

Crystal frame:

$$J_A = -D_A \nabla C_A$$

$$J_B = -D_B \nabla C_B$$

$$D_B = \left(\frac{L_{BB}}{x_B} - \frac{L_{AB}}{x_A} \right) x_B \frac{\partial K_B}{\partial x_B} \Omega$$

Laboratory frame

$$v = -V_m (J_A + J_B) = \frac{J_v}{C_0}$$

velocity in the lab frame.

$$V_m = \frac{1}{C_0} = \frac{V}{N_A + N_B}$$

$$\tilde{J}_A = J_A + v C_A = -\tilde{D} \nabla C_A$$

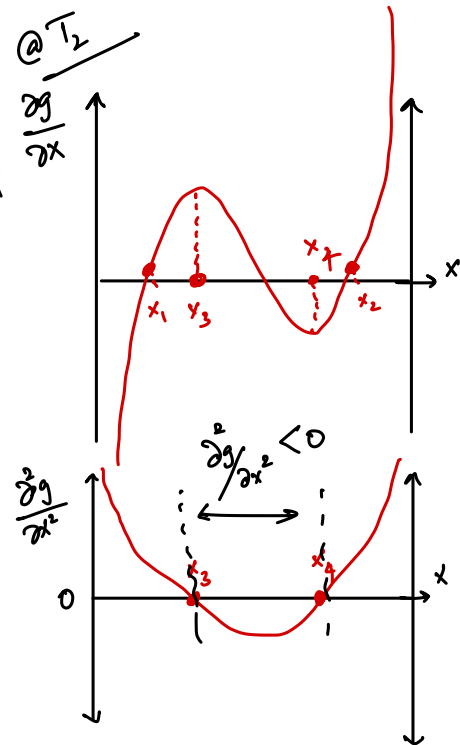
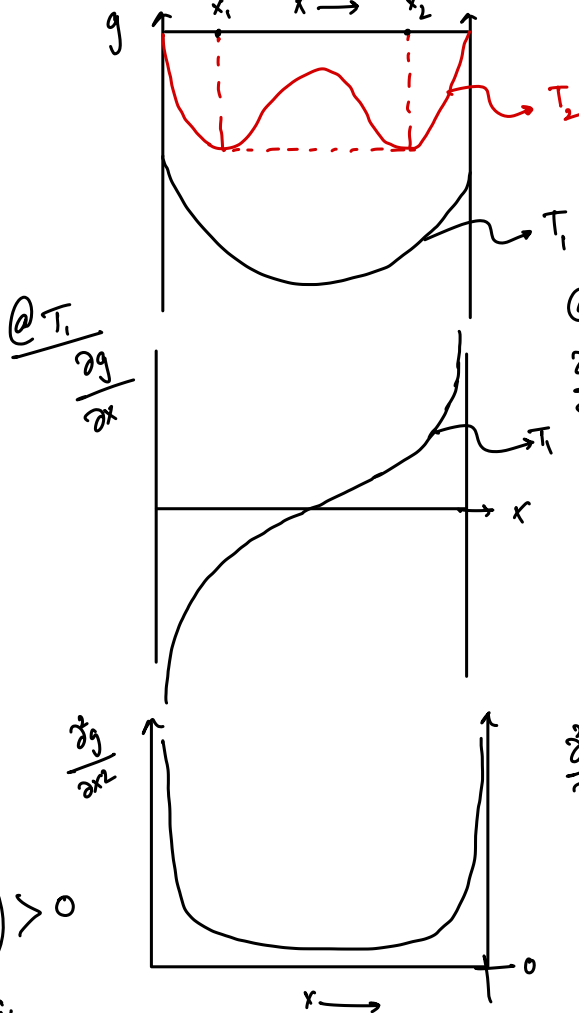
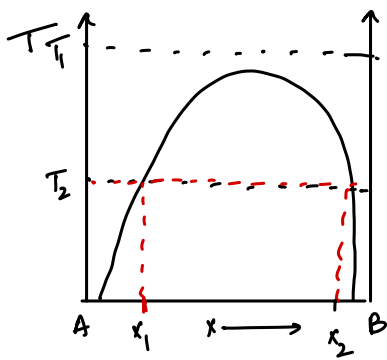
$$\tilde{J}_B = J_B + v C_B = -\tilde{D} \nabla C_B$$

$$\tilde{D} = x_B D_A + x_A D_B$$

$$\tilde{D} = \tilde{L} \Omega \frac{\partial^2 g}{\partial x^2}$$

Similar to the diffusion coefficient in the crystal frame.

SPINODAL DECOMPOSITION: (Thermodynamics is described by the model in Question 2 in PSET III)



$$\frac{\partial^2 g}{\partial x^2} > 0 \Rightarrow \left(\tilde{D} \propto \frac{\partial^2 g}{\partial x^2} \right) > 0$$

\Rightarrow Diffusion flux is down the composition gradient
(i.e.) diffusion is from high concentration to low concentration.

$$\frac{\partial^2 g}{\partial x^2} < 0 \text{ b/w } x_3 \text{ \& } x_4$$

$$J_A = -\tilde{D} \frac{\partial c}{\partial x}$$

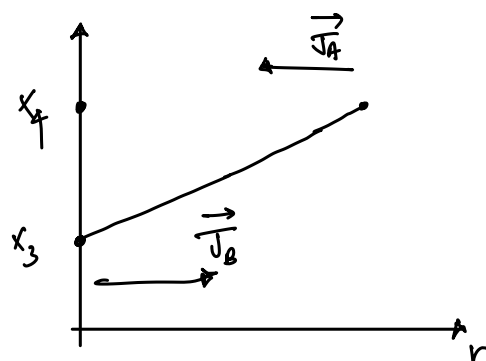
$$\left(\tilde{D} \propto \frac{\partial^2 g}{\partial x^2} \right) < 0$$

\Rightarrow UPHILL DIFFUSION

x_3 & x_4 are often called the SPINODES

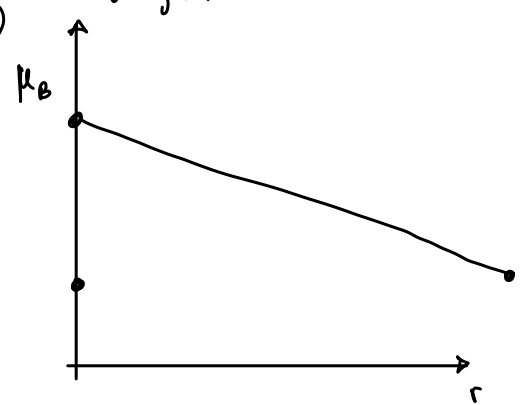
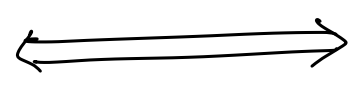
* UPHILL DIFFUSION & SPINODAL DECOMPOSITION:

$$J_A = -\tilde{D} \frac{\partial C}{\partial r} \Rightarrow @ T = T_2 \text{ \& } x_3 < x < x_4 \Rightarrow \tilde{J}_A = |\tilde{D}| \frac{\partial C}{\partial r}$$

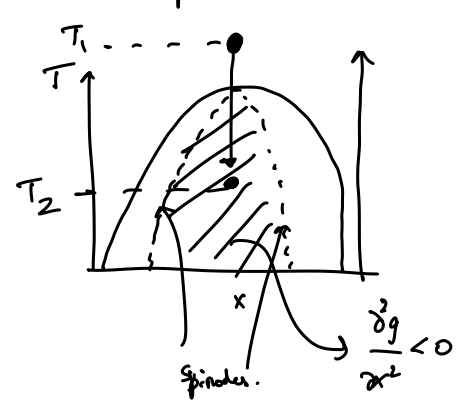
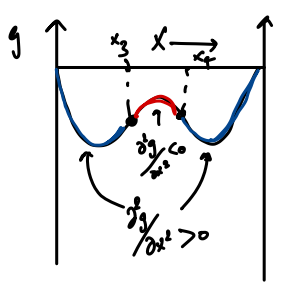


absolute value.

A-rich regions will get richer in A, & B-rich regions will get richer in B
 DOES THIS MAKE SENSE??

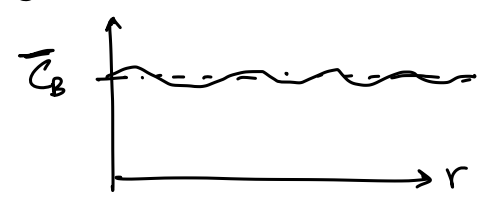


B (and A) are still diffusing down a chemical potential gradient despite moving up a composition gradient.



- What happens to an alloy annealed @ T1 that is then quenched to T2?

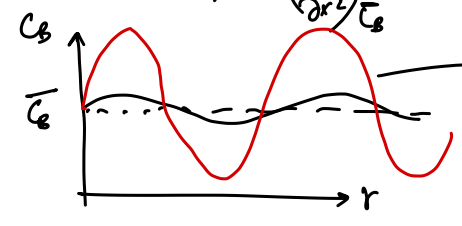
1) @ T1



$$\frac{\partial^2 g}{\partial x^2} > 0 \text{ everywhere} \Rightarrow D > 0$$

there will always be spontaneous fluctuations in concentrations around \bar{C}_B , however they will decay and not grow

2) Quench to T2; $\left(\frac{\partial^2 g}{\partial x^2}\right)_{\bar{C}_B} < 0 \Rightarrow D < 0 \Rightarrow J = |D| \nabla C$



concentration fluctuations are amplified. B diffuses from low C_B to high C_B

* ATTEMPT 1 @ ANALYZING THE MICROSTRUCTURE EVOLUTION DURING SPINODAL DECOMPOSITION:

- Understand the behavior of small fluctuations around an otherwise uniform concentration profile.

- ASSUME:

- ① Infinite domain (sample size)
- ② 1 dimensional model ONLY
- ③ Kinetic parameters independent of alloy concentration

- To predict time evolution:

Ficks II LAW: $\frac{\partial c}{\partial t} = -\nabla J$

$$\frac{\partial c}{\partial t} = -\nabla(-D\nabla c) \quad \xrightarrow{\text{Ficks I LAW}} \quad \boxed{D \text{ is a constant}} \rightarrow \text{ASSUME}$$

$$\boxed{\frac{\partial c}{\partial t} = D\nabla^2 c} \rightarrow \text{Integrate to compute } c(\vec{r}, t)$$

in 1D: $\frac{\partial c}{\partial t} = D \frac{\partial^2 c}{\partial r^2}$

* FOURIER TRANSFORMS TO SOLVE PDE:

$c(\vec{r}, t)$

$$\tilde{c}(\vec{k}, t) = \mathcal{F}\{c(\vec{r}, t)\} = \frac{1}{(2\pi)^{3/2}} \iiint c(\vec{r}, t) \exp(+i\vec{k}\cdot\vec{r}) d\vec{r}$$

Inverse Fourier transform:

$$c(\vec{r}, t) = \frac{1}{(2\pi)^{3/2}} \iiint \tilde{c}(\vec{k}, t) \exp(-i\vec{k}\cdot\vec{r}) d\vec{k} = \mathcal{F}^{-1}\{\tilde{c}(\vec{k}, t)\}$$

In 1D: $k = \frac{2\pi}{\lambda}$; $k \rightarrow 0 \Rightarrow \lambda \rightarrow \infty \Rightarrow$ "long wavelengths"
 $k \rightarrow \infty \Rightarrow$ "short wavelengths"

$$\tilde{c}(k, t) = \mathcal{F}\{c(r, t)\} = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} c(r, t) \exp(ikr) dr$$

$$c(r, t) = \mathcal{F}^{-1}\{\tilde{c}(k, t)\} = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \tilde{c}(k, t) \exp(-ikr) dk$$

* FOURIER TRANSFORM THE DIFFUSION EQUATION:

$$\frac{\partial c}{\partial t} = D \frac{\partial^2 c}{\partial r^2} \rightarrow \mathcal{F}\left\{\frac{\partial c}{\partial t}\right\} = D \mathcal{F}\left\{\frac{\partial^2 c}{\partial r^2}\right\}$$

$$\mathcal{F}\left\{\frac{\partial c}{\partial t}\right\} = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \frac{\partial c}{\partial t} \exp(ikr) dr = \frac{\partial}{\partial t} \left[\frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} c(r, t) \exp(ikr) dr \right] = \frac{\partial \tilde{c}(k, t)}{\partial t}$$

$$\mathcal{F}\left\{\frac{\partial^2 c}{\partial r^2}\right\} = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \frac{\partial^2 c}{\partial r^2} \exp(ikr) dr$$

\uparrow
 Integrate by parts
 $= -k^2 \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} c(r, t) \exp(ikr) dr$

In general: $\mathcal{F}\left\{\frac{d^n c}{dr^n}\right\} = (-ik)^n \mathcal{F}\{c(r,t)\} = (-ik)^n \tilde{c}(k,t)$

Diffusion equation: $\frac{\partial c}{\partial t} = D \frac{\partial^2 c}{\partial r^2} \rightsquigarrow \boxed{\frac{\partial \tilde{c}(k,t)}{\partial t} = -Dk^2 \tilde{c}(k,t)}$ ODE!
 $\tilde{c}(k,t) = \tilde{c}(k,0) \exp(-Dk^2 t)$

— Lets make a few variable transforms:

Instead of solving for $c(r,t) \rightarrow$ we will solve for $\Delta c(r,t) = c(r,t) - \bar{c}$ ↙ average concentration.

$$\underbrace{\frac{\partial \Delta c}{\partial t}} = D \underbrace{\frac{\partial^2 \Delta c}{\partial r^2}} \Rightarrow \frac{\partial (c - \bar{c})}{\partial t} = D \frac{\partial^2 (c - \bar{c})}{\partial r^2} \Rightarrow \frac{\partial c}{\partial t} = D \frac{\partial^2 c}{\partial r^2}$$

These are equivalent:

$$\mathcal{F}\left\{\frac{\partial \Delta c}{\partial t}\right\} = \mathcal{F}\left\{D \frac{\partial^2 \Delta c}{\partial r^2}\right\} \Rightarrow \boxed{\frac{\partial \Delta \tilde{c}}{\partial t} = -k^2 \Delta \tilde{c}(k,t)}$$

Say the initial composition profile is $\Delta c_0(r,0)$

$$\Delta \tilde{c}(k,0) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \Delta c_0(r,0) \exp(ikr) dr$$

Solution to the eqn:

$$\boxed{\Delta \tilde{c}(k,t) = \Delta \tilde{c}_0(k,0) \exp(-k^2 D t)}$$

To evaluate $\Delta c(r,t) \rightarrow$ Inverse Fourier transform.

$$\Delta c(r,t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \Delta \tilde{c}(k,t) \exp(-ikr) dk =$$

$$\boxed{\Delta c(r,t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \Delta \tilde{c}_0(k,0) \exp(-k^2 D t) \exp(-ikr) dk}$$

k is integrated out!

time dependence position dependence.

* How Do WE INTERPRET THE SOLUTION?

Define $A(k) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \Delta C_0(r, 0) \cos(kr) dr$ $B(k) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \Delta C_0(r, 0) \sin(kr) dr$

$\Delta \tilde{C}_0(k, 0) = A(k) + iB(k)$

$$\Delta C(r, t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \exp(-k^2 Dt) \exp(-i kr) \overbrace{\Delta \tilde{C}_0(k, 0)}^{A(k) + iB(k)} dk$$

$\left[\cos(kr) - i \sin(kr) \right]$
Euler's formula

$$\Delta C(r, t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \exp(-k^2 Dt) \left[A(k) \cos(kr) + B(k) \sin(kr) + i(B(k) \cos(kr) - A(k) \sin(kr)) \right] dk$$

$$\Delta C(r, t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \exp(-k^2 Dt) \left[A(k) \cos(kr) + B(k) \sin(kr) \right] dk$$

$+ i \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \exp(-k^2 Dt) \left[B(k) \cos(kr) - A(k) \sin(kr) \right] dk$

odd function of $k \Rightarrow$ integral is zero.

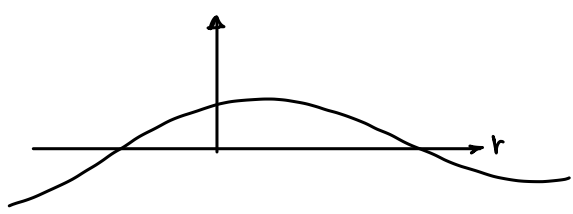
$$\Delta C(r, t) = \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{\infty} \exp(-k^2 Dt) \left[A(k) \cos(kr) + B(k) \sin(kr) \right] dk$$

time Superposition of sines and cosines. each having wavelength $\lambda = 2\pi/k$

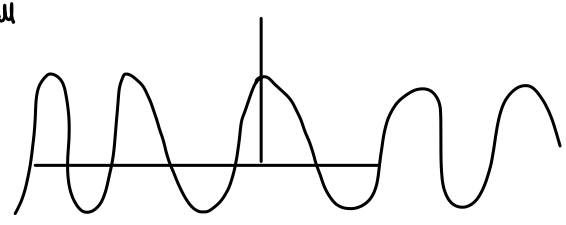
— each modulation of wavelength λ has a DECAY (OR AMPLIFICATION) factor of $\boxed{\exp(-k^2 Dt)}$

— Behavior of different wavelengths:

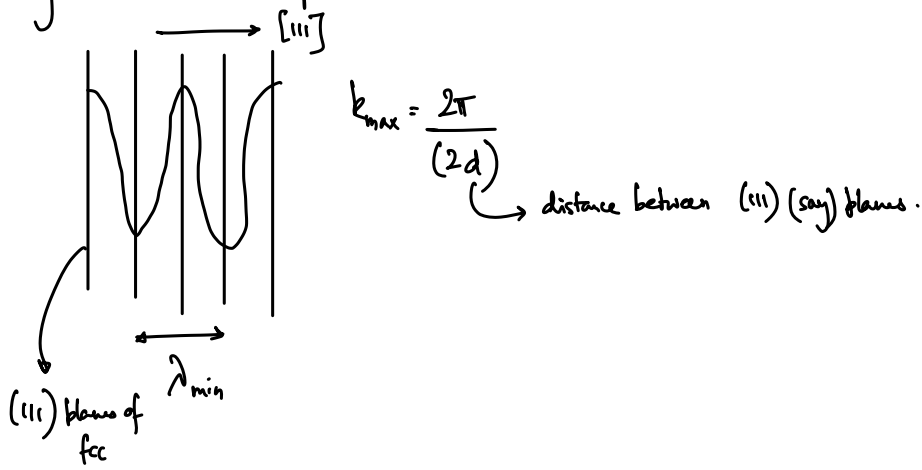
① $k = \text{small} \quad \lambda = \frac{2\pi}{k} \rightarrow \text{large}$



② $k = \text{large} \quad \lambda = \frac{2\pi}{k} \rightarrow \text{small}$



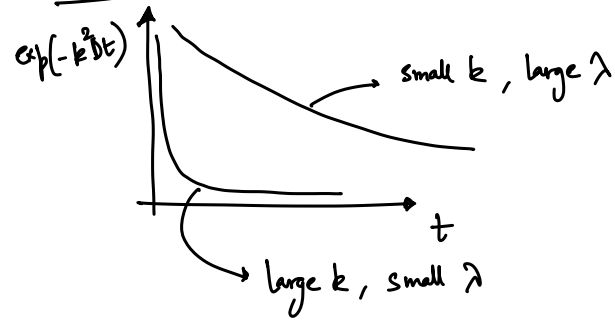
— Physically fluctuations in composition smaller than the distance between atoms would not make sense.



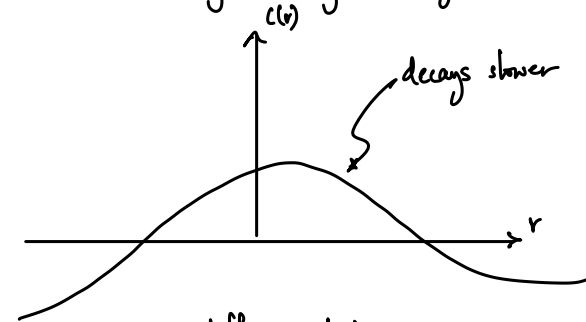
— What is the decay (or amplification) rate for different waves?

$$\exp(-k^2 Dt) = \exp\left(-\frac{4\pi^2}{\lambda^2} Dt\right)$$

CASE I: $D > 0$

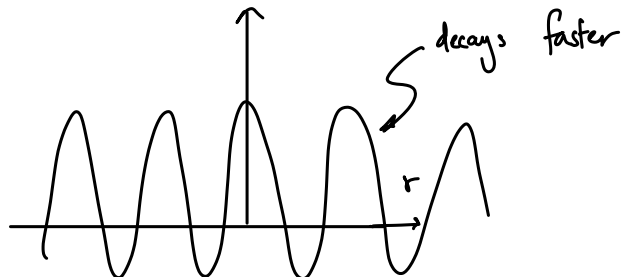


→ time decay of long wavelengths is slower than short wavelengths.



diffusion distances to even out composition fluctuations are longer

⇒ Takes more time to redistribute matter



shorter diffusion distances to even out composition fluctuations.

CASE II: $D < 0$

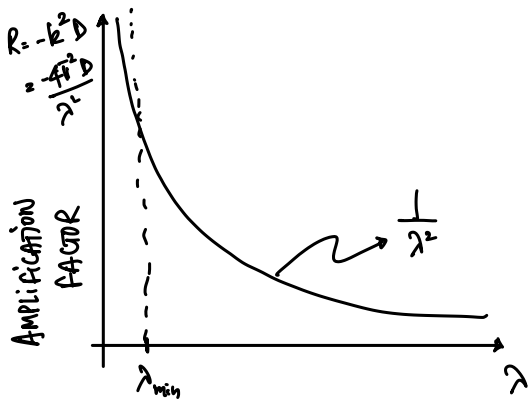
$$\exp(-k^2 D t) = \exp(k^2 |D| t) = \exp\left(\frac{|D| 4\pi^2 t}{\lambda^2}\right) \Rightarrow \text{Concentration fluctuations get amplified.}$$

$k \rightarrow \text{large} \Rightarrow \lambda \rightarrow \text{small} \Rightarrow \text{amplification factor is large}$

$k \rightarrow \text{small} \Rightarrow \lambda \rightarrow \text{large} \Rightarrow \text{amplification factor is small.}$

- time amplification of short wavelengths is much faster than that of long wavelengths

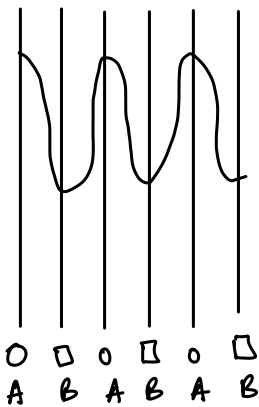
\Rightarrow Atoms want to phase separate !!



Asymptotic Microstructure: in the long time limit

λ with the largest amplification factor will dominate

$$\Rightarrow \Delta c(r, t) \approx \underset{\text{approximate}}{\uparrow} \left[A(k_{\max}) \cos(k_{\max} r) + B(k_{\max}) \sin(k_{\max} r) \right] \exp(-k_{\max}^2 D t)$$



\Rightarrow A & B separate and occupy alternate atomic planes.

..... DO WE HAVE A PROBLEM?

PROBLEMS/ISSUES:

- ① We expected phase separation, but we got a particular ordering of A & B atoms \Rightarrow This is @ odds with the eq^m phase diagram.
- ② Experimentally spinodal decomposition is found to have a "characteristic" length scale.

WHAT WENT WRONG:

- We (implicitly) assumed that we could assign local thermodynamic properties to depend on only local concentration:

$$(k) @ \vec{r} \Rightarrow g(x(\vec{r})), \mu(x(\vec{r}))$$

within this approximation:

$$J = -L \nabla \mu = -D \nabla c = -L \Omega \frac{\partial^2 g}{\partial x^2} \nabla c$$

- This approximation is OK as long as gradients in concentration are not too large.
- Should be applicable OUTSIDE the spinodal.
- Inside the spinodal: \rightarrow large gradients.
- Our model does not penalize interfaces! \Rightarrow We end up forming atomically thin layers.

SOLUTION:

The local free energy depends on the concentration and higher order derivatives of $x(\vec{r})$.

$$(k) g\left(x(\vec{r}), \frac{dx(\vec{r})}{dr}, \frac{d^2 x(\vec{r})}{dr^2}, \dots\right)$$

In 3-dimensions:

$$g\left(x(\vec{r}), \frac{\partial x}{\partial r_\alpha}, \frac{\partial^2 x}{\partial r_\alpha \partial r_\beta}, \dots\right)$$

\swarrow 3 of these \searrow 6 of these

Total free energy:

$$G = \int_V g\left(x(\vec{r}), \frac{\partial x}{\partial r_\alpha}, \frac{\partial^2 x}{\partial r_\alpha \partial r_\beta}, \dots\right) \frac{d\vec{r}}{\Omega}$$

$$g\left(x(\vec{r}), \frac{\partial x}{\partial r_\alpha}, \frac{\partial^2 x}{\partial r_\alpha \partial r_\beta}, \dots\right) = g(x(\vec{r})) + \vec{A}(x(\vec{r})) \cdot \vec{\nabla} x + \frac{1}{2} \sum_{\alpha} \sum_{\beta} m_{\alpha\beta}(x(\vec{r})) \frac{\partial x}{\partial r_\alpha} \frac{\partial x}{\partial r_\beta} + \dots$$

\uparrow function of composition.

$$+ \sum_{\alpha} \sum_{\beta} n_{\alpha\beta} \frac{\partial^2 x}{\partial r_\alpha \partial r_\beta} + \dots$$

Lets simplify the Taylor expansion in 1D:

Assume:

$$g\left(x, \frac{dx}{dr}\right) = g(x, 0) + \underbrace{\left. \frac{\partial g}{\partial \left(\frac{dx}{dr}\right)} \right|_{(x,0)}}_{H(x)} \cdot \left(\frac{dx}{dr}\right) + \frac{1}{2} \underbrace{\left. \frac{\partial^2 g}{\partial \left(\frac{dx}{dr}\right)^2} \right|_{(x,0)}}_{K(x)} \cdot \left(\frac{dx}{dr}\right)^2 + \dots$$

\uparrow
 Only a function of x and gradient of $x \rightarrow$ not $d^2x/dr^2, \dots$