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Ground-state oxygen holes and the metal-insulator transition in the negative charge-transfer rare-earth nickelates

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The metal-insulator transition and the intriguing physical properties of rare-earth perovskite nickelates have attracted considerable attention in recent years. Nonetheless, a complete understanding of these materials remains elusive. Here we combine X-ray absorption and resonant inelastic X-ray scattering (RIXS) spectroscopies to resolve important aspects of the complex electronic structure of rare-earth nickelates, taking NdNiO₃ thin film as representative example. The unusual coexistence of bound and continuum excitations observed in the RIXS spectra provides strong evidence for abundant oxygen holes in the ground state of these materials. Using cluster calculations and Anderson impurity model interpretation, we show that distinct spectral signatures arise from a Ni $3d^8$ configuration along with holes in the oxygen 2p valence band, confirming suggestions that these materials do not obey a conventional positive charge-transfer picture, but instead exhibit a negative charge-transfer energy in line with recent models interpreting the metal-insulator transition in terms of bond disproportionation.

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he intriguing perovskite nickelates family ReNiO3 (with $Re = rare-earth)^{1-4}$ have garnered significant research interest in recent years, due to the remarkable properties they exhibit. These include a sharp metal to insulator transition (MIT) tunable with the Re radius², unusual magnetic order⁵ and the suggestion of charge order⁶ in the insulating phase. While ReNiO₃ single crystals are very hard to synthesize, causing earlier experiments to be mainly restricted to powder samples, extremely high-quality epitaxial thin films can be now produced. As an additional asset, ReNiO3 in thin film form can exhibit even richer properties compared to bulk, for example, tunability of the metal-insulator transition by strain^{7,8}, thickness⁹⁻¹¹ or even by ultrafast optical excitation of the substrate lattice degree of freedom¹². In addition, there has been a lot of interest in nickelate-based heterostructures, motivated on one hand by theoretical predictions of possible superconductivity¹³, and on the other hand by recent observations of exchange bias effects 14 and modulation of the orbital occupation due to strain and interface effects15-18.

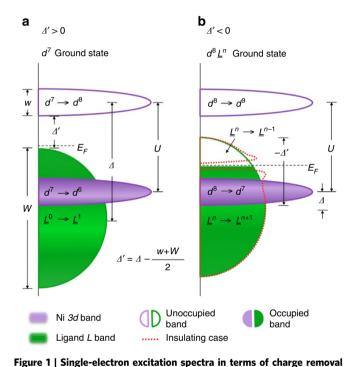
The origin of the rich physics behind these unique properties is complicated by the usual electron correlation problem of transition metal oxides¹⁹. As a consequence, a full understanding of the mechanism driving the MIT in ReNiO₃ remains elusive still today. Hampering the discovery of a universally accepted description of the MIT is the more fundamental problem of understanding the ReNiO₃ electronic structure and the corresponding Ni 3d orbital occupation.

In Fig. 1, we illustrate in a schematic representation of the single-electron excitation spectra how different electronic configurations can result from the two possible regimes of the effective charge-transfer energy Δ' (called for simplicity charge-transfer energy throughout the text). Using formal valence rules, it is expected that the Ni atoms exhibit a $3d^7$ (Ni³⁺) character, likely in a low spin (S = 1/2) configuration. This ground state (GS) is obtained in Fig. 1a for $\Delta' > 0$. However, high-valence Ni³⁺ systems are rare, and although many studies indeed view ReNiO3 as conventional positive charge-transfer compounds yet with the addition of a strong Ni-O covalency and consequently a GS configuration of the type of $\alpha \cdot |3d^7\rangle + \beta \cdot |3d^8\underline{L}\rangle$ (where \underline{L} is an O 2p hole)^{1,20–23}, mounting evidence suggests that the ground state disobeys conventional rules. Alternatively, a negative charge-transfer situation^{24,25}, where a finite density n of holes \underline{L}^n is self-doped into the O 2p band and Ni takes on a $3d^8$ configuration (that is, Ni $3d^8\underline{L}^n$), is recently receiving an increasing interest. This scenario is represented in Fig. 1b for $\Delta' < 0$. Notably, the negative charge-transfer picture is at the base of recent charge or bond disproportionation model theories where, as first suggested by Mizokawa²⁶, the disproportioned insulating state is characterized by alternating Ni $3d^8$ (n=0) and Ni $3d^8\underline{L}^2$ (n=2) sites arranged in a lattice with a breathing-mode distortion^{27–30}. These models distinctively differ from the more traditional charge-disproportionation ones where the Ni $3d^7$ lattice moves towards an alternation of Ni $3d^6$ and $3d^8$ sites in the insulating phase^{6,31–34}.

To get a full understanding of the MIT and of the unique physical properties of the rare-earth nickelates, it is crucial to investigate the ground-state electronic structure in this class of materials. To this purpose, we stress that while both positive and negative charge-transfer interpretations introduced above can be described as highly covalent, there are striking inherent differences between the two. For the former 16,20,21,32,34 , the GS $\alpha \cdot |3d^7\rangle + \beta \cdot |3d^8\underline{L}\rangle$ can be modelled as a Ni $3d^7$ impurity hybridizing with a full O 2p band. Here the primary low-energy charge fluctuations that couple to the GS in first order are the ones from the O 2p to the Ni $3d^7$ impurity, mixing in some Ni $3d^8L$ and higher-order character into the wavefunction. However,

for the negative charge-transfer case^{19,25–30}, all Ni sites assume a $3d^8$ state with on average n=2 holes in the six oxygens coordinating a central Ni ion. This case is more aptly modelled by a Ni $3d^8$ impurity hybridizing with a partially filled O 2p band. The electronic structure and consequently the character of the gap are vastly different in the two scenarios: $\alpha \cdot |3d^7\rangle + \beta \cdot |3d^8\underline{L}\rangle$ with a O 2p-Ni 3d-like gap or $3d^8\underline{L}^n$ with a O 2p-O 2p-like gap (refer to Fig. 1a,b, respectively).

Here we combine two X-ray spectroscopies, namely X-ray absorption (XAS) and resonant inelastic X-ray scattering (RIXS) at the Ni L₃-edge, to resolve if the electronic structure of ReNiO₃ follows a positive or negative charge-transfer picture. While the XAS results are similar to those previously reported, the first ever measurements of high-resolution Ni L₃ RIXS on NdNiO₃ provide crucial insights into the nature of the excitations present.



and charge addition. This sketch introduces conventional parameters used to describe charge dynamics in transition metal oxide (TMO) compounds with formal $3d^7$ filling: (1) charge-transfer energy Δ : energy cost for transferring an electron/hole from the L band to the Ni 3d band (with respect to the band centre of mass); (2) Hubbard U: energy cost needed to remove an electron from the occupied 3d band and to add it to the unoccupied 3d band; (3) effective charge-transfer energy Δ' : key parameter to distinguish between the two different regimes discussed here. Δ' is defined by the equation in the figure, starting from Δ . This figure identifies the ground state (GS) and the gap character obtained for two Δ' regimes: (a) $\Delta' > 0$. In conventional positive charge-transfer compounds the lowest energy removal states are ligand based, and the lowest energy addition states are transition metal based, leading to a charge-transfer derived energy gap (O 2p-Ni 3d like) and a 3d⁷ GS. (**b**) Δ' < 0. In negative charge-transfer compounds, one hole per Ni is doped into the ligand band, giving a density of ligand holes n, L^n . The GS is here Ni $3d^8L^n$. The red-dashed contour bands are a cartoon-like demonstration of the opening of the gap in the mainly O 2p continuum resulting in the metal to insulator transition. In this case, the lowest energy removal and addition states are both ligand based, leading to an O 2p-O 2p like gap. Under no circumstances can this type of gap result from configuration a, unless active doping is considered. Note that this figure neglects Ni-O hybridization, to provide clear distinction between the regimes.

The unusual coexistence of bound and continuum contributions across the narrow Ni L₃ resonance and the specific nature of the orbital excitations, allows us to verify that the electronic ground state contains abundant O 2p holes and that the Ni sites are indeed best described as Ni $3d^8\underline{L}^n$, rather than a low spin Ni $3d^7$, showing that the ReNiO₃ are indeed self-doped, negative charge-transfer materials. Further, the RIXS spectra exhibit a clear suppression of the low-energy electron–hole pair continuum in the insulating phase, providing not only a fingerprint of the opening of the insulating gap at $T < T_{\rm MI}$ but also experimental evidence of the dominant O 2p-character for the states across $E_{\rm F}$, as expected for a negative charge-transfer system.

Results

Bulk-like NdNiO3 thin film. Several high-quality NdNiO3 thin films grown on a variety of substrates were investigated. Epitaxial films were prepared by off-axis radiofrequency magnetron sputtering 8,9,35,36 and were fully characterized by X-ray diffraction measurements, atomic force microscopy, transport and soft X-ray scattering measurements. In the following, we will focus on 30 nm thick NdNiO3 film grown on (110)-oriented NdGaO3 substrate under tensile strain conditions (+1.6% of strain) as a representative example of bulk ReNiO3 in general. In this case, coupled metal–insulator and paramagnetic-to-antiferromagnetic transitions have been found at $T \sim 150 \, \text{K}$, consistent with the corresponding bulk compound 1.

Bound and continuum excitations across Ni L₃ resonance. XAS and RIXS measurements were carried out by exciting at the Ni L₃ edge, corresponding to the $2p_{3/2}$ to 3d electronic transition at around ~ 852 eV. The XAS spectra have been acquired in the partial fluorescence yield mode, by integrating the RIXS spectra for each incident photon energy $hv_{\rm in}$ to insure the bulk sensitivity.

Figure 2a–c presents an overview of XAS and RIXS data for the 30 nm thick $NdNiO_3$ film, measured at both 300 K (metallic phase, red colour) and at 15 K (insulating phase, blue colour). The Ni $\rm L_{2,3}$ XAS shown in Fig. 2a is in good agreement with the previously published data on $NdNiO_3$ (refs 20,21,37–39). At 15 K, the Ni $\rm L_3$ region of the XAS (from 850 to 860 eV), is characterized by two clear structures—a sharp peak at 852.4 eV (A) and a broader peak at 854.3 eV (B)—both of which are present in other ReNiO_3 as well 1,16,20,24 . At 300 K both peaks are still recognizable, however, their separation is less evident.

A series of high-resolution RIXS spectra have been recorded across the Ni-L3 resonance in steps of 0.1 eV, as shown in the intensity colour maps of Fig. 2b,c. Each spectrum obtained for a specific hvin measures the intensities of the emitted photons as a function of the energy loss $hv = hv_{in}-hv_{out}$, where hv_{out} is the outgoing photon energy. RIXS is able to simultaneously probe excitations of diverse nature, for example, lattice, magnetic, orbital and charge excitations^{40,41}. In addition, one can distinguish with RIXS between localized, bound excitations and delocalized excitations involving continua. For localized electronic excitations, the RIXS signal appears at a fixed hv while scanning hv_{in} across a corresponding resonance (Raman-like behaviour). Conversely, for delocalized electronic excitations involving continua, the RIXS signal has a constant hvout and therefore presents a linearly dispersing energy loss as a function of hvin (fluorescence-like behaviour)^{40–43}.

From the RIXS maps of the NdNiO₃ thin film in Fig. 2b,c one directly observes a clear, strong Raman-like response at around 1 eV of energy loss when tuning $h\nu_{\rm in}$ to the XAS peak A. These atomic-like dd-orbital excitations, which are sensitive to the local ligand field symmetry, behave similarly to those observed in other oxide materials like the prototypical Ni $3d^8$ system NiO⁴⁴.

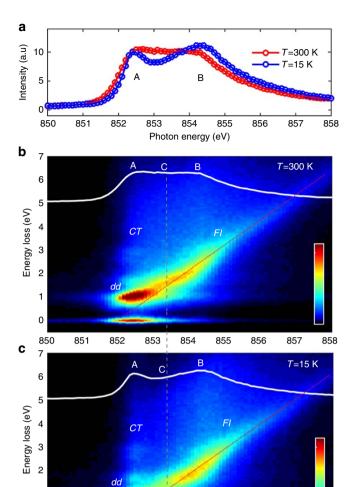


Figure 2 | Overview of XAS and RIXS measurements for a 30 nm NdNiO3 film on NdGdO3. (a) Ni $L_{3,2}$ XAS measured in partial fluorescence yield mode at 300 K (metallic phase) in red and at 15 K (insulating phase) in blue. Refer to Supplementary Note 2 and Supplementary Fig. 2 for XAS in total electron yield mode. (b,c) RIXS intensity map measured across the Ni L_3 edge at 300 K (15 K); intensity scale bar from 0 to 5 (a.u.). The white solid line displays the XAS measured at the same temperature. The letters A, B and C mark the three different incident energies mentioned in the text, while dd, CT and FI refer to RIXS excitations of different character, also discussed in the text. The grey-dashed line indicates the incident energy giving the most pronounced changes in the RIXS map and in the XAS across the MIT. The red-dotted line provides a guide to the eye for the linearly energy dispersing FI feature.

854

Photon energy (eV)

855

856

857

858

850

851

852

853

A fluorescence-like contribution resonates instead at the XAS peak B, contrary to the Raman-like response dominated by multiplet effects observed in NiO at the corresponding Ni L_3 XAS shoulder 44 . Already by looking at the colour map, this fluorescence-like spectral signature is clearly visible all across the Ni L_3 -edge and always with a linearly dispersing behaviour, as suggested by the reddotted line overimposed to the data. Interestingly, the fluorescence intensity distribution in the NdNiO $_3$ RIXS map has also a strong temperature dependence, while at 300 K it merges continuously with the dd-excitations (Fig. 2b), at 15 K a dip in intensity is created corresponding to the incident photon energy C, $hv_{in} = 853$ eV (see Fig. 2c, dashed grey line).

To gain more insight into the origin of the observed Ramanand fluorescence-like spectral response, we closely examine the individual RIXS spectra and we perform a fitting analysis to extract the general behaviour of the main spectral components. As shown in Fig. 3a, the raw RIXS spectrum is decomposed into three different contributions (see Supplementary Note 1 and Supplementary Fig. 1). Referring to the photon energy $hv_{\rm in} = A$, we identify from the corresponding RIXS spectrum localized dd-orbital excitations extending from 1 to 3 eV (black line), a broad background centred around 4 eV (CT, green line), and a residual spectral weight (Fl, magenta line) peaking at 0.7 eV between the elastic line and the dominating dd-profile.

Remarkably, at this photon energy even the fine multiplet structure of the dd-excitations is in good agreement with that of NiO⁴⁴, suggesting immediately that NdNiO₃ has an unusual Ni $3d^8$ -like S=1 local electronic structure similar to NiO. Figure 4

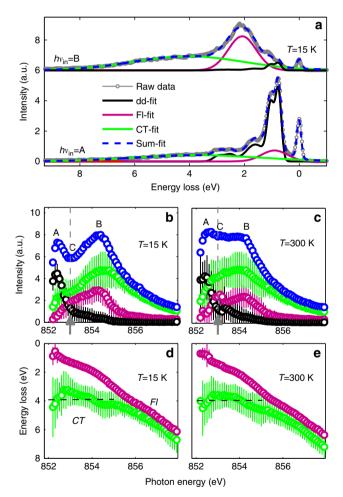


Figure 3 | RIXS data analysis across the MIT. (a) RIXS line spectra (grey open-dot line) measured at $hv_{\rm in}={\sf A}$ and $hv_{\rm in}={\sf B}$ at 15 K. The thin solid lines refer to Gaussian fits of dd-excitations (black), the CT-excitation (green) and the FI excitation (magenta). The blue-dashed line refers to the sum of the three fitting contributions, plus the elastic line at 0 eV. (**b,c**) Integrated intensity of the fit dd-, CT- and FI-excitations together with the total integrated intensity at 15 K (300 K). The grey arrow marks the incident energy giving the most pronounced changes in the RIXS integrated intensity across the MIT. (**d,e**) Peak energy dispersion for the CT- and FI-excitation at 15 K (300 K). The same colour code as in **a** is used throughout the figure. The error bars of the model parameters are evaluated using the least square fitting routine and expressed in s.e.d. The error bars of the initial RIXS spectra were estimated assuming Poisson statistics.

furthermore endorses this concept, displaying the comparison between the RIXS data and cluster model calculations performed for a Ni $3d^7$ GS (grey line, model parameters taken from ref. 39) and a crystal field calculation of the dd-excitations for a Ni 3d8 GS (black line, model parameters from NiO⁴⁵; see also Supplementary Note 3 and Supplementary Table 1). The dd-excitations obtained for the Ni $3d^7$ case strongly differ from the present NdNiO₃ data not only in the peak energy positions displaced to higher energies, but also in the intensity distribution profile, thus ruling out a Ni $3d^7$ GS scenario. The dd-excitations calculated for the NiO in Ni 3d⁸ configuration instead present a remarkably good correspondence with the present data, verifying the d^8 -like character of Ni in this compound: such a finding is the first key result of our study and directly poses the question if NdNiO₃ deviates from a conventional positive charge-transfer picture based on a Ni 3d⁷ GS in favour of a negative chargetransfer scenario based on a Ni $3d^8$ GS, as illustrated in Fig. 1b. Finally, we note that in the insulating phase an extra dd-peak emerges from the experimental data at $\sim 0.75 \, \text{eV}$, which is not captured by the Ni $3d^8$ calculation. We speculate that this contribution could be caused by symmetry-breaking phenomena (related to the presence of different Ni sites in the insulating phase). However, more advanced and detailed calculations have to be developed to reproduce this finding.

Referring now to $hv_{\rm in} = B$ incident photon energy, the three contributions identified above for $hv_{\rm in} = A - dd$, CT and FI—can be still distinguished in the corresponding RIXS spectrum. However, comparing the two spectra in Fig. 3a, we observe a shift in energy for the CT- and FI-contributions, contrary to the dd-excitations which are fixed in energy loss, and a strong redistribution of spectral weight between the three spectral components.

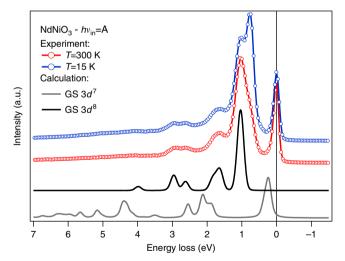


Figure 4 | Comparison between experimental and calculated *dd***excitations.** Data: RIXS spectra at $T=15\,\mathrm{K}$ (blue circles) and at $T=300\,\mathrm{K}$ (red circles), $hv_{\mathrm{in}}=A$, σ polarization. Calculation: crystal field RIXS calculation for a Ni $3d^8$ GS based on NiO parameters ⁴⁵ (black line); cluster calculation for a Ni $3d^7$ GS using the parameters in ref. 39 (grey line). Please note that the elastic lines have been removed from the calculated spectra to better display the differences at the low energies. The experimental dd-line shape is well reproduced by the Ni $3d^8$ cluster calculation, supporting the negative charge-transfer scenario for ReNiO₃. The good matching between data and calculation shows that: (1) there is no inversion of crystal field in ReNiO₃ compared with NiO, in agreement with ref. 25 for dominating Coulomb interaction; (2) being the Ni–O distance shorter in ReNiO₃ than in NiO, the negative charge-transfer energy may lead to a reduction of the expected t_{2g} - e_g splitting.

To disentangle localized versus delocalized character of the RIXS excitations, we extend in Fig. 3b-e the fitting analysis to all the RIXS spectra between 852 and 858 eV incident photon energies and we track for each contribution the integrated intensity (Fig. 3b,c) and the peak energy dispersion (Fig. 3d,e; see Supplementary Note 1 and Supplementary Fig. 1). While the ddexcitations (black dots) have a Raman-like behaviour throughout the energy range and clearly resonate at $hv_{in} = A$, the other two contributions (CT in green and Fl in magenta dots) resonate instead at $hv_{in} = B$ (see Fig. 3b, 15 K). Interestingly, and contrary to the similarity in their resonant behaviour, the CT- and the Fl-contributions present different peak energy dispersions as a function of the incident photon energy (Fig. 3d). The broad CT-peak (FWHM \sim 6 eV) displays a behaviour characteristic of well-known charge-transfer excitations between the central Ni site and the surrounding oxygens: constant energy loss ($\sim 4 \,\mathrm{eV}$) up to the resonant energy $hv_{in} = B$, and a fluorescence-like linear dispersion at higher incident photon energies. The Fl-peak (FWHM ~1 eV), instead, linearly disperses versus incident photon energy across the full Ni L3 resonance: as introduced above, this behaviour is a clear fingerprint for a delocalized excitation involving continua. Overall, these findings are common to both low and high temperature data sets. However, in the metallic state (see Fig. 3c,e, 300 K) the intensity of both CT- and Fl-peaks is enhanced at $hv_{in} = C$: the resulting extra weight in the integrated RIXS intensity profile (Fig. 3c, blue dots) mimics the filling of the valley observed in the XAS spectrum at 300 K, and explains the absence of a dip in the intensity distribution of the RIXS map (Fig. 2b) at the same incident energy.

In addition, we examine the Fl-excitations more closely in Fig. 5, where we focus on the low energy loss range ($<1.5\,\mathrm{eV}$) of the high-statistics RIXS spectra obtained at the Ni L₃ pre-peak region, starting 1 eV before peak A. In Fig. 5a, we observe changes in the dd-excitations between 300 and 15 K (across the MIT) likely due in part to local rearrangements of the NiO₆ octahedra. Moreover, a sizeable spectral weight continuum around 0.2–0.5 eV (see Fig. 5a, inside the ellipse area) is present only at 300 K in the metallic phase, as better displayed by the RIXS colour maps for 300 and 15 K (see Fig. 5b,c, respectively).

Anderson impurity model. To understand the origin of the various RIXS excitations and their link to the electronic structure, one can employ an Anderson impurity model (AIM) interpretation (Supplementary Note 4). While the schematics in Fig. 1a,b show the single-particle removal and addition excitations for different charge-transfer scenarios, RIXS actually measures charge neutral excitations, which are well represented in a configuration interaction-based AIM. These charge neutral excitations are shown schematically in Fig. 6a for the negative charge-transfer case $\Delta' < 0$ and the positive charge-transfer case $\Delta' > 0$ (NiO like) of a Ni $3d^8$ impurity. As previously mentioned, for the $\Delta' > 0$ case, the only charge fluctuations possible in the AIM are from the full O 2p band to the Ni impurity level. While conserving the total charge, these fluctuations give rise to a Ni 3d⁹L band corresponding to charge-transfer excitations (shown in green in Fig. 6a for NiO). However, for the Δ' < 0 case (recall Fig. 1b), the presence of a self-doped, partially filled O 2p density of states (DOS) extending across the Fermi level opens additional pathways for the neutral charge fluctuations. Electrons can either hop from the O 2p valence band to the O 2p conduction band leaving the Ni impurity occupation unchanged, yielding a characteristic low energy electron-hole pair continuum of m excitations $d^8\underline{v}^mc^m$ marked in magenta (\underline{v}) is a hole in the valence band and c an electron in the conduction band, as shown in the small O 2p DOS inset of Fig. 6a); or, to and from the Ni impurity level, from the O 2p valence band and toward the O 2p conduction band, respectively, causing the impurity occupation to change by plus or minus one and yielding a charge-transfer like continuum of excitations at higher energy. Eventually these charge-transfer excitations can be dressed by associated electronhole pair excitations, $\underline{v}^m c^m$, resulting in $d^9 \underline{v}^{m+1} c^m$ or $d^7 \underline{v}^m c^{m+1}$ bands of excitations (green band). We stress that the here introduced low-energy electron-hole pair excitations can be obtained only for the $\Delta' < 0$ case, where the O 2p DOS crosses E_E .

The effects of these two impurity models on RIXS are detailed in Fig. 6b. The case of NiO can be solved numerically including the full correlations within the Ni 3d shell, and we show the calculated RIXS map in Fig. 6b. Comparing this to the schematic NiO configurations in Fig. 6a, we see that there are Raman-like dd-excitations below $4\,\mathrm{eV}$, corresponding to reorganized Ni $3d^8$ orbital occupations, and a charge-transfer band at distinctively higher energy losses which is Raman-like for lower incident photon energies up to ca. $855\,\mathrm{eV}$, before dispersing like fluorescence for higher photon energies. However, as the schematic in Fig. 6a suggests, the charge-transfer excitations do not extend down to low energy losses for the NiO case $(\Delta' > 0)$. To gain further insight into the NdNiO $_3$ experimental data, the extracted CT- and Fl-dispersion curves of Fig. 3d are overlaid on the calculated RIXS map in Fig. 6b. Indeed, the NdNiO $_3$

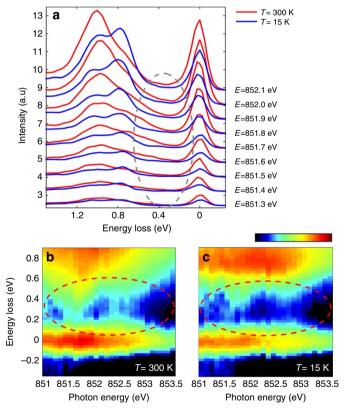


Figure 5 | Low-energy electron-hole pair continuum in the RIXS spectra. (a) RIXS line spectra measured for incident energies going from $hv_{\rm in}=$ A-1 eV up to $hv_{\rm in}=$ A-0.2 eV in steps of 0.1 eV at 15 K (blue) and 300 K (red). Each spectrum has been acquired for 5 min. The grey ellipse highlights the energy loss region where the electron-hole pair continuum is more prominent. (b,c), Magnification of the low energy loss region (<0.9 eV) of the RIXS map with a logarithmic intensity scale at 300 K (15 K); intensity scale bar from -3 to 1 (a.u.). The spectra have been normalized to the dd-area to have comparable background signal in the low energy loss region. The red ellipses underline the electron-hole pair continuum present at 300 K (b) and the intensity gap in the same energy window at 15 K (c).

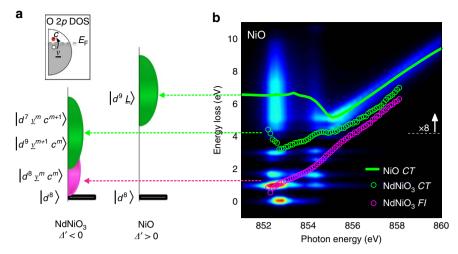


Figure 6 | Anderson impurity model interpretation of the RIXS and XAS spectra. (a) AIM schematic for charge neutral RIXS excitations. The configurations of the AIM are very different for $\Delta' < 0$ (NdNiO₃) and $\Delta' > 0$ (NiO) $3d^8$ compounds. CT-like excitations are obtained in both cases (green bands), while FI-like excitations (magenta band) are found only for $\Delta' < 0$ and correspond to electron-hole pair excitations as shown in the O 2p DOS inset. (b) Calculated RIXS map for the positive charge-transfer compound NiO, using the AIM. The NdNiO₃ CT- and FI-excitations dispersion curves from Fig. 3d are overlaid for comparison (green and magenta open dots, respectively), as well as the NiO CT-excitation dispersion (green solid line). Horizontal coloured lines make the connection between the RIXS excitations and the assigned interpretation in the AIM schematic of a.

CT-excitations (open green dots) show a similar behaviour as the NiO CT ones (solid green line). The NdNiO₃ Fl-excitations, instead, with their distinctive fluorescence-like dispersion differ from any of the NiO excitations. Interestingly, the identified Fl-contribution propagating down to very low energy losses is compatible instead with the electron-hole pair continuum excitations $d^8\underline{\nu}^m c^m$ coming from the broad O 2p band: this finding is the second key result of our study and naturally occurs for the negative charge-transfer case (Δ' < 0) as represented in the AIM schematic of Fig. 6a (magenta band).

Discussion

The main findings of the presented data analysis and interpretation are as follows: localized dd-excitations sharply resonate at the XAS peak A with a lineshape consistent with a Ni $3d^8$ configuration; delocalized Fl-excitations mostly resonate at the XAS peak B and are interpreted as electron–hole pair excitations across the O 2p band cut by E_F (Fig. 1b, green contours); spectral weight reduction of the electron–hole pair excitations close to zero energy loss (Fig. 5b,c) suggests the opening of an O–O insulating gap (Fig. 1b, red-dashed contours) at low temperature, in line with previously reported optical conductivity 46,47 studies and similarly as in more recent ARPES data 48 revealing a spectral weight transfer from near E_F to higher binding energies across MIT.

This collection of results clearly identifies NdNiO₃ as a negative Δ' charge-transfer material, with a local Ni $3d^8$ configuration⁴⁹, a predominant O 2p character across the Fermi level⁵⁰ and a consequent GS of mainly Ni $3d^8\underline{L}^n$. This picture is compatible with the scenario proposed by Mizokawa²⁶, also discussed as bond disproportionation model in recent theoretical approaches^{27–30,51,52}, which comprises an expanded $3d^8$ Ni site (n=0, S=1) and a collapsed $3d^8\underline{L}^2$ Ni site (n=2, S=0) alternating in the insulating phase with the following spin order $\uparrow 0 \downarrow 0$ and a homogeneous Ni $3d^8\underline{L}$ (n=1) GS in the metallic phase. As underlined in previous works^{28,29}, this model is in agreement with several breakthrough experimental findings: (1/2 0 1/2) antiferromagnetic Bragg peak in the insulating phase⁵; charge ordering⁶, which in this model is distributed among both Ni and O

sites instead of only Ni sites; absence of orbital order³⁴; evidence of strong Ni–O covalence in the GS^{4,20,21}.

Furthermore, the different resonant behaviour extracted in this study for localized and delocalized RIXS excitations suggests that the two distinct XAS peaks marked at low temperature mostly result from the two different components of the GS, being XAS peak A mostly associated with a Ni $3d^8$ configuration, and XAS peak B with the delocalized ground-state Ni $3d^8\underline{L}^2$ configuration. This is in line with the energy dependence of the $(1/2\ 0\ 1/2)$ peak resonating at $hv_{\rm in}=A$ (refs 31,37), here assigned to the magnetically active S=1 site.

In conclusion, by combining Ni L₃ XAS and RIXS measurements we studied the electronic ground-state properties of ReNiO₃, to discriminate the electronic structure between a negative and a positive charge-transfer scenario. By analysing the first ever highresolution Ni L3 RIXS data obtained for ReNiO3, we identified the coexistence of bound, localized excitations and strong continuum excitations in both the XAS and the RIXS spectra, in contrast to earlier absorption studies which assumed primarily charge-transfer multiplet effects in the XAS. Further, we disentangled the continuum features in the RIXS spectra into charge-transfer and fluorescence excitations, showing the latter to arise due to the presence of a ground state containing holes in the oxygen 2p band. Electron-hole pair excitations from oxygen 2p states across the Fermi level have been identified down to zero energy loss, mimicking the opening of a gap for $T < T_{MI}$. All these experimental observations provide clear indication of an O 2p hole-rich ground state with Ni $3d^8L^n$ electronic configuration as the main component, as expected for a negative charge-transfer system. This GS configuration lends support to the treatment of the ReNiO₃ as a S=1 Kondo or Anderson lattice problem with a Ni $3d^{8}L^{n}$ (n = 1) metallic GS, and realizing the MIT by a bond disproportionation leading to two Ni site environments: Ni 3d8 (n=0, S=1) and Ni $3d^8L^2$ (n=2, S=0), differing in the hybridization with the O 2p hole states yet leaving the charge at the nickel sites almost equal. While this result is vital for the understanding of the rare-earth nickelate family per se, the combined XAS and RIXS approach demonstrated here opens the opportunity to classify the electronic structure for other cases of very small or negative charge-transfer gaps Δ' , which could

be common to other materials with unconventionally high formal oxidation states (as for example sulfides, selenides and tellurides).

Methods

Experimental details. XAS and high-resolution RIXS measurements have been performed at the ADRESS beamline of the Swiss Light Source 53 , Paul Scherrer Institute. The sample has been oriented in grazing incidence, with the incoming photon beam impinging at 15° with respect to the sample surface. The scattering plane for the RIXS measurements was coinciding with the crystallographic ac-plane (or bc-plane, equivalently). All the data displayed here have been measured for incoming photons polarized parallel to the a (b) axis (also referred to as σ polarization, perpendicular to the scattering plane). For the RIXS measurements we used the SAXES spectrometer 54 prepared with a scattering angle of 130° and a total energy resolution of 110 meV. The spectrometer was set in the high-efficiency configuration, using the 1,500 lines per mm VLS spherical grating. This set-up allowed acquiring around 600–800 photons at the maxima of the prominent spectral structures already in 1 min. The recorded scattered photons were not filtered by the outgoing polarization.

Data availability. The XAS and RIXS data that support the findings of this study are available from the corresponding authors V.B. and T.S. upon request.

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Author contributions

V.B., S.C., M.G., R.S., G.S., J.-M.T. and T.S. planned the project. S.C., M.G., R.S., P.Z. and J.-M.T. grew and characterized the thin film samples. V.B., S.C., R.S., Y.H., V.S. and T.S. carried out the experiment. V.B. and T.S. analysed the data. R.G., S.B. and G.S. carried out the theoretical calculations. V.B., R.G., G.S. and T.S. wrote the paper with contributions from all authors.

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