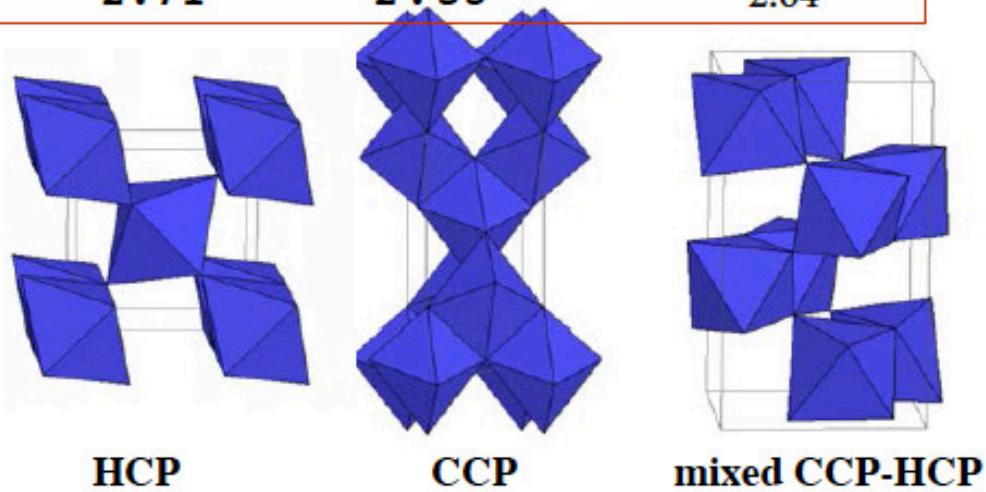


TiO₂

	Rutile	Anatase	Brookite
Form.Wt.	TiO₂	TiO₂	TiO₂
Z	79.890	79.890	79.890
CrystalSystem	Tet	Tet	Orth
PointGroup	4/mmm	4/mmm	mmm
SpaceGroup	P4₂/mnm	I4₁/amd	Pbca
UnitCell			
a(Å)	4.5845	3.7842	9.184
b(Å)			5.447
c(Å)	2.9533	9.5146	5.145
Vol	62.07	136.25	257.38
MolarVol	18.693	20.156	19.377
Density	4.2743	3.895	4.123
n	2.71	2.53	2.64

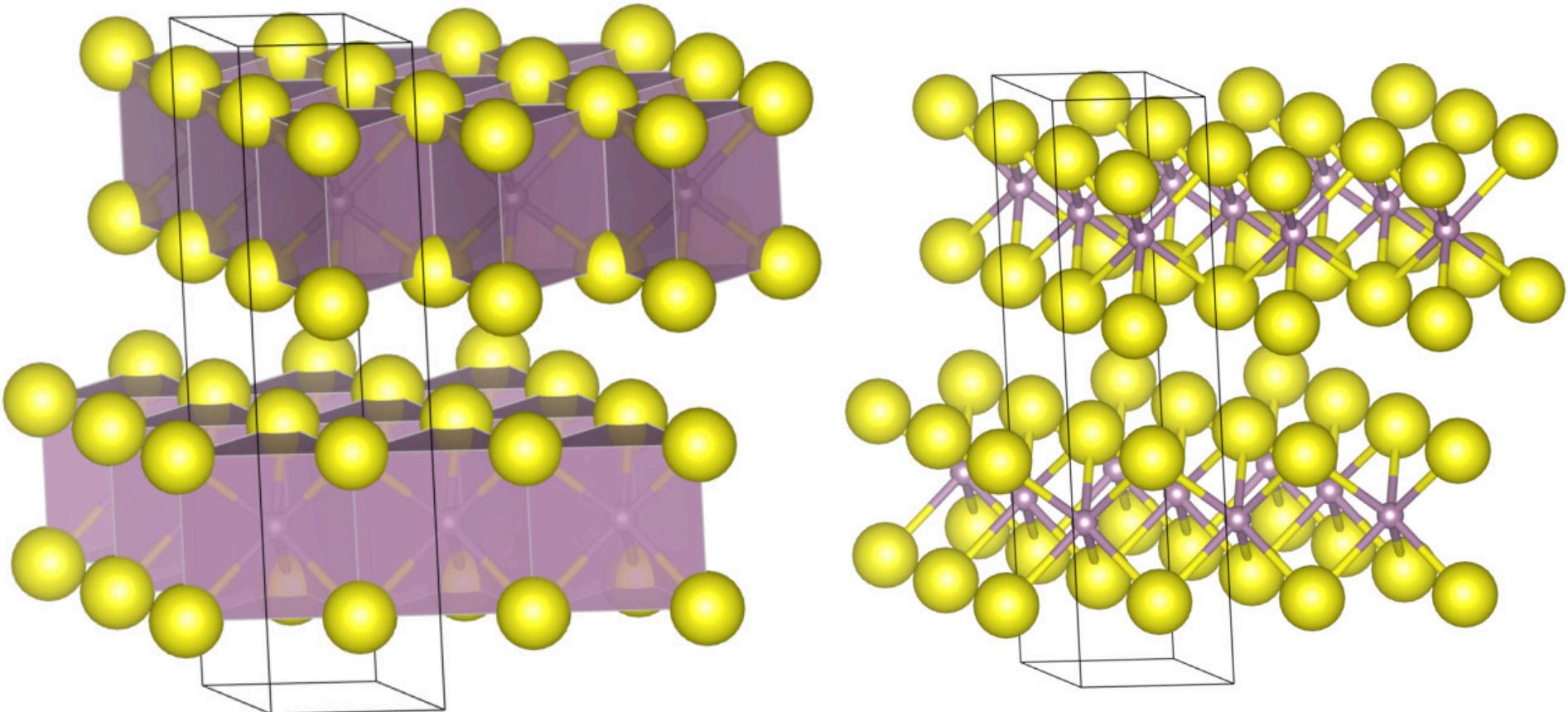
High density → high refractive index

Natural polymorphs of totania



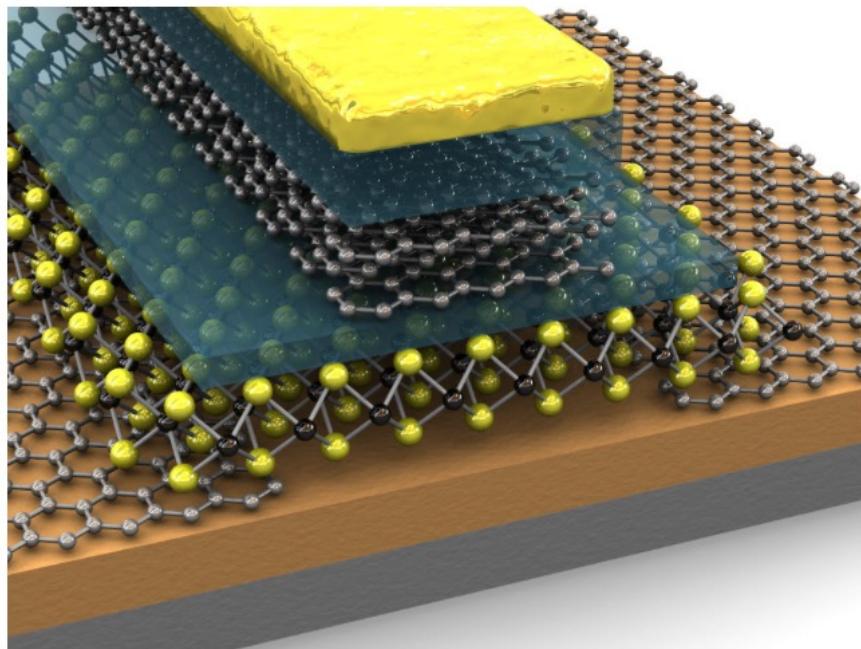
AX_2 structures

Layer structures: Molybdenum disulfide: MoS_2 polytype, WS_2 ,
Van der Waals bonds between planes. Hexagonal structure



Trigonal prismatic coordination of S around Mo

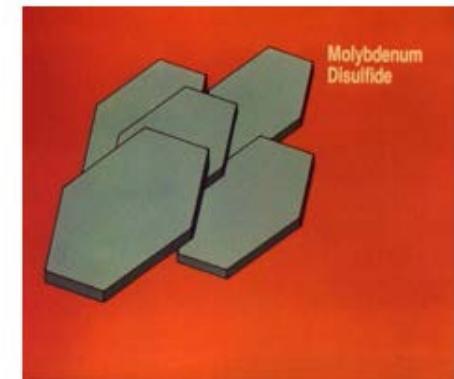
MoS₂ applications



Graphene and molybdenite combine into a flash memory prototype. Yellow-black molecules: molybdenite; gray hexagons: graphite (credit: EPFL)

MOLY PRO-SPEC® Motor Oil

- HIGH LEVEL OF MOLY
- INCREASES FUEL ECONOMY
- REDUCES WEAR, FRICTION AND HEAT
- REDUCES OIL CONSUMPTION
- HIGH TBN
- USES SUPER-FINE MOLY PARTICLES



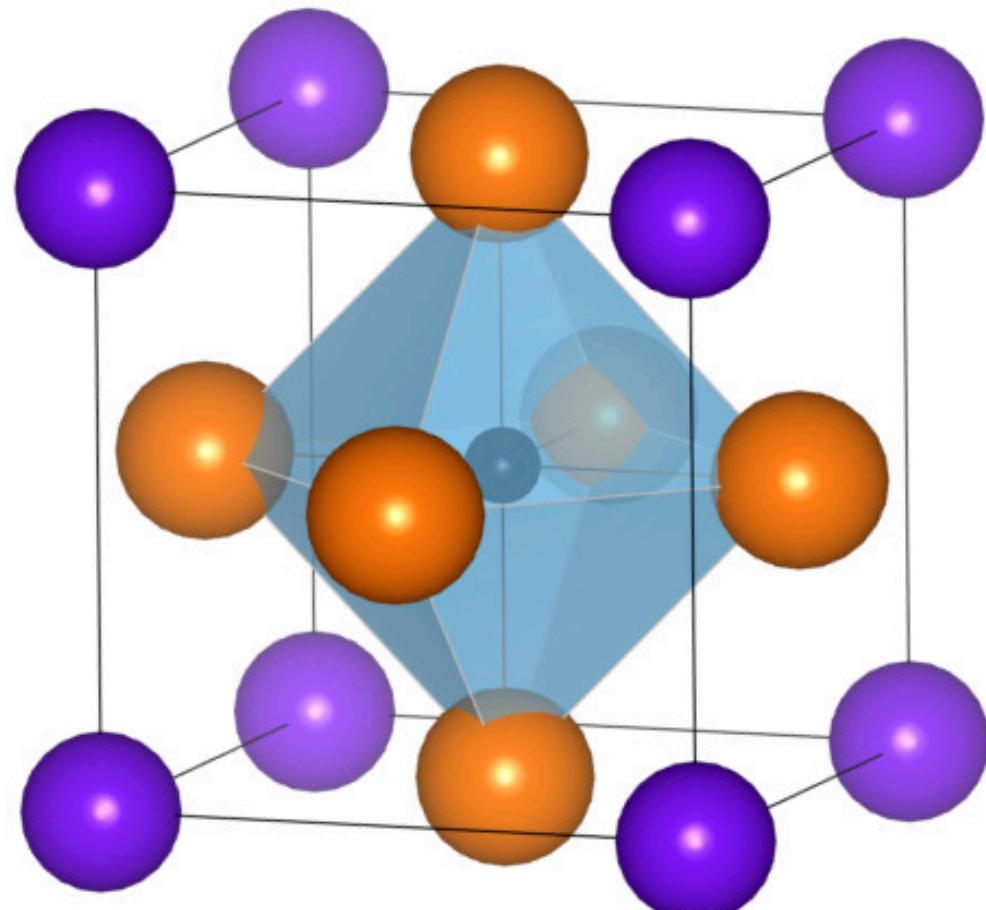
The MoS₂ platelets are capable of supporting up to 500,000 psi at the same time they offer little resistance to shear forces which tend to slide the platelets over one another thus reducing friction and wear.

- MoS₂ is a transition metal dichalcogenide (TMD) with a layered structure similar to graphene.
- Unlike graphene (which is gapless), monolayer MoS₂ has a direct bandgap (~1.8 eV), making it highly suitable for transistors, photodetectors, and optoelectronic applications.

ABX₃ structures

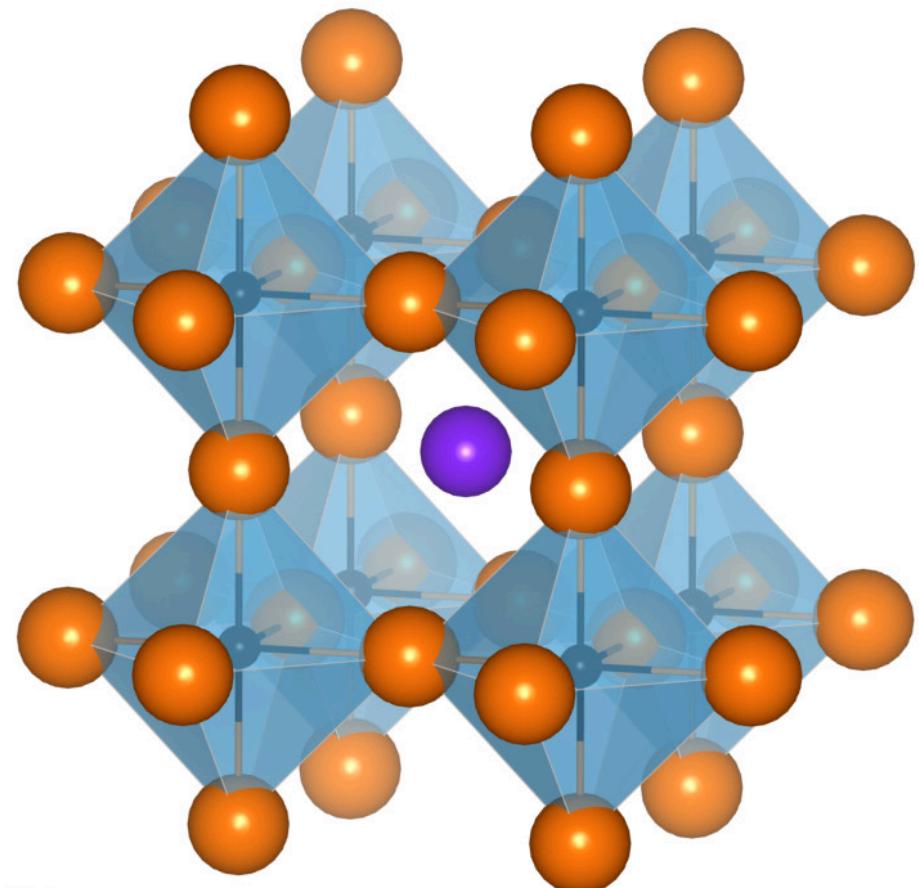
ABX₃: perovskites BaTiO₃, Pb(Zr,Ti)O₃, CH₃NH₃PbI₃

FCC derived structure; cubic parent structure, distortion in low-temperature phases (tetragonal, rhombohedral) – new properties



BaTiO₃

Black:Ti; orange: O; Purple: Ba



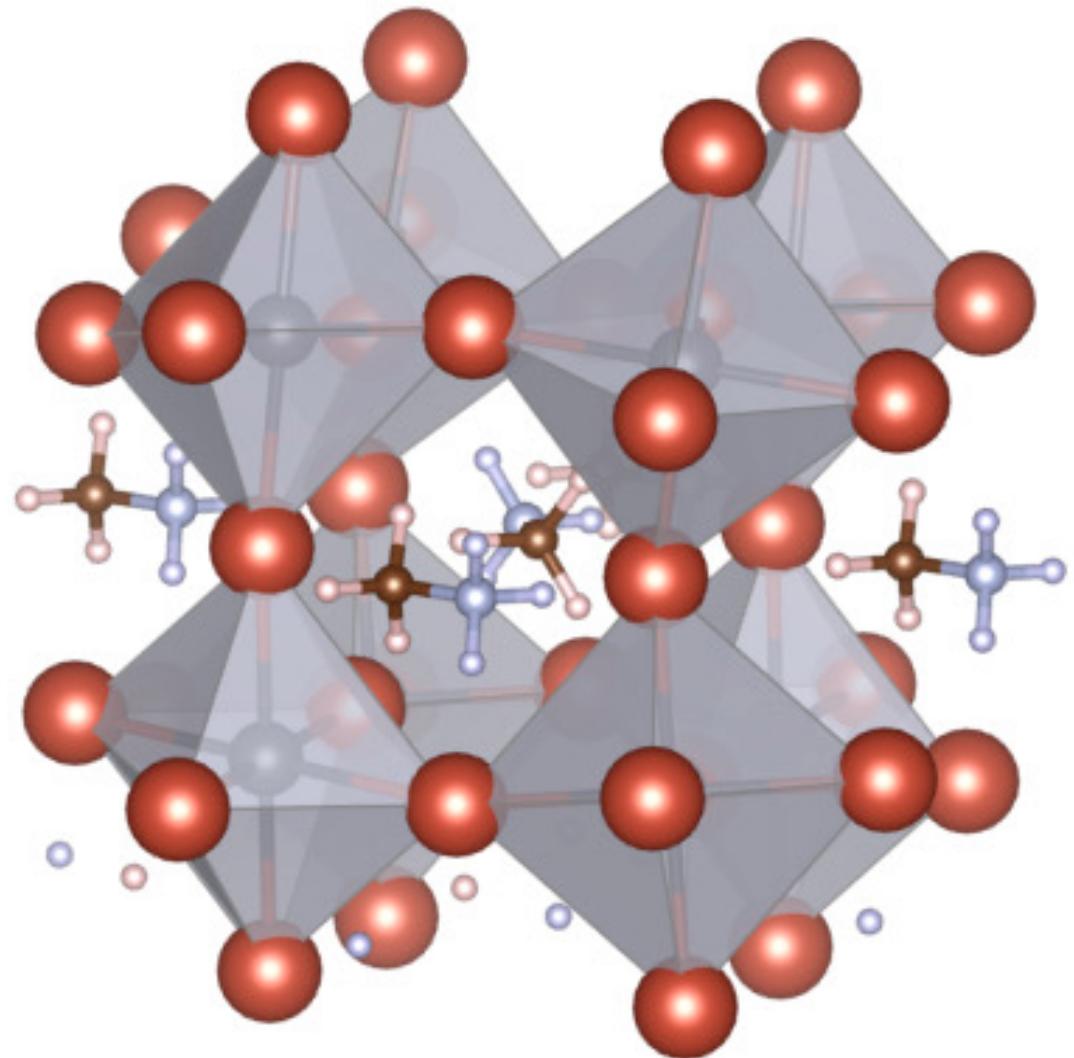
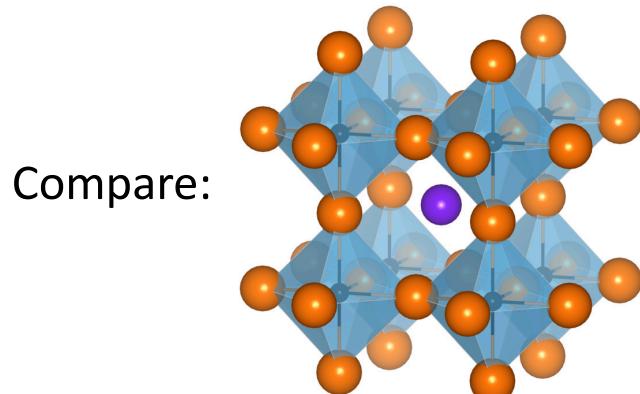
ABX₃ structures

Perovskite structure ABX₃:

CH₃NH₃PbI₃; orthorhombic structure or tetragonal or...

Methylammonium Lead Iodide
methylammonium trihalogenoplumbates

Distorted octahedra;
Large atom (Pb) in the center of octahedra,



Red: I; Black: Pb; Small atoms: H,N,C- CH₃NH₃ molecule
organic molecules are at the place of A-atoms – solar cells

Crystalline symmetry

Macroscopic crystalline symmetry and point groups

Translational symmetry in 3D

$$\vec{t} = p\vec{a}_1 + m\vec{a}_2 + n\vec{a}_3$$

translation vector

fundamental translation vectors

p, m, n

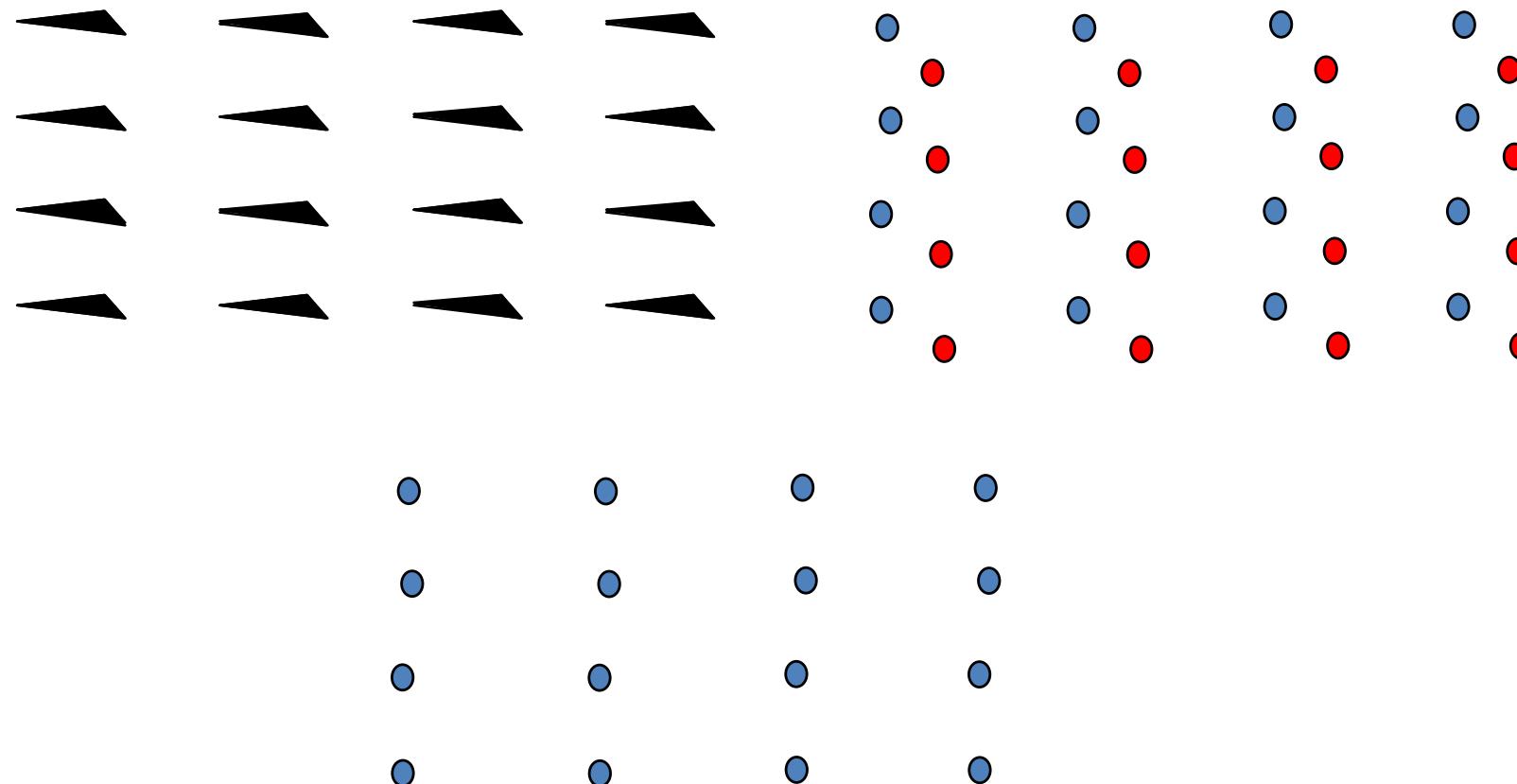
integer

Translation vector makes ***lattice (Bravais Lattice)***

$|\vec{a}_1|$ $|\vec{a}_2|$ $|\vec{a}_3|$

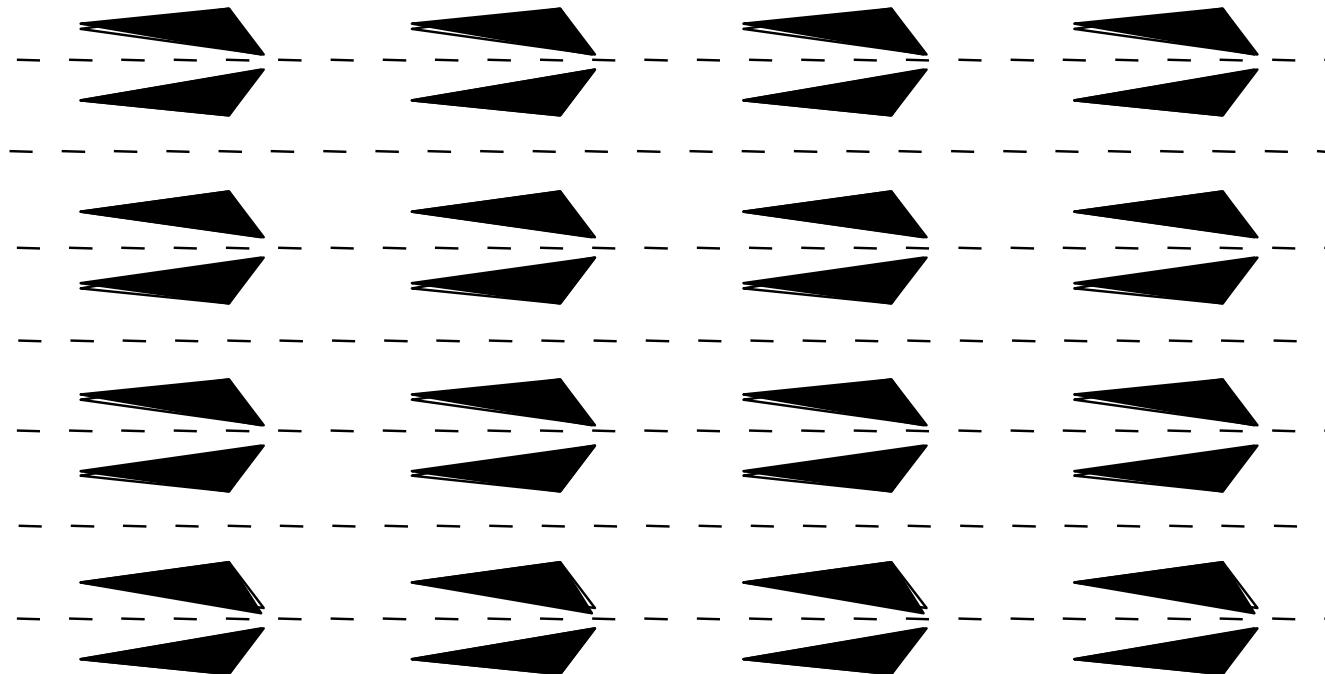
lattice constants

These crystalline structures have the same
Bravais lattice



Crystalline Symmetry

Reflections



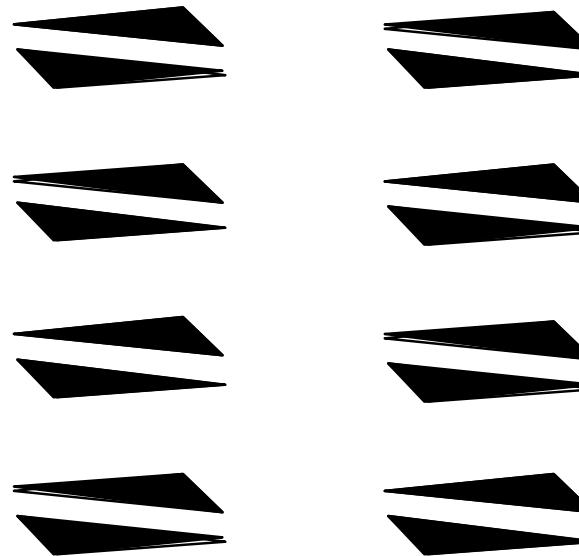
mirror planes

Crystalline Symmetry

rotation axes:

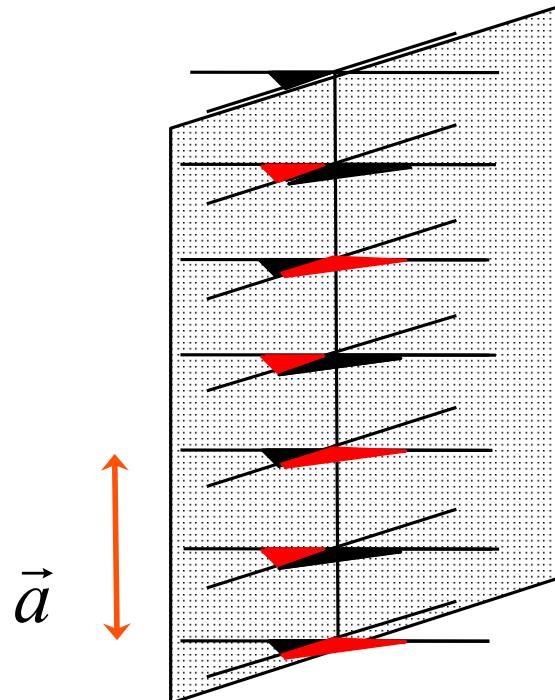
Order 2, 3, 4, 6

2-fold axes:



Crystalline Symmetry

Reflection + translation

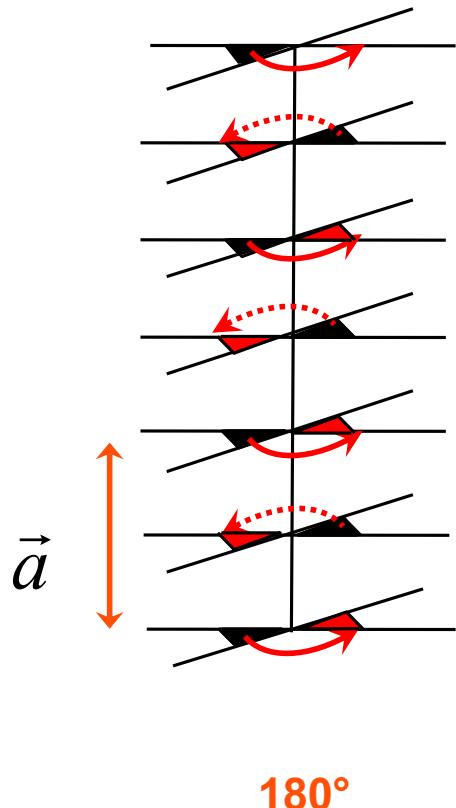


Symmetry element
= reflection + $a/2$ translation

glide plane

Crystalline Symmetry

rotation + translation



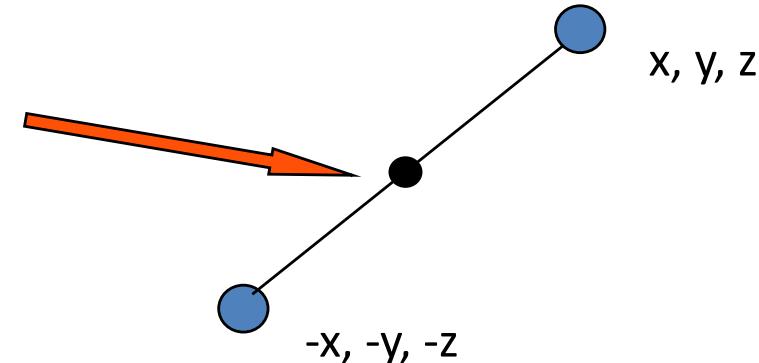
Symmetry element
= **180 ° rotation** + **$a/2$ translation**

2-fold screw axis

Elements of crystalline symmetry

translations

center of symmetry



mirror plane

glide plane

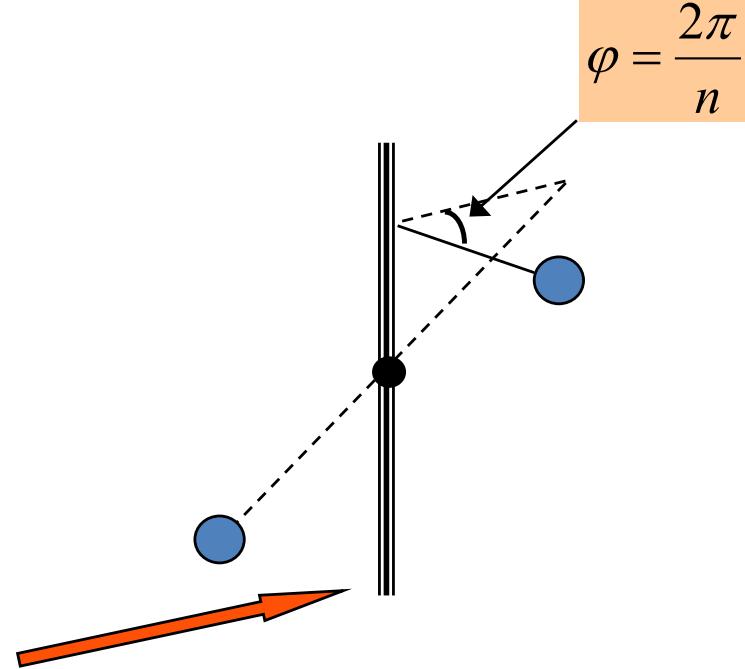
$$\varphi = \frac{2\pi}{n}$$

$n = 2, 3, 4, 6$

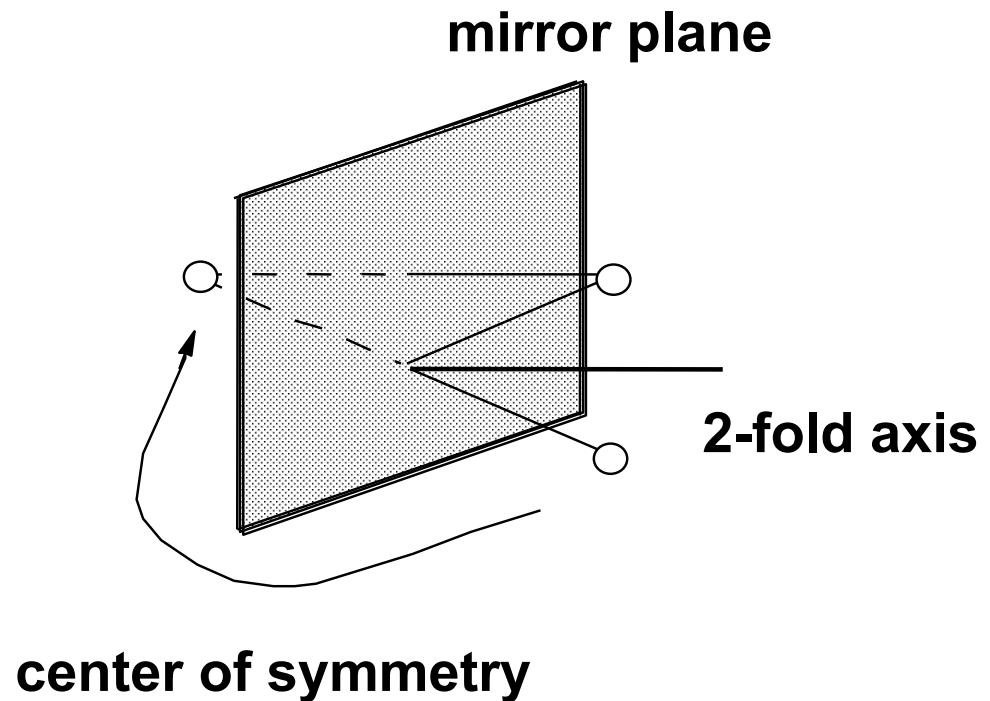
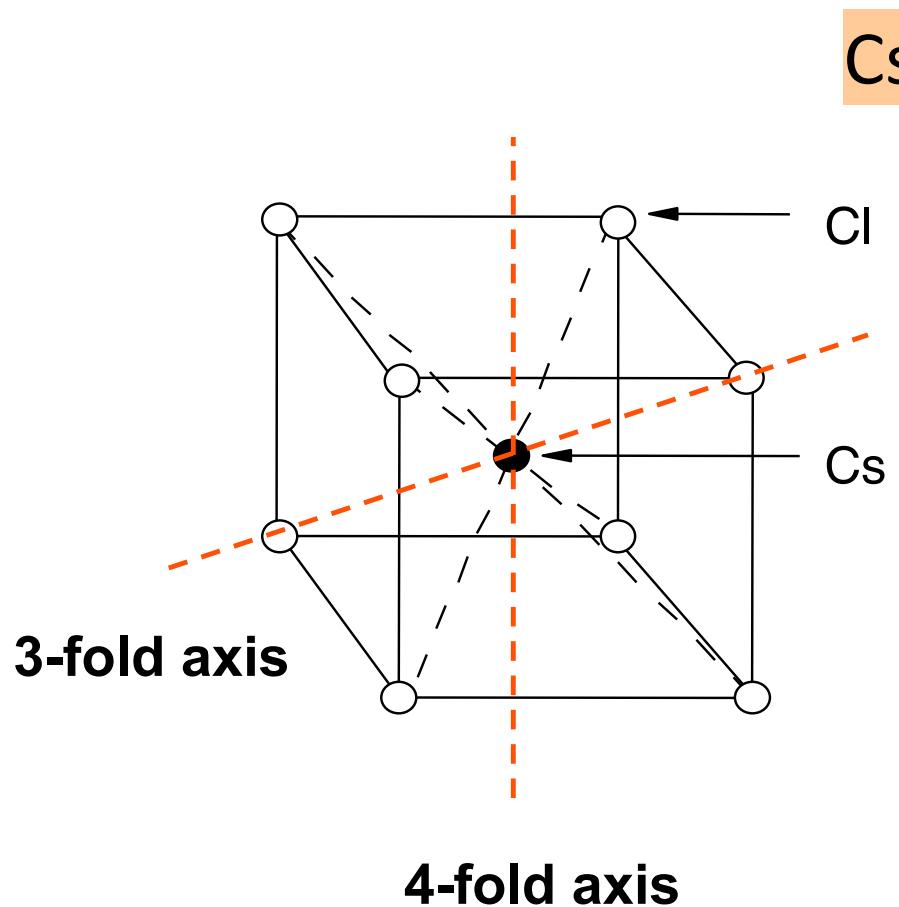
n -fold rotation axis

n - fold screw axis

n - fold inversion axis



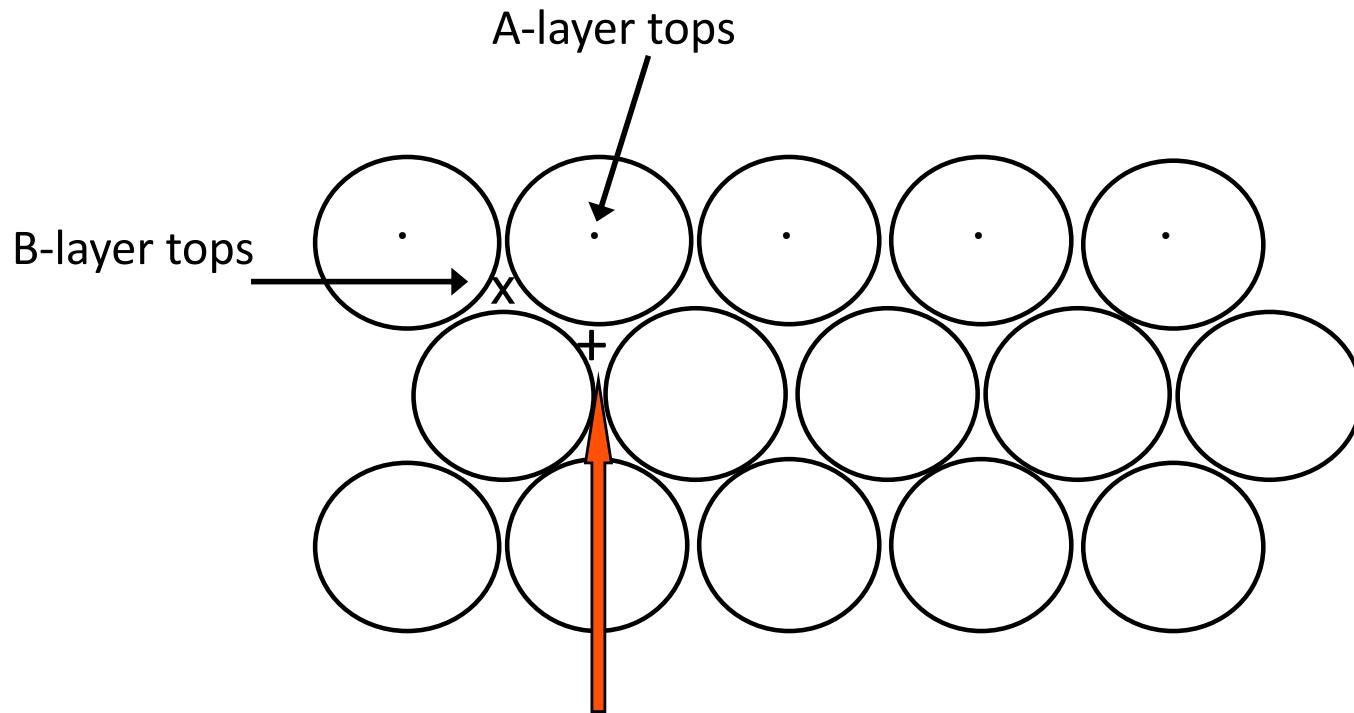
Elements of crystalline symmetry



- Not all symmetry elements are independent
- A 2-fold rotoinversion axis is equivalent to a mirror plane, that is why a symbol “bar 2” is rarely used

Elements of crystalline symmetry

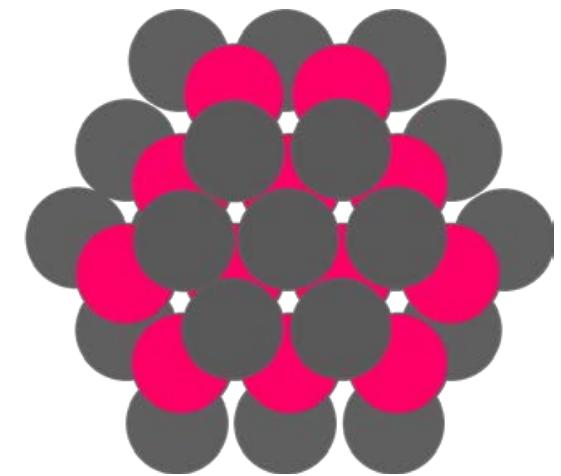
Hexagonal Close-packed Structure (hcp)



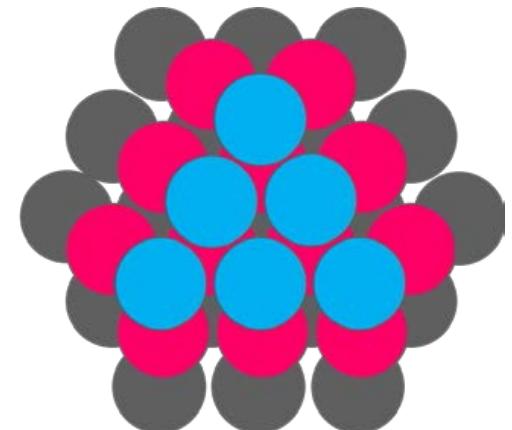
Other symmetry elements:

- mirror planes (passing through atoms)
- glide planes (passing between the atoms)

Stacking A + B + A



Stacking A + B + C



Elements of MACroscopic crystalline symmetry

Fine symmetry

translations

center of symmetry

mirror plane

glide plane

n-fold rotation axis

n- fold screw axis

n- fold inversion axis

MACroscopic symmetry

center of symmetry

mirror plane

n-fold rotation axis

n- fold inversion axis

Elements of MACroscopic crystalline symmetry

Fine symmetry

MACroscopic symmetry

230 space groups

32 point groups

group = set of symmetry elements of an object

Space group → point group (example)

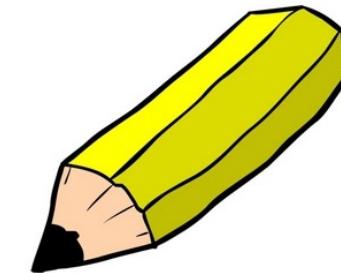
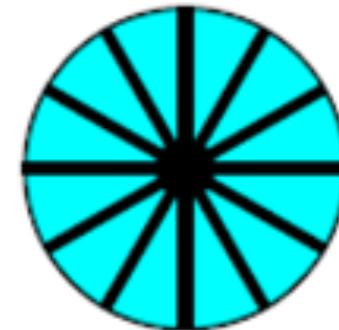
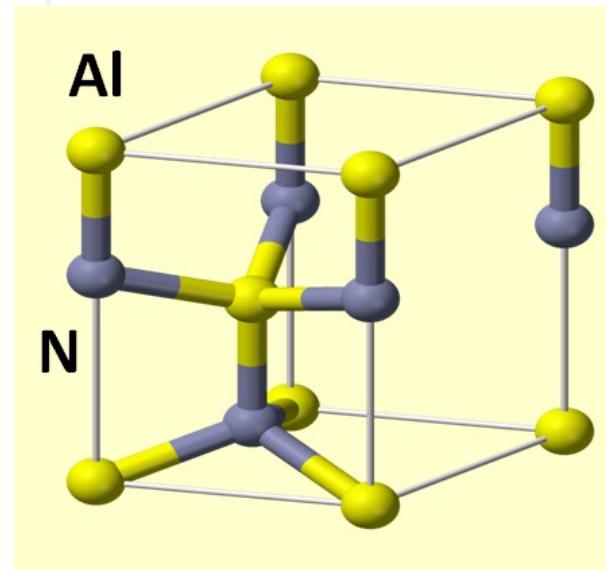
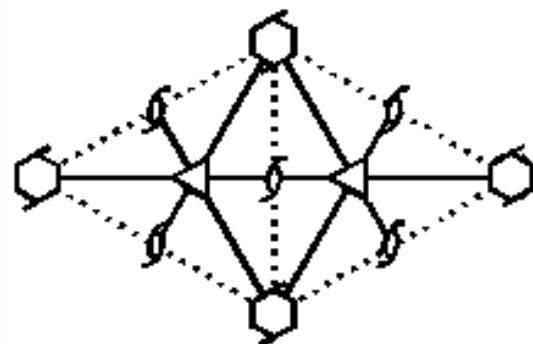


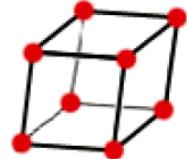
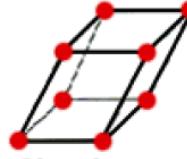
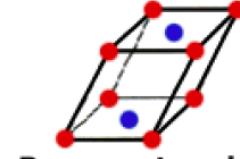
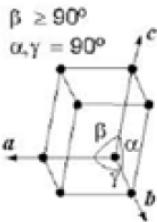
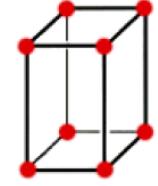
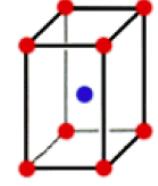
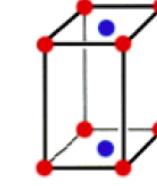
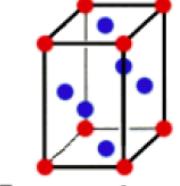
**AlN-based
filter**

P6₃mc



6mm

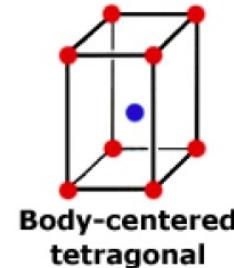
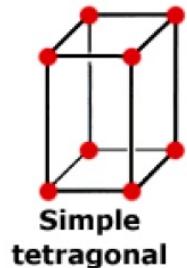


Crystal system	The 14 Bravais lattices	Defining symmetry
Triclinic $a \neq b \neq c$ $\alpha \neq \beta \neq \gamma \neq 90^\circ$	 Triclinic	1-fold axis
Monoclinic $a \neq b \neq c$; $\alpha = \gamma = 90^\circ; \beta \neq 90^\circ$	 Simple Monoclinic  Base-centered monoclinic 	2-fold axis
Orthorhombic $a \neq b \neq c$ $\alpha = \beta = \gamma = 90^\circ$	 Simple orthorhombic  Body-centered orthorhombic  Base-centered orthorhombic  Face-centered orthorhombic	3 x 2 fold axis

Tetragonal

$a = b \neq c$

$\alpha = \beta = \gamma = 90^\circ$

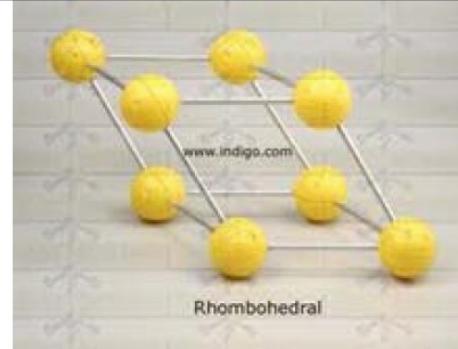
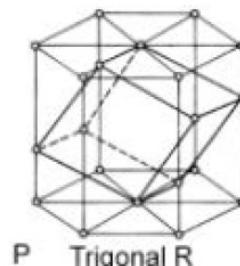
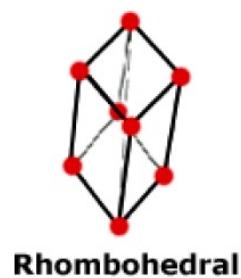


4-fold axis

Rhombohedral (trigonal)

$a = b = c$

$\alpha = \beta = \gamma \neq 90^\circ$

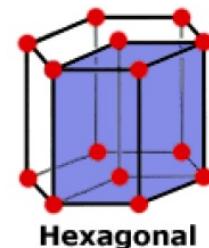


3-fold axis

Hexagonal

$a = b \neq c$

$\alpha = \beta = 90^\circ, \gamma = 120^\circ$

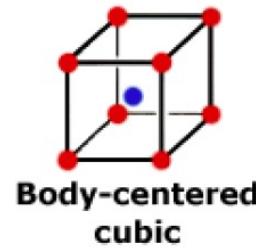
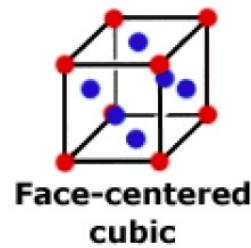
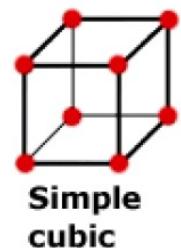


6-fold axis

Cubic

$a = b = c$

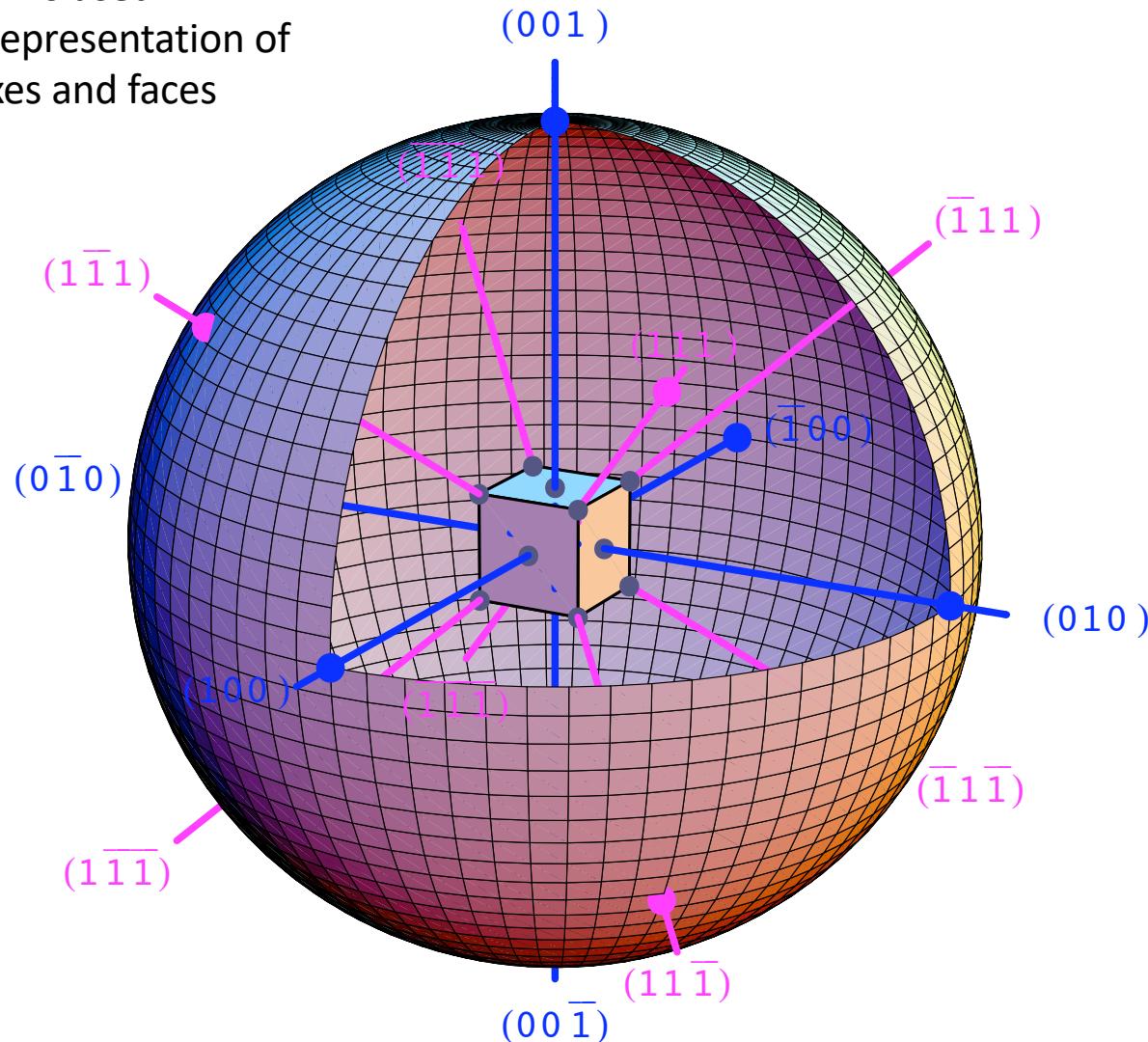
$\alpha = \beta = \gamma = 90^\circ$



4 x 3-fold axis

Stereographic Projection

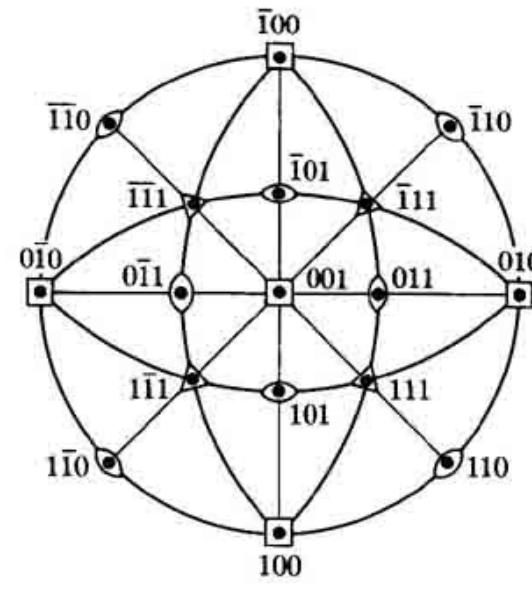
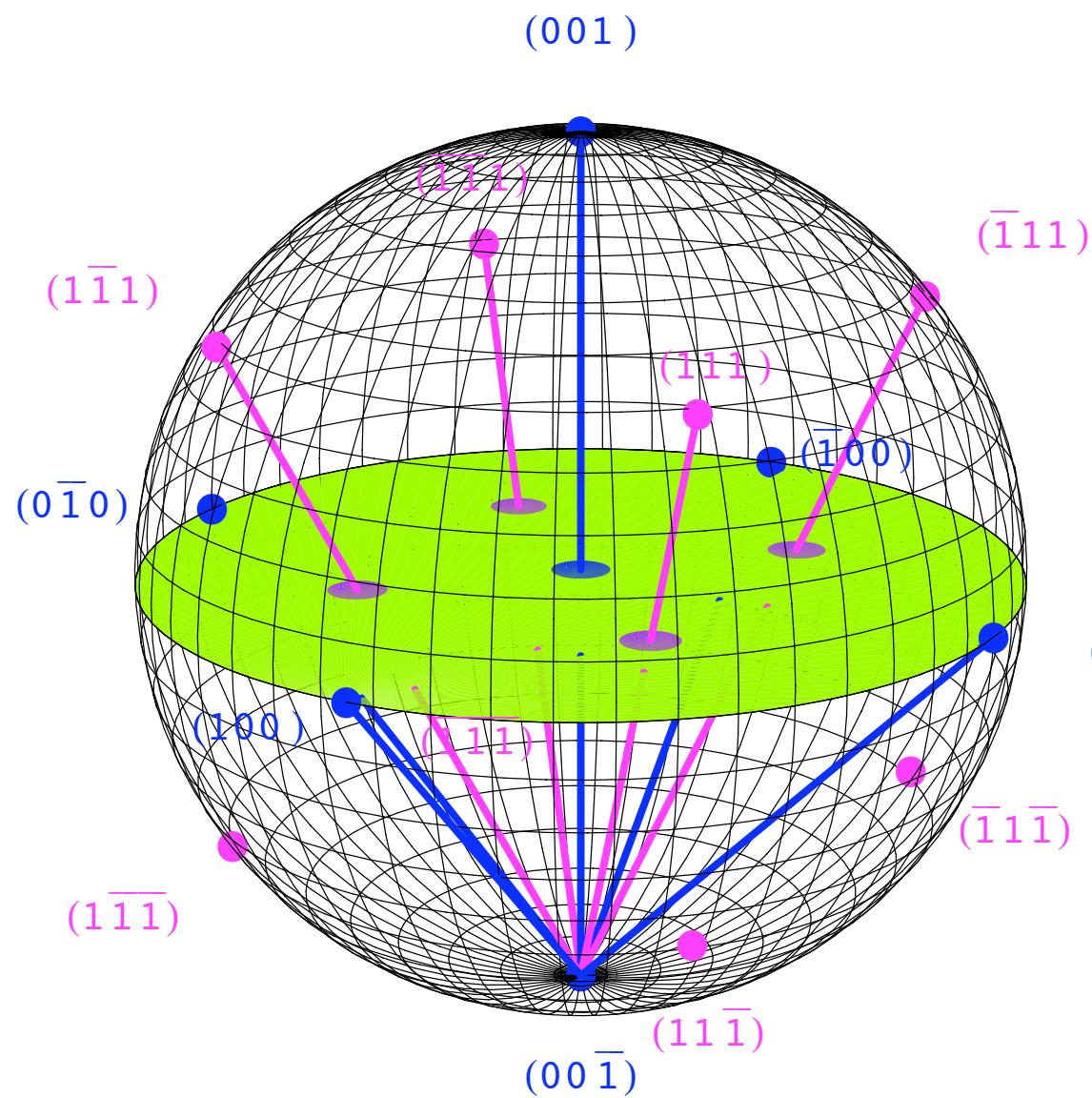
Stereographic projection is used in crystallography for 2D representation of orientation of crystal axes and faces



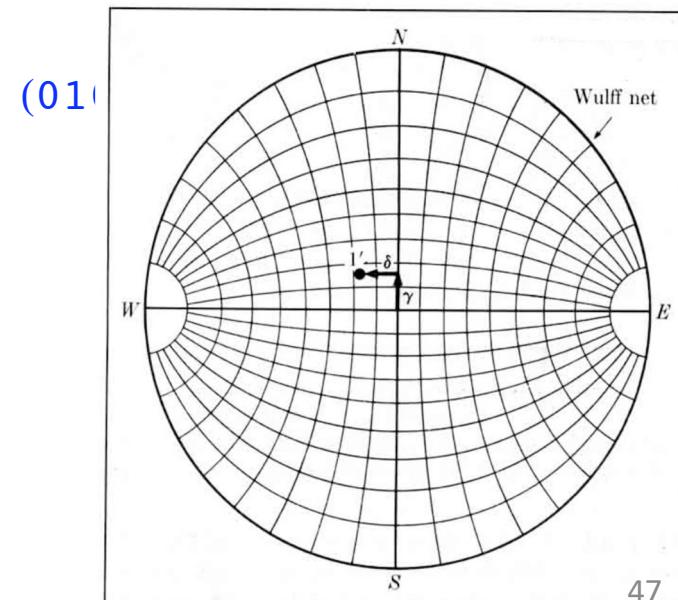
For building up stereographic projection the crystal is imagined to be at the centre of a sphere (called hereafter a stereographic sphere). The normals to the faces are imagined to radiate from the centre and to intersect the sphere in an array of points...

Now to we represent this 3D array of points in 2D as shown below i.e. we project them on the equatorial plane of the stereographic sphere using south pole as a projection center

Wulff net is used for measuring angles between the poles



(a) cubic crystal:



Systems

32 point groups

Triclinic

 1 (C ₁)			 1-bar (C ₁)				
 2 (C ₂)				 m (C _s)		 2/m (C _{2h})	
				 mm2 (C _{2v})	 222 (D ₂)	 mmm (D _{2h})	
 3 (C ₃)			 3-bar (S ₆)	 3m (C _{3v})	 32 (D ₃)	 3-bar m (D _{3d})	
 4 (C ₄)	 4-bar (S ₄)	 4-bar 2m (D _{2d})	 4/m (C _{4h})	 4mm (C _{4v})	 422 (D ₄)	 4/mmm (D _{4h})	
 6 (C ₆)	 6-bar (C _{3h})	 6-bar 2m (D _{3h})	 6/m (C _{6h})	 6mm (C _{6v})	 622 (D ₆)	 6/mmm (D _{6h})	
 23 (T)			 m3-bar (T _h)	 43m (T _d)	 432 (O)	 m3-bar m (O _h)	

no axis



inversion

rotation axes

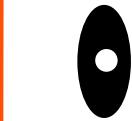
2 (2-bar)



2 ⊥ 2 ⊥ 2 (2-bar)



3 (3-bar)



4 (4-bar)



6 (6-bar)

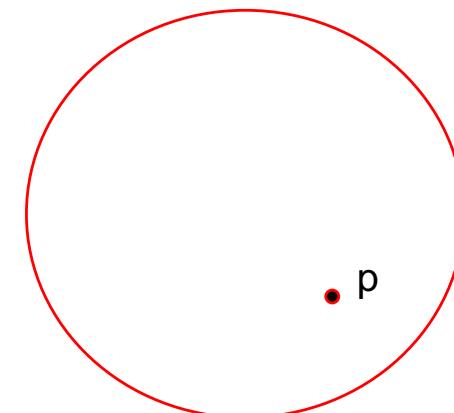
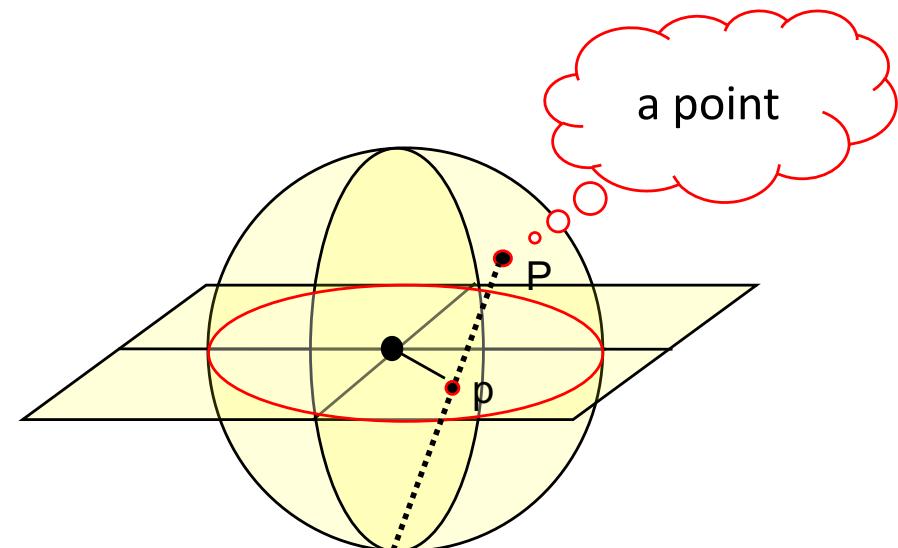


inversion axes

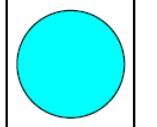
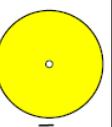
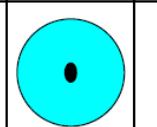
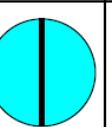
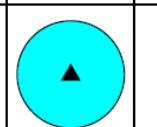
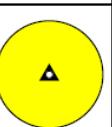
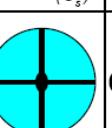
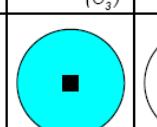
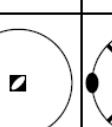
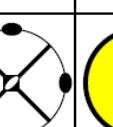
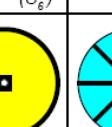
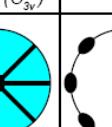
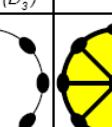
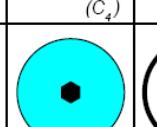
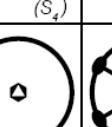
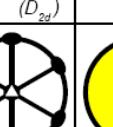
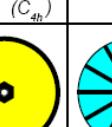
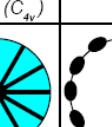
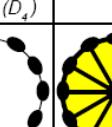
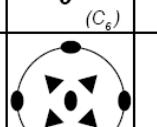
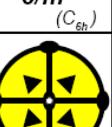
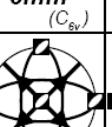
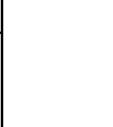
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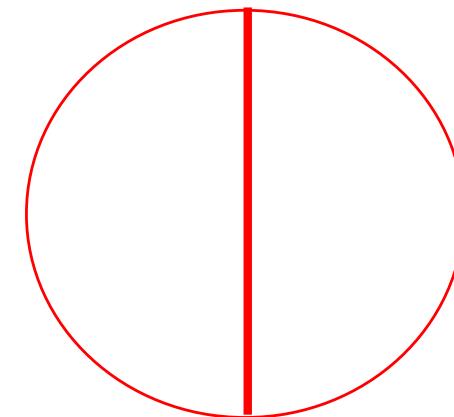
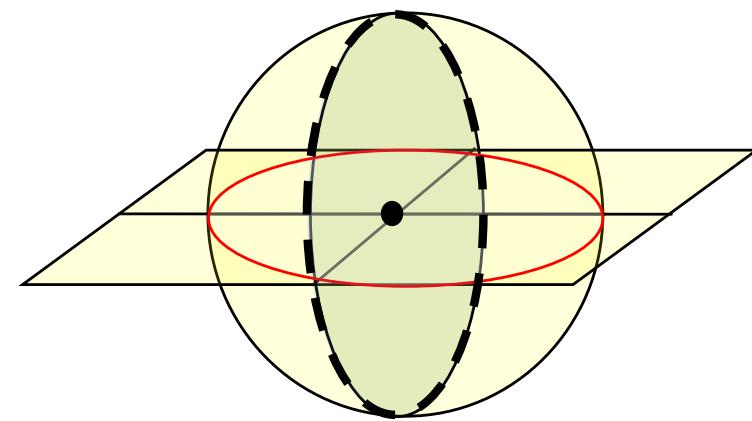
Stereographic projections

1 (C_1)			1̄ ($C_{\bar{1}}$)			
2 (C_2)				m (C_5)		$2/m$ (C_{2h})
				$mm2$ (C_{2v})	222 (D_2)	mmm (D_{2h})
3 (C_3)			3̄ (S_6)	$3m$ (C_{3v})	32 (D_3)	$\bar{3}m$ (D_{3d})
4 (C_4)	$\bar{4}$ (S_4)	$\bar{4}2m$ (D_{2d})	$4/m$ (C_{4h})	$4mm$ (C_{4v})	422 (D_4)	$4/mmm$ (D_{4h})
6 (C_6)	$\bar{6}$ (C_{3h})	$\bar{6}2m$ (D_{3h})	$6/m$ (C_{6h})	$6mm$ (C_{6v})	622 (D_6)	$6/mmm$ (D_{6h})
23 (T)			$m\bar{3}$ (T_h)	$\bar{4}3m$ (T_d)	432 (O)	$m\bar{3}m$ (O_h)



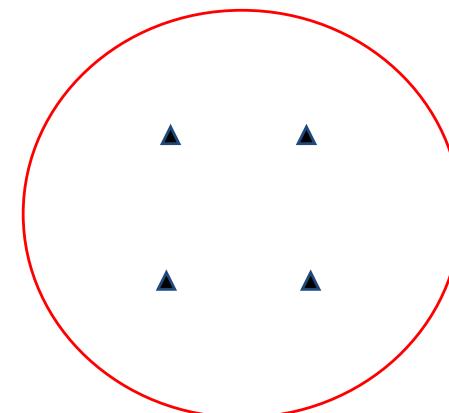
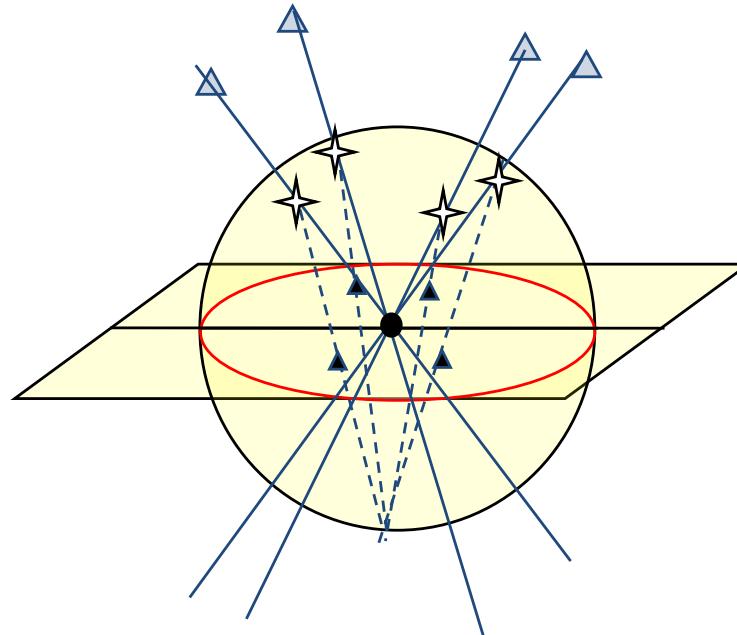
Stereographic projections

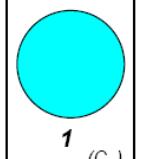
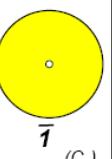
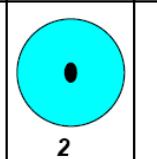
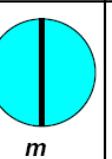
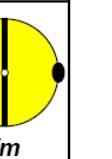
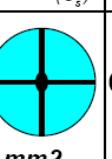
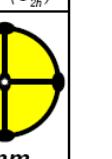
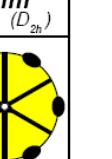
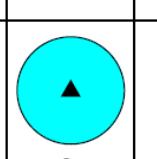
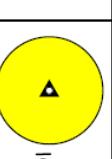
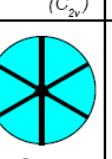
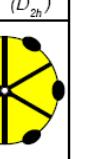
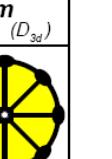
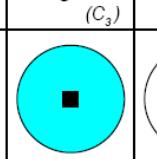
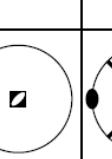
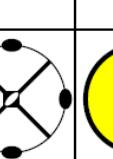
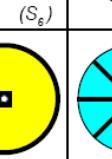
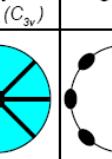
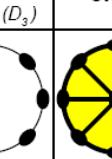
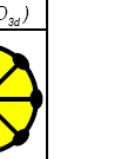
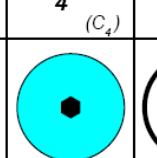
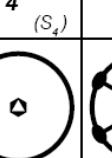
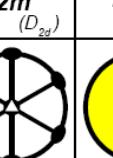
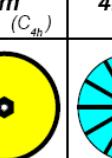
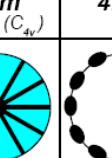
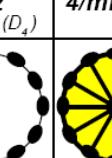
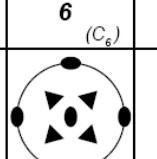
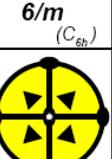
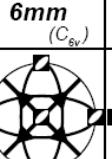
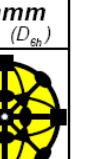
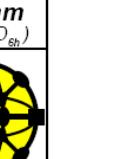


Stereographic projections

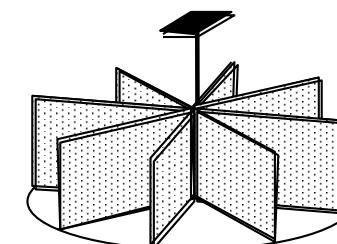
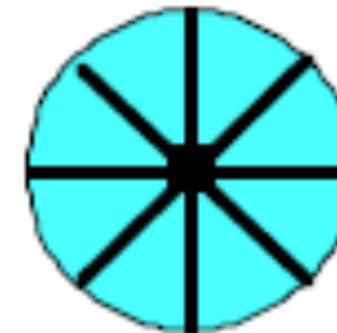
1 (C_1)			1̄ ($C_{\bar{1}}$)			
2 (C_2)				m (C_5)		$2/m$ (C_{2h})
				$mm2$ (C_{2v})	222 (D_2)	mmm (D_{2h})
3 (C_3)			3̄ (S_6)	$3m$ (C_{3v})	32 (D_3)	$\bar{3}m$ (D_{3d})
4 (C_4)	$\bar{4}$ (S_4)	$\bar{4}2m$ (D_{2d})	$4/m$ (C_{4h})	$4mm$ (C_{4v})	422 (D_4)	$4/mmm$ (D_{4h})
6 (C_6)	$\bar{6}$ (C_{3h})	$\bar{6}2m$ (D_{3h})	$6/m$ (C_{6h})	$6mm$ (C_{6v})	622 (D_6)	$6/mmm$ (D_{6h})
23 (T)			$m\bar{3}$ (T_h)	$\bar{4}3m$ (T_d)	432 (O)	$m\bar{3}m$ (O_h)



Stereographic projections

 1 (C ₁)			 1-bar (C ₁)			
 2 (C ₂)			 m (C _s)			 2/m (C _{2h})
			 mm2 (C _{2v})	 222 (D ₂)		 mmm (D _{2h})
 3 (C ₃)			 3-bar (S ₆)	 3m (C _{3v})	 32 (D ₃)	 3-bar m (D _{3d})
 4 (C ₄)	 4-bar (S ₄)	 42m (D _{2d})	 4/m (C _{4h})	 4mm (C _{4v})	 422 (D ₄)	 4/mmm (D _{4h})
 6 (C ₆)	 6-bar (C _{3h})	 6-bar 2m (D _{3h})	 6/m (C _{6h})	 6mm (C _{6v})	 622 (D ₆)	 6/mmm (D _{6h})
 23 (T)			 m-bar 3 (T _h)	 43m (T _d)	 432 (O)	 m-bar 3m (O _h)

4mm, C_{4v}



Systems

	Si	R/I	R/I-P	In	P	Ro	Su
Triclinic							
Monoclinic							
Orthorhombic							
Trigonal							
Tetragonal							
Hexagonal							
Cubic							

Types



Si = simple

R/I = inversion axes

R/I - P = inversion axes + planes

I = inversion

P = planes

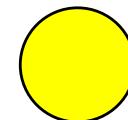
Ro = rotations only

Su = super

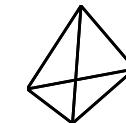
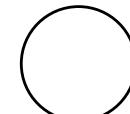
 1 (C ₁)			 1-bar (C ₁)			
 2 (C ₂)				 m (C ₅)		 2/m (C _{2h})
				 mm2 (C _{2v})	 222 (D ₂)	 mmm (D _{2h})
 3 (C ₃)			 3-bar (S ₆)	 3m (C _{3v})	 32 (D ₃)	 3-bar m (D _{3d})
 4 (C ₄)	 4-bar (S ₄)	 4-bar 2m (D _{2d})	 4/m (C _{4h})	 4mm (C _{4v})	 422 (D ₄)	 4/mmm (D _{4h})
 6 (C ₆)	 6-bar (C _{3h})	 6-bar 2m (D _{3h})	 6/m (C _{6h})	 6mm (C _{6v})	 622 (D ₆)	 6/mmm (D _{6h})
 23 (T)			 m-bar 3 (T _h)	 43m (T _d)	 432 (O)	 m-bar 3m (O _h)

Polarity

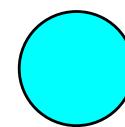
Centrosymmetric



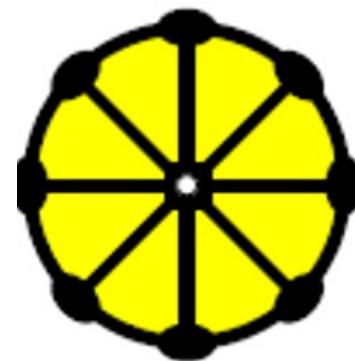
Non-centrosymmetric Non-polar



Non-centrosymmetric Polar



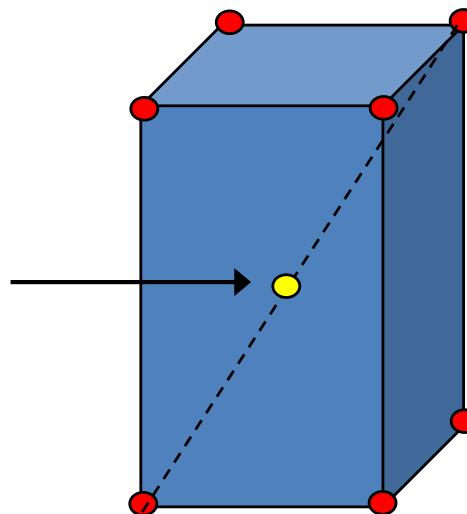
$4/mmm, D_{4h}$



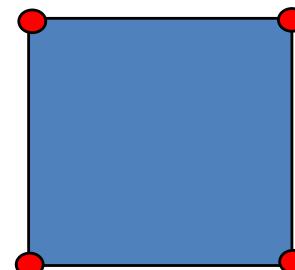
Tetragonal

Centrosymmetric

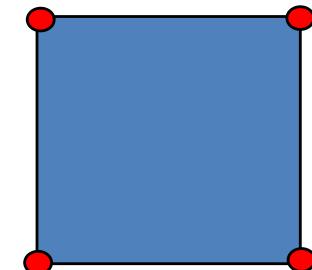
center of
symmetry



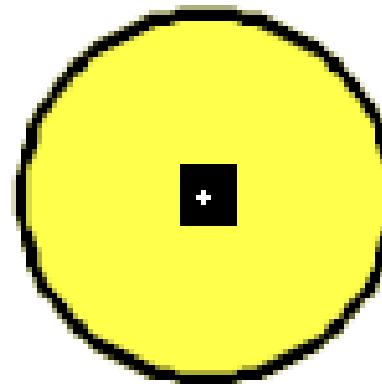
top



bottom



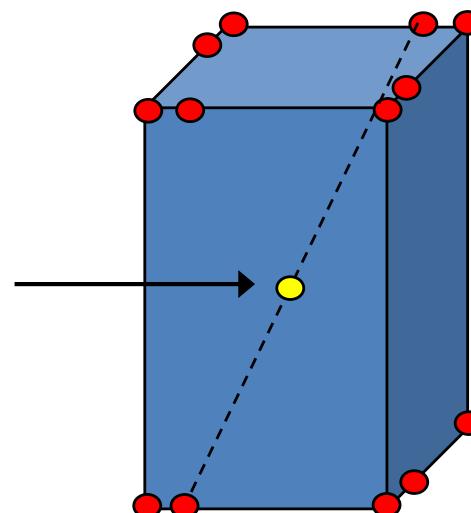
$4/m, C_{4h}$



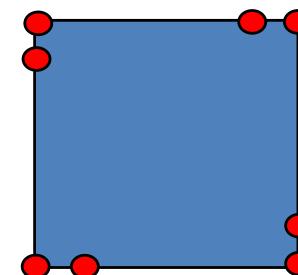
Tetragonal

Centrosymmetric

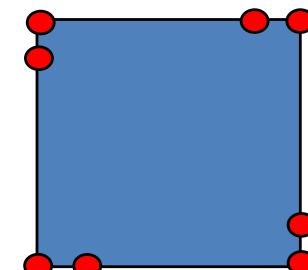
center of
symmetry



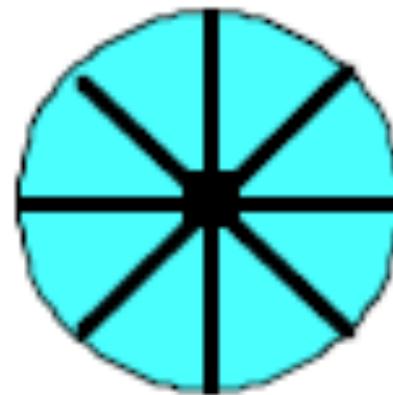
top



bottom

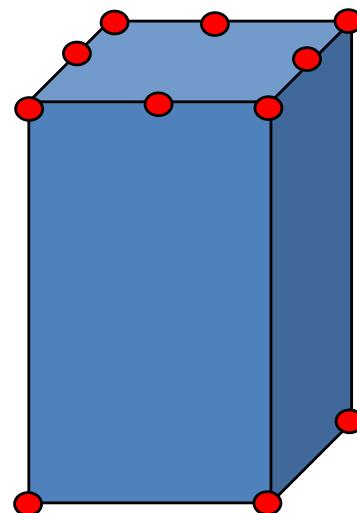


$4mm, C_{4v}$



Tetragonal

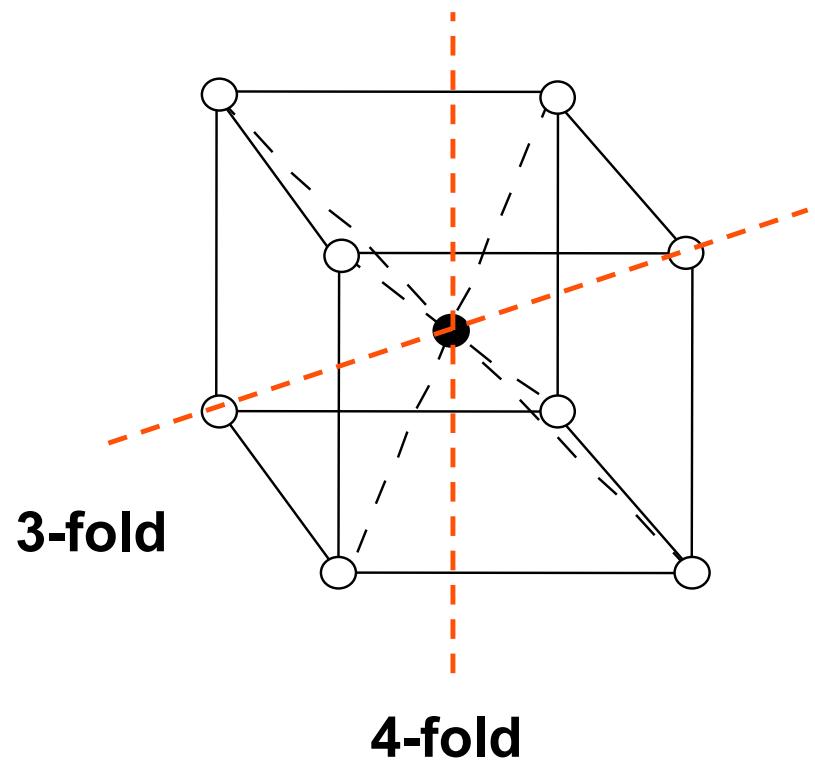
Non-centrosymmetric



top

bottom

Symmetry of CsCl structure



3-fold

4-fold

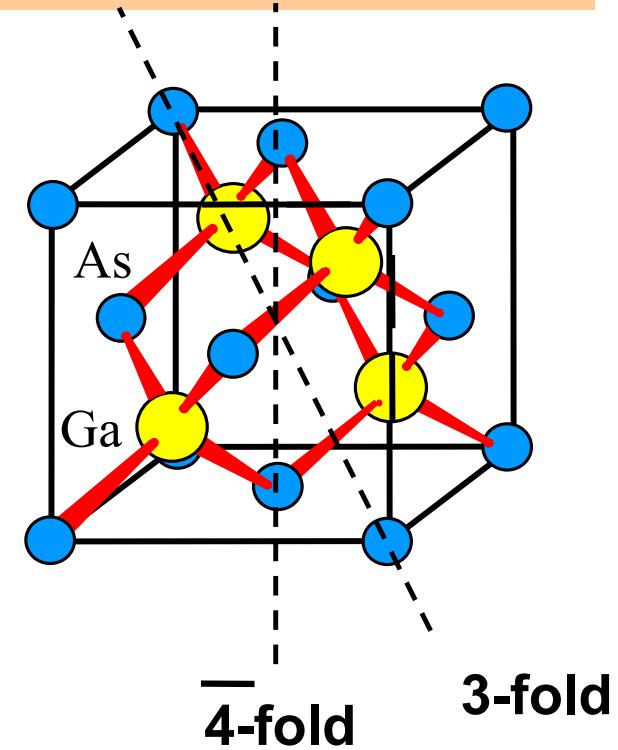
Cubic

$m\bar{3}m$

Centrosymmetric

 1 (C ₁)			 1-bar (C ₁)			
 2 (C ₂)				 m (C _s)	 2/m (C _{2h})	
 3 (C ₃)			 3-bar (S ₆)	 mm2 (C _{2v})	 222 (D ₂)	 mmm (D _{2h})
 4 (C ₄)	 4-bar (S ₄)	 4-bar 2m (D _{2d})	 4/m (C _{4h})	 4mm (C _{4v})	 422 (D ₄)	 4/mmm (D _{4h})
 6 (C ₆)	 6-bar (C _{3h})	 6-bar 2m (D _{3h})	 6/m (C _{6h})	 6mm (C _{6v})	 622 (D ₆)	 6/mmm (D _{6h})
 23 (T)			 m3-bar (T _h)	 43m (T _d)	 432 (O)	 m3m-bar (O _h)

Symmetry of ZnS zinc blende structure

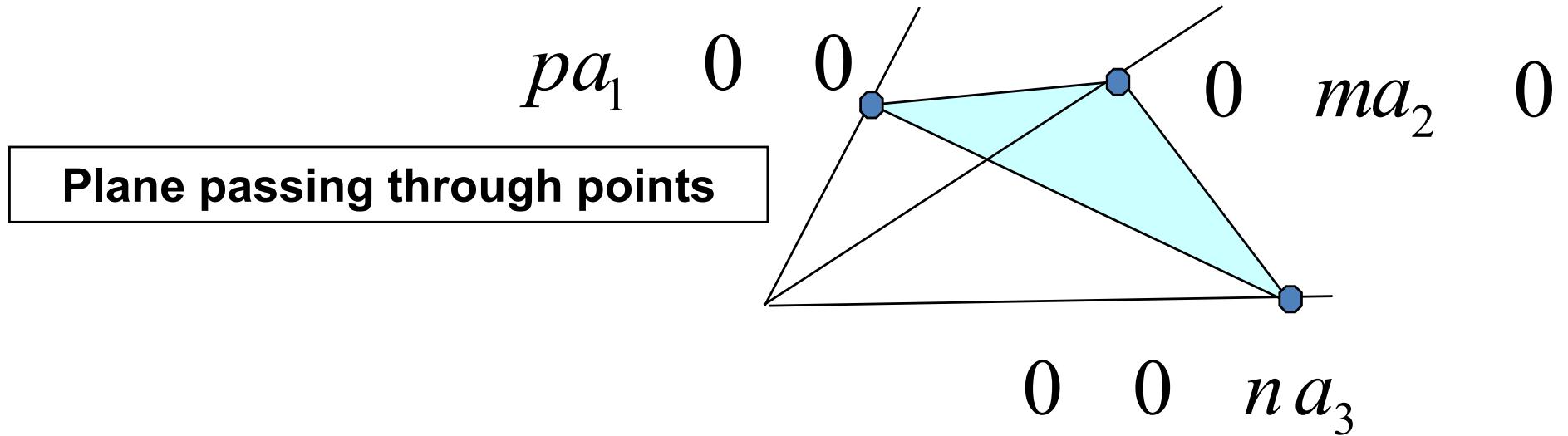


Cubic

$\bar{4}3m$

Non-centrosymmetric

Miller indices for planes

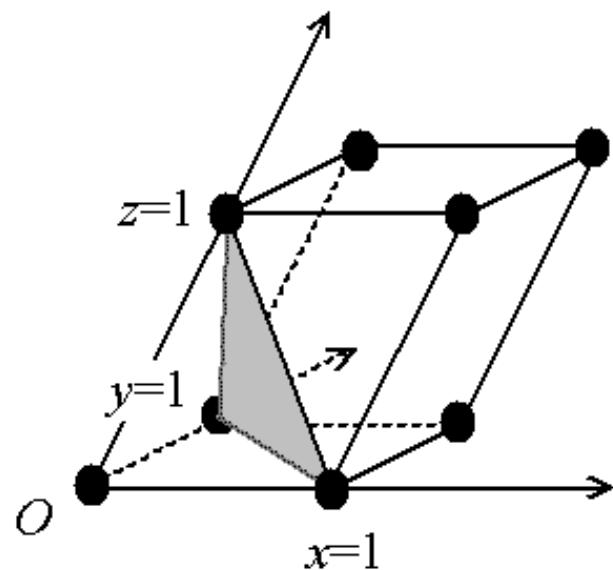


$$(h \ k \ l) = \left(\frac{b}{p} \ \frac{b}{m} \ \frac{b}{n} \right)$$

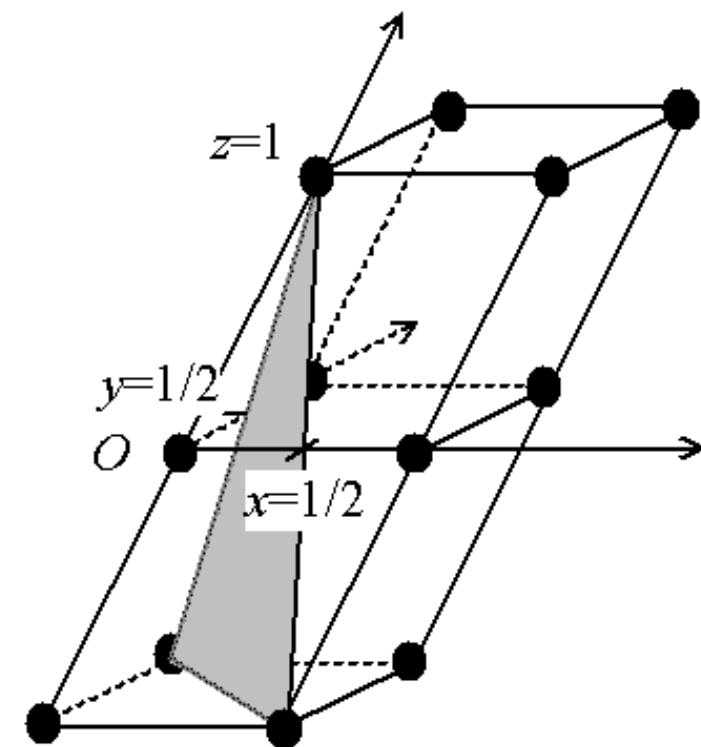
with minimal possible b

Miller indices for planes

plan (111)



plan (221)



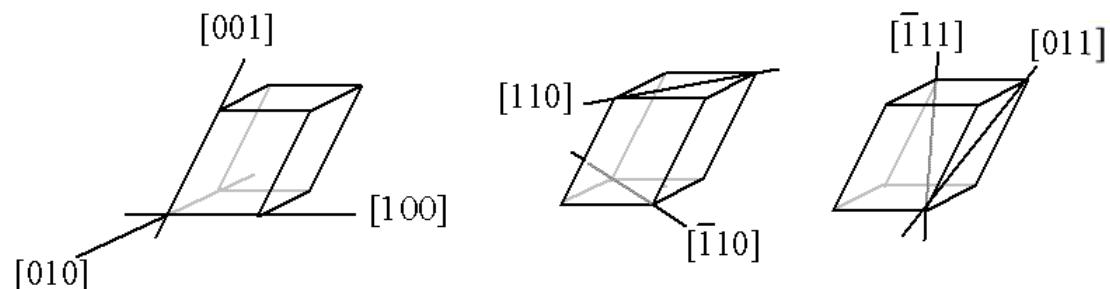
Miller indices for directions

$$\vec{t} = p\vec{a}_1 + m\vec{a}_2 + n\vec{a}_3$$

$$[h \ k \ l] = \begin{bmatrix} p & m & n \\ b & b & b \end{bmatrix} \quad -1 \Rightarrow \bar{1}$$

p, m, n, h, k, l, b with maximal possible b

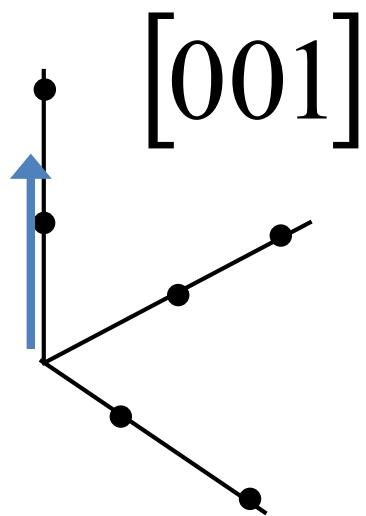
integer



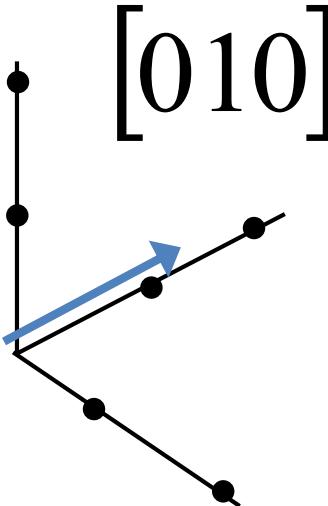
Miller indices for directions

Directions equivalent by symmetry =

**Directions that can be obtained from a given direction
by application of the point symmetry operations of the
structure**



symmetry



$m\bar{3}m$

$[001]$

$[00\bar{1}]$

$[010]$

$[0\bar{1}0]$

$[100]$

$[\bar{1}00]$

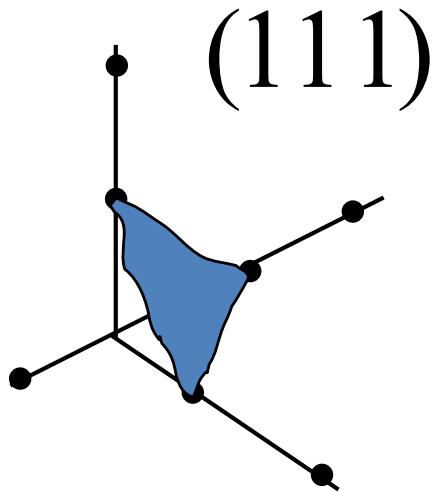
$\langle 100 \rangle$



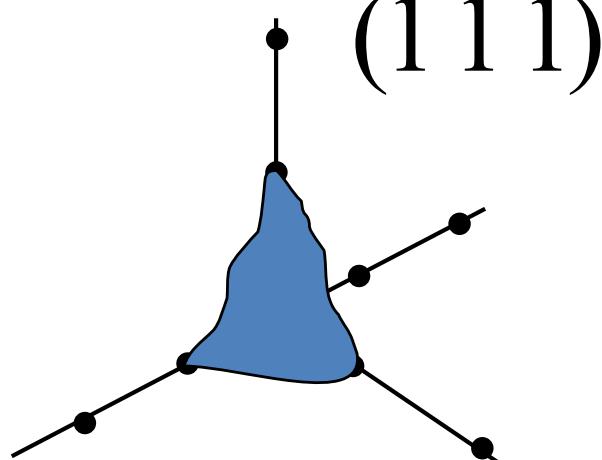
Miller indices for planes

Planes equivalent by symmetry =

**Planes that can be obtained from a given plane
by application of the point symmetry operations of the
structure**



symmetry

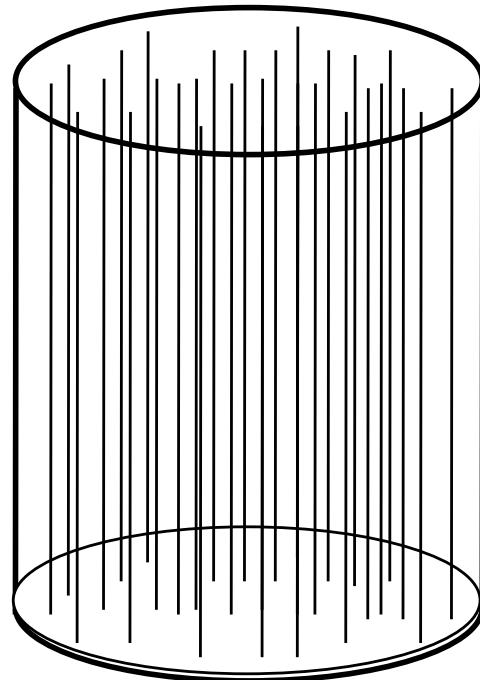


$m\bar{3}m$

(111)
(1-11)
(11-1)
(-111)
(-1-11)
(-11-1)
(1-1-1)
(-1-1-1)

$\{1\bar{1}1\}$

Macroscopic symmetry of non-crystalline materials

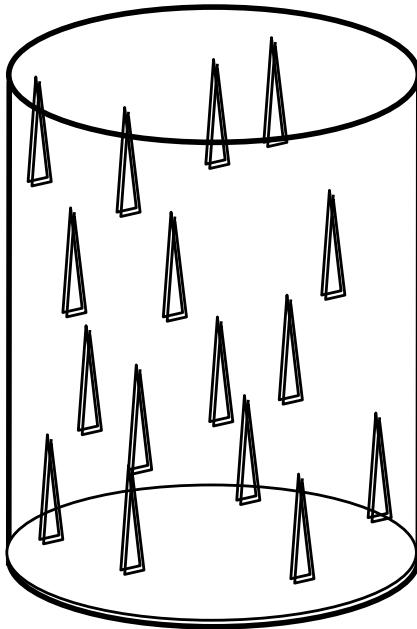


$$\frac{\infty}{m} m$$

Curie group

**Symmetry operations link
macroscopically indistinguishable
states**

Macroscopic symmetry of non-crystalline materials

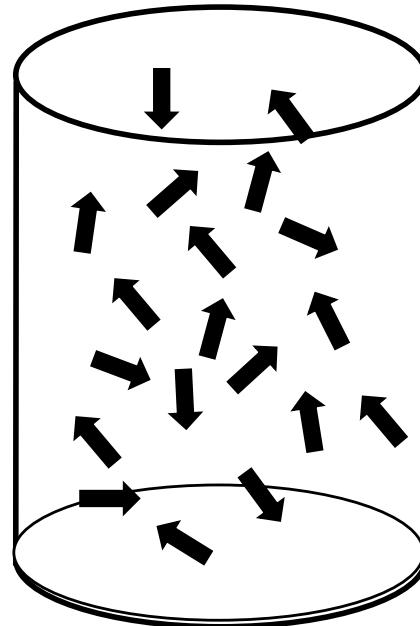


∞m

Curie group

**Symmetry operations link
macroscopically indistinguishable
states**

Macroscopic symmetry of non-crystalline materials



$\infty\infty m$

Curie group

**Symmetry operations link
macroscopically indistinguishable
states**

Macroscopic symmetry of non-crystalline materials

Curie group

$\infty\infty m$

∞ / mm

∞m

$\infty\infty$

∞

$\infty 2$

∞ / m

Geometrical bodies



sphere



cylinder

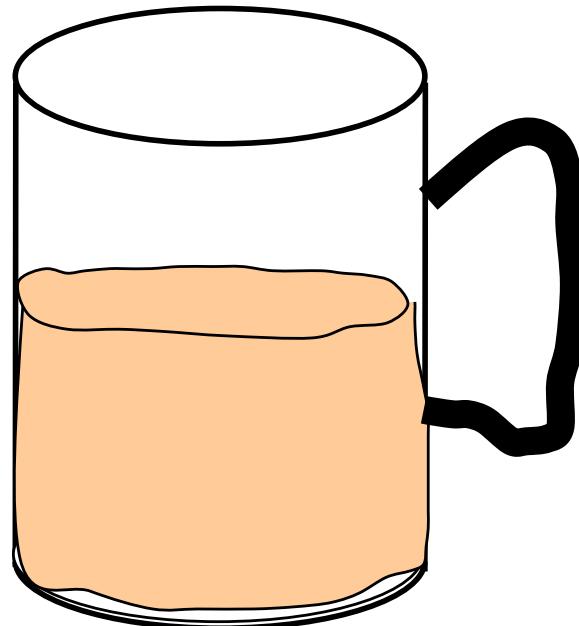


cone

?

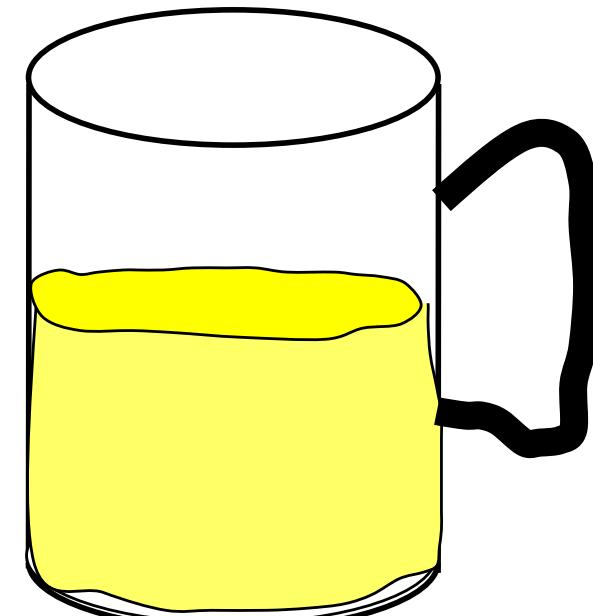
Macroscopic symmetry of non-crystalline materials

$\infty\infty m$



tee without sugar

$\infty\infty$



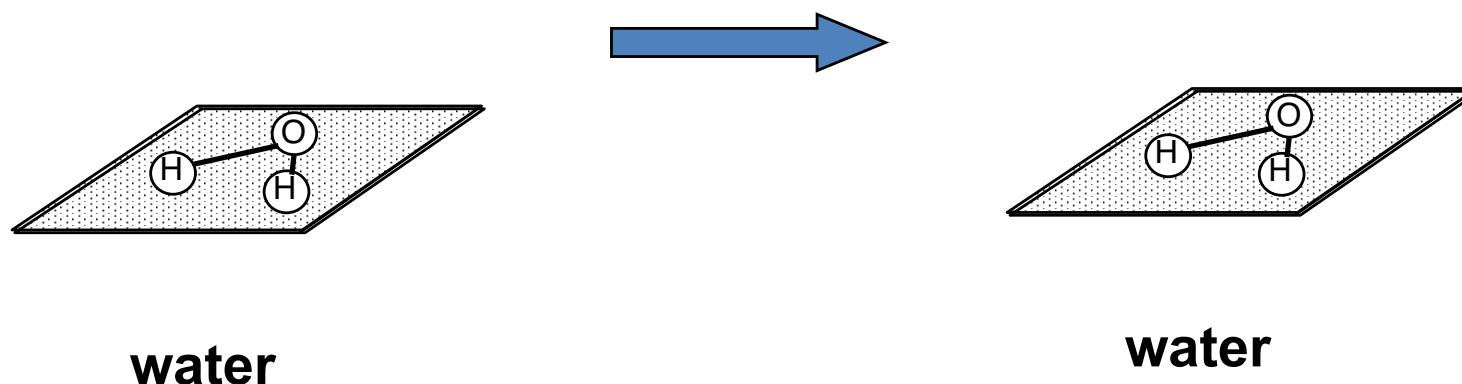
tee with sugar

Macroscopic symmetry of non-crystalline materials

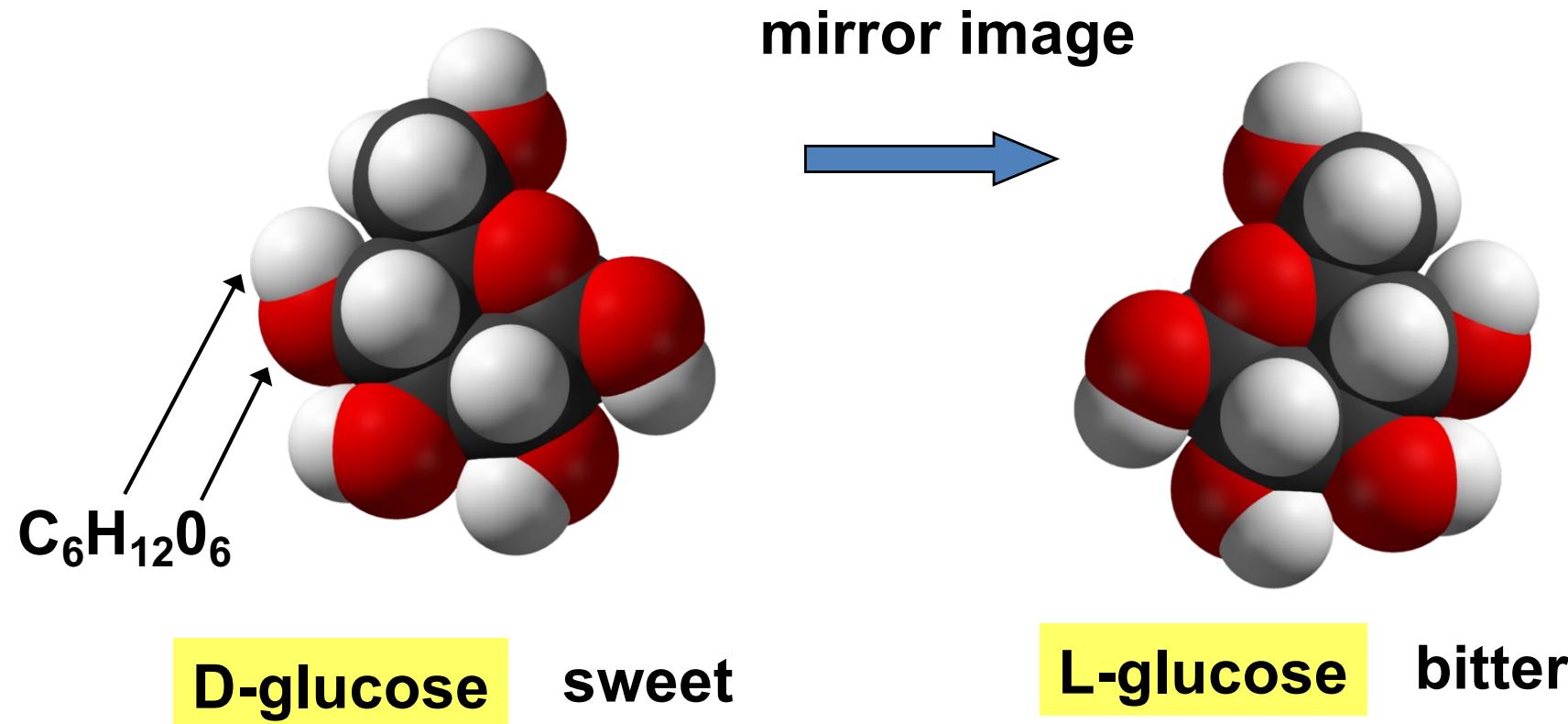
molecular water



mirror image



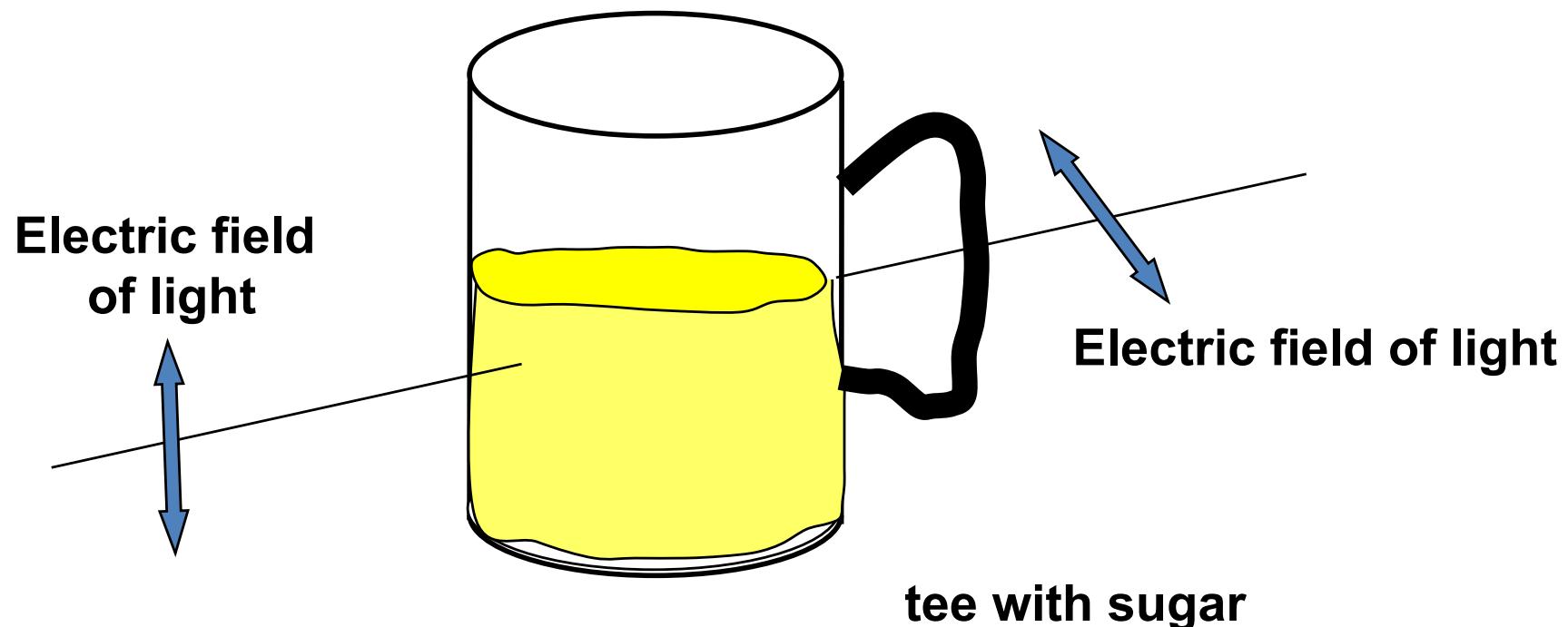
Sugar (D-glucose)



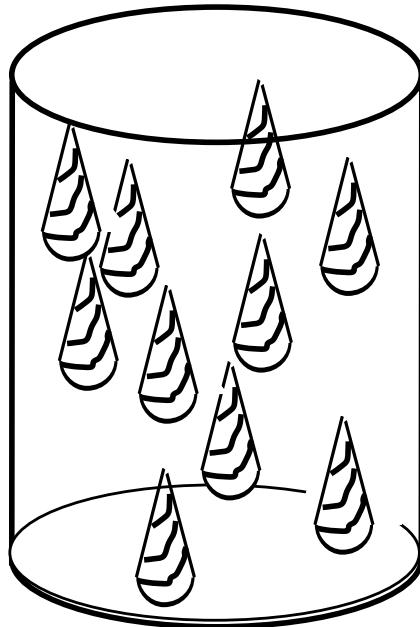
Macroscopic symmetry of non-crystalline materials

$\infty\infty m - \infty\infty$ difference may have important consequences

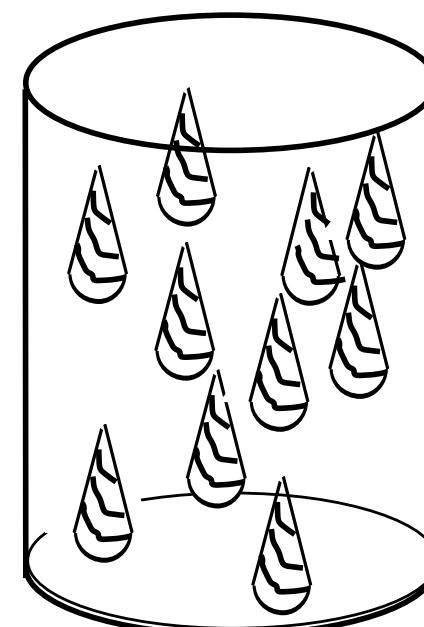
$\infty\infty$ materials are optically active



Macroscopic symmetry of non-crystalline materials

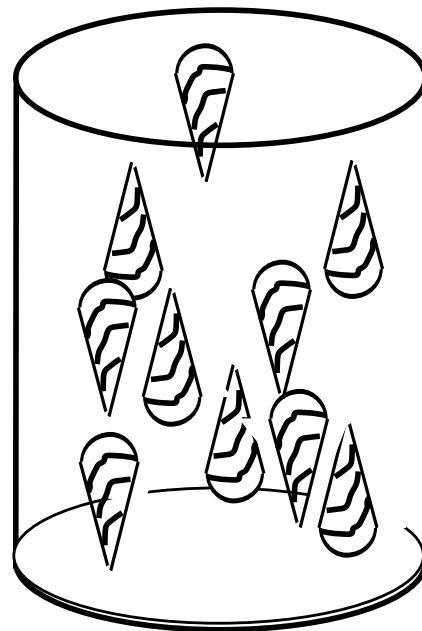
 ∞_R

Right-handed screws, tips up

 ∞_L

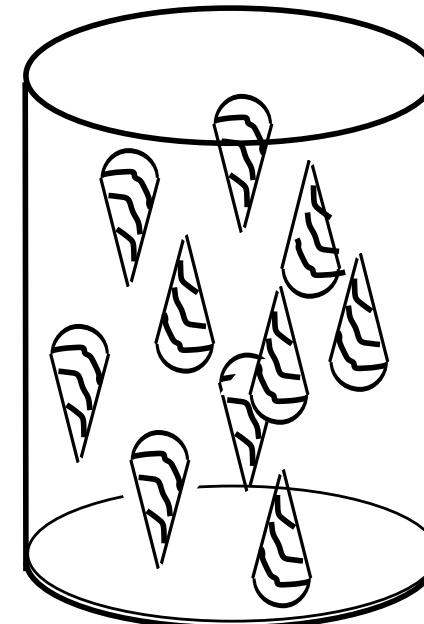
left-handed screws, tips up

Macroscopic symmetry of non-crystalline materials



$\infty 2_R$

**Right-handed screws,
50% tips up+ 50% tips down**



$\infty 2_L$

**Left-handed screws,
50% tips up+ 50% tips down**

Macroscopic symmetry of non-crystalline materials

Crystalline vs. non-crystalline group relation

$$2, 3, 4, 6 \Rightarrow \infty$$

$$222, 32, 422, 622 \Rightarrow \infty 2$$

$$mm2, 3m, 4mm, 6mm \Rightarrow \infty m$$

$$2/m, \bar{6}, 4/m, 6/m \Rightarrow \infty/m$$

$$mmm, 4/mmm, 6/mmm \Rightarrow \infty/mm$$

Essential I

MICroscopic symmetry

230 space groups

MACroscopic symmetry

**32 point
crystallographic
groups**

+

7 Curie groups

-Additional literature for basics of crystallography: “The basics of Crystallography and Diffraction” (Ch. Hammond)

Essential II

 1 (C ₁)			 1-bar (C ₁)			
 2 (C ₂)				 m (C ₅)		 2/m (C _{2h})
				 mm2 (C _{2v})	 222 (D ₂)	 mmm (D _{2h})
 3 (C ₃)			 3-bar (S ₆)	 3m (C _{3v})	 32 (D ₃)	 3-bar m (D _{3d})
 4 (C ₄)	 4-bar (S ₄)	 4-bar 2m (D _{2d})	 4/m (C _{4h})	 4mm (C _{4v})	 422 (D ₄)	 4/mmm (D _{4h})
 6 (C ₆)	 6-bar (C _{3h})	 6-bar 2m (D _{3h})	 6/m (C _{6h})	 6mm (C _{6v})	 622 (D ₆)	 6/mmm (D _{6h})
 23 (T)			 m-bar 3 (T _h)	 43m (T _d)	 432 (O)	 m-bar 3m (O _h)

Centrosymmetric



Non-centrosymmetric Non-polar



Non-centrosymmetric Polar

