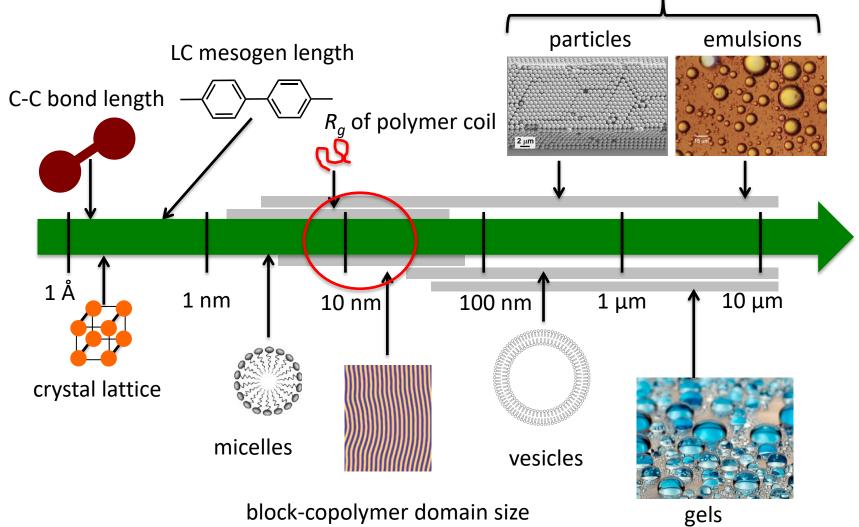
EPFL

Polymers



colloids



Challenges

14.12.2022: Groups 6, 7, 8, 9, 10, 12

21.12.2022: Groups 14, 16, 20, 23, 25

Presentation:

Maximum 10 min + 5 min questions, each team member must present a part

Note that there is NO report to be handed in.

Course outline



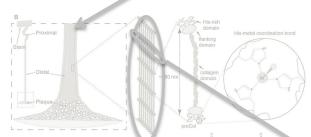
Introduction





Ordered materials

Thermotropic liquid crystals



E. Degtyar, et al., *Angew. Chem. Int. Ed.*, 2014, **53**, 12026-12044

Lyotropic liquid crystals

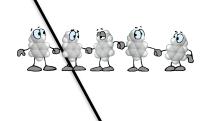
Cell Membrane



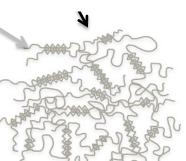


Disordered materials

Polymers







Colloids

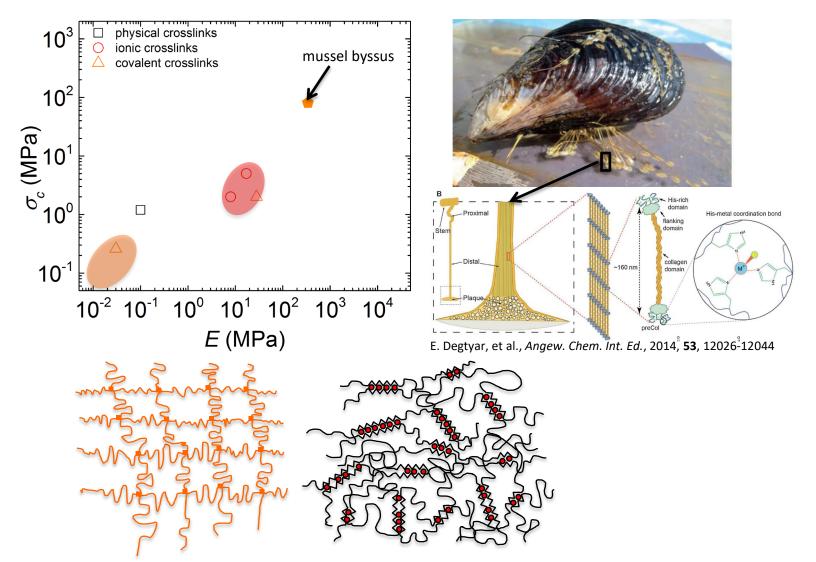
Nanoparticles



Emulsions



Tough and strong soft materials



Outline

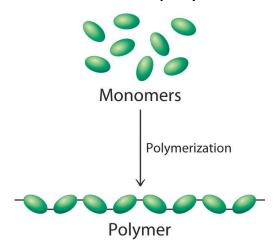
- Introduction
 - Isomerism
 - Tacticity
- Influence of polymer structure on crystallinity
- Polymer solutions
- Polymer melts
- Polymers
 - Thermoplasts
 - Thermosets
 - Elastomers
- Recycling of polymers
- Characterization of polymers
- Application

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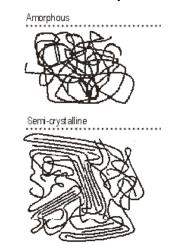
Polymers

How to make polymers



- Polymer chemistry and macromolecular engineering: Harm-Anton Klok
- 2. Organic electronic materials: Holger Frauenrath

How do they arrange?



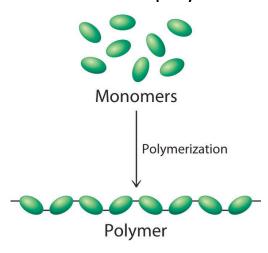
Physical chemistry of polymeric materials: Eva Klok-Lermann

How can we make gels?

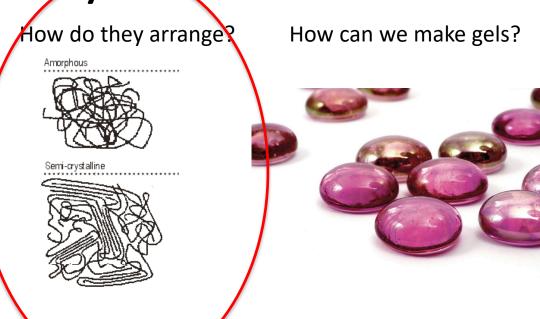


Polymers

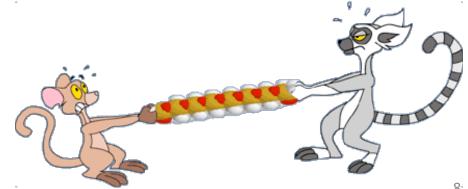
How to make polymers



- Polymer chemistry and macromolecular engineering: Harm-Anton Klok
- 2. Organic electronic materials: Holger Frauenrath



How does the structure influence their properties?



Isomerism

Isomers are molecules that have the same chemical formula but different structures.

Example 1: C₃H₇OH

isopropanol

$$T_m = -89^{\circ}\text{C}$$

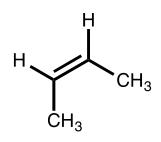
 $T_b = 83^{\circ}\text{C}$
 η at 25°C = 1.96 mPas

$$H_3C$$
 OH

1-propanol

$$T_m = -126$$
°C
 $T_b = 97$ °C
 η at 25°C = 1.96 mPas

Example 2: C₄H₈



cis-2-butene

$$T_m = -139$$
°C
 $T_b = 4$ °C

$$H$$
 CH_3
 CH_3

trans-2-butene

$$T_m = -106$$
°C
 $T_b = 4$ °C

Why does isopropanol have a much higher melting temperature than 1-propanol?

- A. Stronger VdW forces
- B. More H-bonds
- C. It is easier to be packed into a crystal structure
- D. All of the above answers

Why are the boiling points of cis- and transbutene the same?

- A. VdW forces between molecules are equally strong.
- B. They are equally easily packed into a crystal structure.
- C. They have equal vapor pressures.

cis-2-butene trans-2-butene
$$H \xrightarrow{CH_3} H \xrightarrow{CH_3} H$$

Isomerism

Isomers are molecules that have the same chemical formula but different structures.

Example 1: C₃H₇OH

isopropanol

$$T_m = -89^{\circ}\text{C}$$

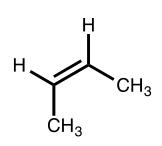
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°C
 $T_b = 4$ °C

trans-2-butene

$$T_m = -106$$
°C
 $T_b = 4$ °C

Tacticity

Example: poly(propylene)

The tacticity influences the ability and propensity of polymers to crystallize.

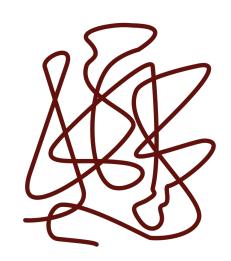
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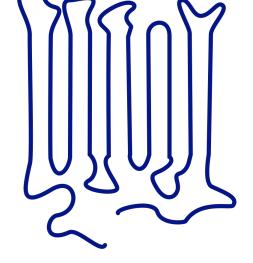
Polymer structure

most polymers are amorphous

Polymers can crystallize if chains are stretched → this is entropically very expensive → there are always some chain ends that coil and thus result in amorphous areas.



Amorphous

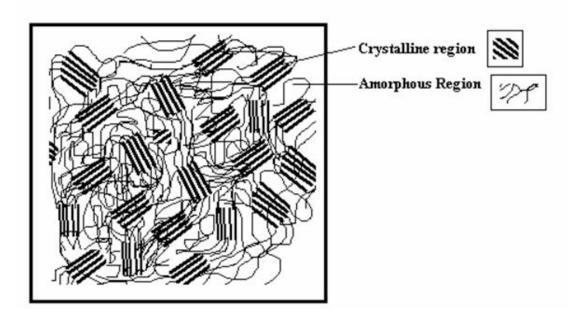


Semicrystalline

Propensity of polymers to crystallize

Typically, the propensity of polymers to crystallize is higher if they

- have rigid straight chains, side chains, or blocks
- have regularly spaced side groups
- are isotactic (compared to atactic polymers)



Examples of semi-crystalline polymers

linear poly(ethylene)

poly(tetrafluoroethylene)

poly(ethylene terephthalate)

$$\begin{array}{c} O \\ \bullet \\ \bullet \\ \bullet \\ \end{array}$$













Liquid crystalline polymers

Main-chain liquid crystal polymers

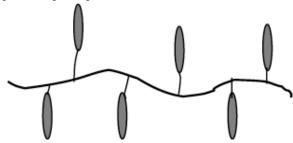


- The main chain contains liquid crystals.
- If the polymer has side chains, they are flexible.

Side-chain liquid crystal polymers



side chain side-on



side chain end-on

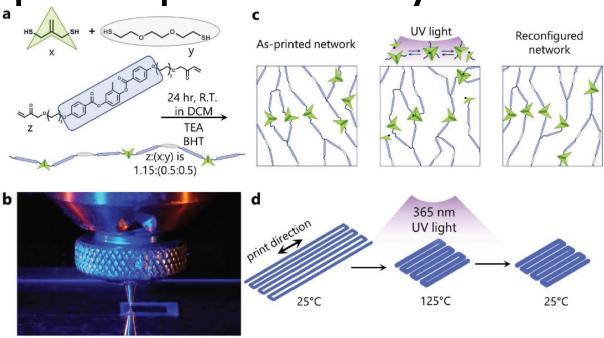
- The main chain is flexible.
- The side chains are composed of or contain liquid crystals.

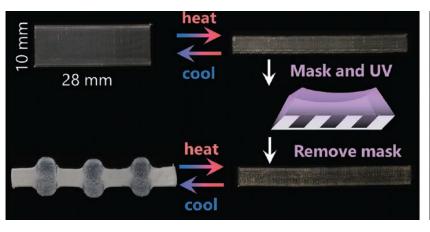
Examples of liquid crystalline polymers

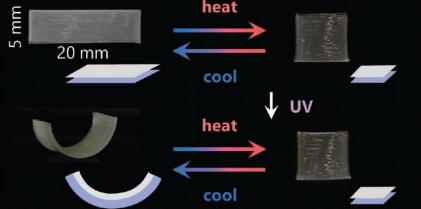
Kevlar



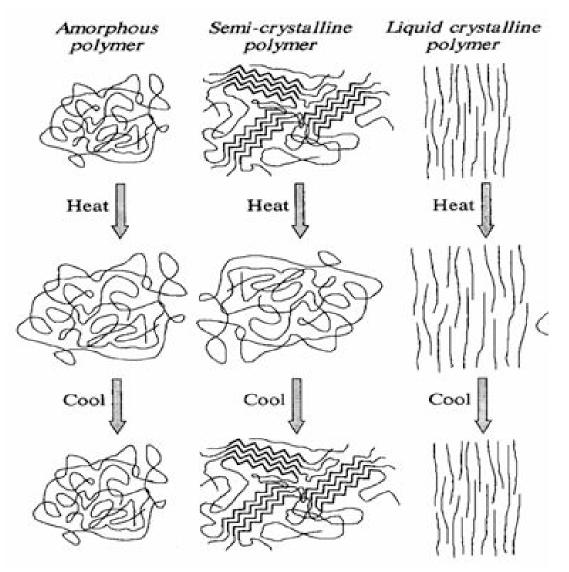
Recap: Shape memory materials







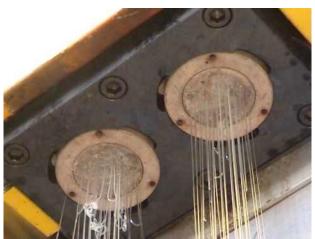
T-dependent polymer structures



Processing of polymers

from melts





from solutions





Processing of polymers in nature



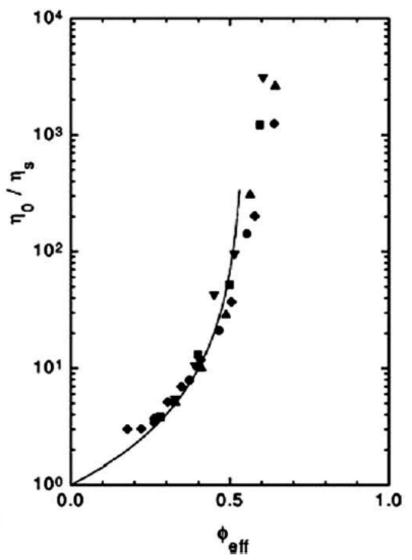
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What happens to the viscosity of a solution if polymers are added?

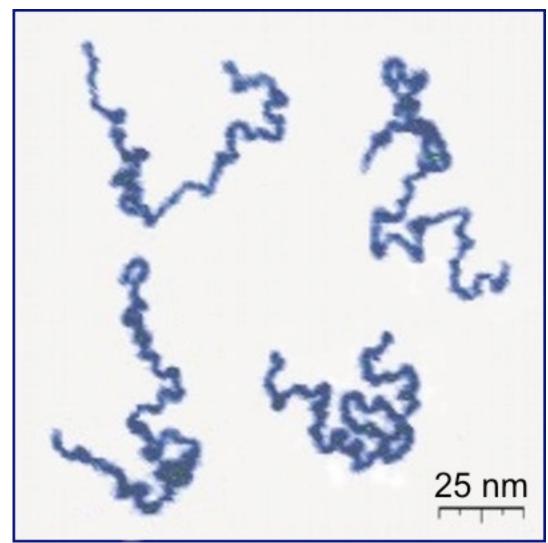
- A. It remains unchanged.
- B. It decreases with increasing polymer concentration.
- C. It increases with increasing polymer concentration.
- D. It starts behaving as a Newtonian fluid.

Viscosity of polymer solutions



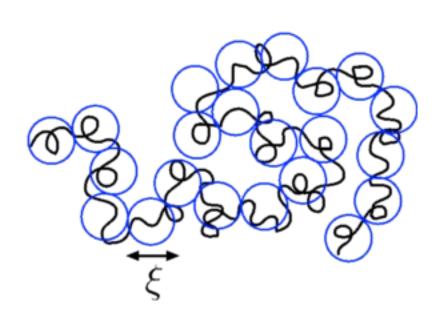
Dimensions of a polymer



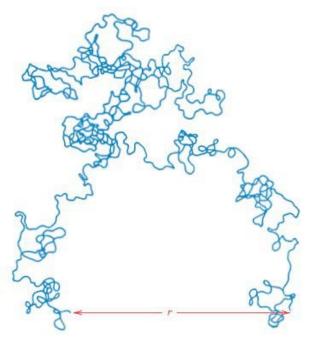


Ideal chains

The simplest way to describe a polymer is to assume that its individual segments undergo random walk and do not interact with each other. This description is called the **Gaussian limit**.

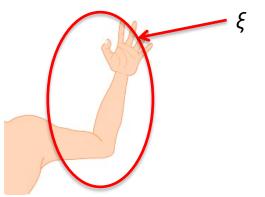


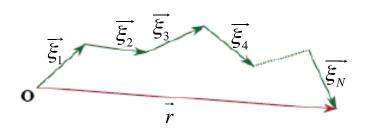
http://www.eng.yale.edu/polymers/docs/classes/polyphys/lecture_notes/3/handout3_wse3.html



https://capolight.wordpress.com/2010/06/14/a-brief-explanation-of-polymers/

Gaussian limit: Kuhn length





$$\vec{r} = \vec{\xi}_1 + \vec{\xi}_2 + \dots + \vec{\xi}_N = \sum_{i=1}^N \vec{\xi}_i$$

$$\left\langle \overrightarrow{rr} \right\rangle = \left\langle \left(\sum_{i=1}^{N} \overline{\xi}_{i} \right) \left(\sum_{i=1}^{N} \overline{\xi}_{i} \right) \right\rangle$$

$$\left\langle \overrightarrow{r} \right\rangle = \left\langle \sum_{i} \sum_{j} \overrightarrow{\xi_{i}} \overrightarrow{\xi_{j}} \right\rangle = N \xi^{2} + \sum_{i \neq j} \overrightarrow{\xi_{i}} \overrightarrow{\xi_{j}}$$

for ideal chains and theta solvents:

$$\langle r^2 \rangle = N \xi^2$$

ξ: Kuhn length [m]

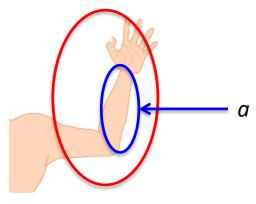
a: persistence length [m]

N: number of repeat "units" [-]

 R_g : radius of gyration [m]

r: end-to-end distance [m]

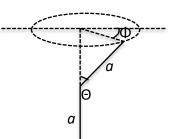
Gaussian limit: persistent length



What about poly(styrene) who has a restricted bond angle Φ?

\(\xi\$: Kuhn length [m]\(\alpha\$: persistence length [m]\(\text{N}: number of repeat "units" [-]\(R_a\$: radius of gyration [m]

Schematic illustration of a section of a polymer chain:



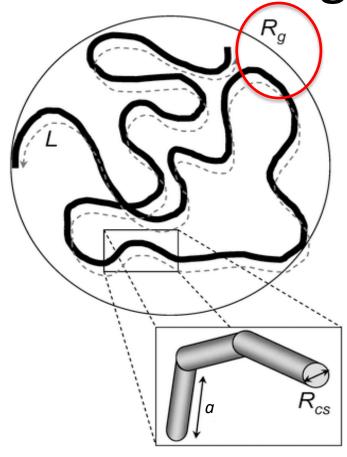
For freely rotating chains in theta solvents (no restrictions on Φ):

$$\langle r^2 \rangle_0 = Na^2 \left(\frac{1 + \cos \theta}{1 - \cos \theta} \right)$$

For hindered rotating chains in theta solvents (with a restricted bond angle Φ):

$$\langle r^2 \rangle_0 = Na^2 \left(\frac{1 + \cos\theta}{1 - \cos\theta} \right) \left(\frac{1 - \langle \cos\phi \rangle}{1 + \langle \cos\phi \rangle} \right)$$

Additional length scales

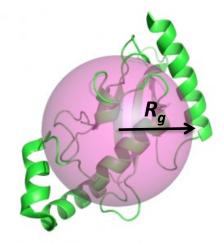


C. A. Dreiss, Soft Matter 3, 956 (2007)

L: contour length [m]a: persistence length [m]N: number of repeat "units" [-]R_q: radius of gyration [m]

Radius of gyration, R_g

The radius of gyration is the distance from the mass center of an object with mass m, that gives an equivalent inertia (with mass m) if the object is spherical.



http://www.pymolwiki.org/index.php/Radius_of_gyration

For an ideal chain, the segments around the center of mass have a Gaussian distribution. Ideal chains are thus often referred to as Gaussian chains.

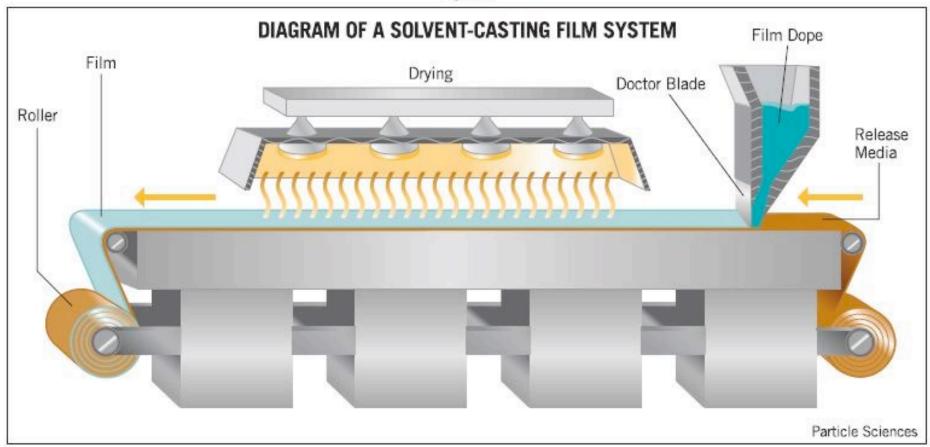
$$\sqrt{\langle R_g^2 \rangle} = \sqrt{\frac{a^2 N}{6}} = \sqrt{\frac{\langle r^2 \rangle_0}{6}}$$

Which length scale do you use to calculate the volume fraction of polymers in solution?

- A. persistence length
- B. Kuhn length
- C. radius of gyration
- D. end-to-end distance
- E. contour length

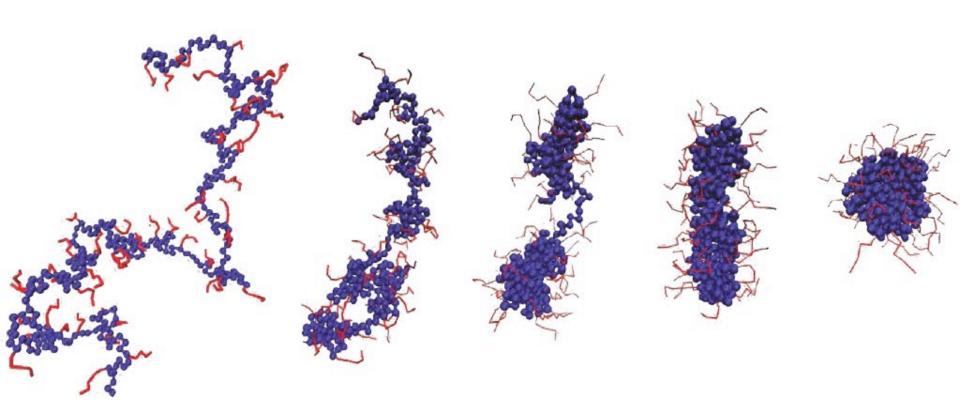
Polymer solutions

Figure 1



http://www.particlesciences.com/news/technical-briefs/2010/dissolving-films.html

Polymers in solution



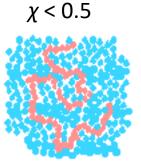
Solvent quality

The solvent quality is a measure for the interactions between polymer repeat units and the solvent molecules.

Good solvent:

Solute-solvent interactions are favored such that the chain is swollen.

Interaction parameter:



Theta solvent:

The interactions between different solute molecules are equal to those between solutes and solvents.

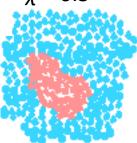
$$\chi = 0.5$$



Poor solvent:

The solute-solute interactions are favored such that the chain is contracted.

$$\chi > 0.5$$



How does the solvent quality influence the end-to-end distance of polymers?

$$\sqrt{\left\langle {m r}^2
ight
angle } \propto {m N}^{v}$$

good solvent:

theta solvent:

poor solvent:

$$v = 0.588$$

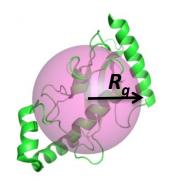
$$\upsilon = 0.5$$

$$v = \frac{1}{2}$$

υ: Flory exponent [-]

 χ : interaction parameter [-]

Influence of solvent on polymer dimensions

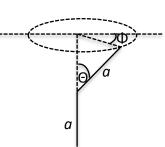


http://www.pymolwiki.org/ind ex.php/Radius_of_gyration For chains with no steric restrictions:

$$\sqrt{\langle R_g^2 \rangle} = \sqrt{\frac{\langle r^2 \rangle}{6}} = \frac{N^{\nu} a}{\sqrt{6}}$$

a: persistence length [m]N: number of repeat "units" [-]R_a: radius of gyration [m]

For chains with steric restrictions:



For freely rotating chains (no restrictions on Φ):

$$\sqrt{\langle r^2 \rangle} = N^{\nu} a \sqrt{\left(\frac{1 + \cos \theta}{1 - \cos \theta}\right)}$$

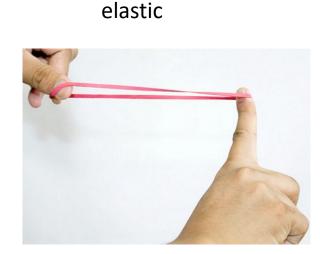
For hindered rotating chains (with a restricted bond angle Φ):

$$\sqrt{\langle r^2 \rangle} = N^{\nu} a \sqrt{\left(\frac{1 + \cos \theta}{1 - \cos \theta}\right) \left(\frac{1 - \langle \cos \phi \rangle}{1 + \langle \cos \phi \rangle}\right)}$$

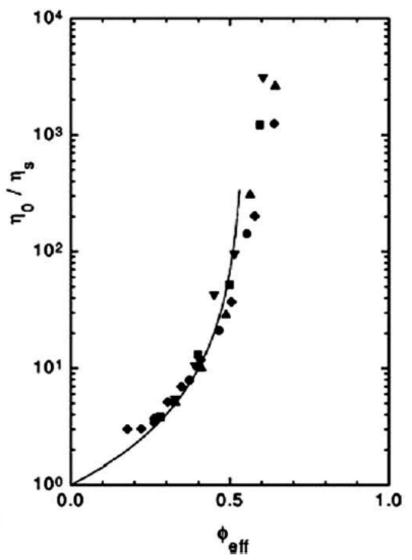
Flow behavior/elastic properties







Viscosity of polymer solutions



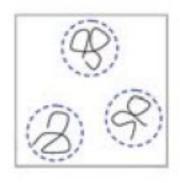
Polymer concentration

Do adjacent polymers interact with each other?

dilute

semi-dilute

concentrated



$$c < c_m^*$$

viscous



$$c = c_m^*$$

viscoelastic

$$c_m^* = \frac{N}{R_a^3}$$

$$c_m^* \propto N^{1-3\nu}$$



$$c > c_m^*$$

elastic

 c_m^* : coil overlap concentration [molecules/m³] N: number of repeat units [-] R_a : radius of gyration [m] υ: Flory exponent [-]

A polymer has a molar mass of 86 kg/mol and is composed of 1000 identical monomers, each 2 nm in length. What is the end-to-end distance?

- A. 7.7 nm
- B. 77 nm
- C. 770 nm
- D. 2000 nm
- E. 5000 nm

$$\langle r^2 \rangle_0 = Na^2 \left(\frac{1 + \cos \theta}{1 - \cos \theta} \right)$$

$$\Theta = 71^{\circ}$$

With this end-to-end distance and assuming the polymer behaves like an ideal chain, what is the concentration at which the solution enters the semi-dilute regime?

- A. $4.1 \, \mu M$
- B. 4.1 mM
- C. 41 mM
- D. 410 mM
- E. 4100 M

$$c_m^* := \frac{N}{R_g^3}$$

 $N_A = 6.02 \times 10^{23}$ molecules/mol

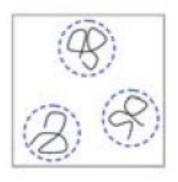
Polymer concentration

Do adjacent polymers interact with each other?

$$c_m^* := \frac{N}{R_g^3}$$
 $c_m^* \propto N^{1-3\nu}$

$$c_m^* \propto N^{1-3\nu}$$

dilute



 $c < c_m^*$

viscous

semi-dilute



viscoelastic

 c_m^* : coil overlap concentration

[molecules/m³]

N: number of repeat units [-] R_a : radius of gyration [m]

υ: Flory exponent [-]

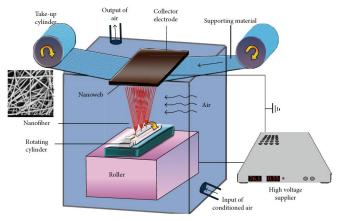
concentrated



 $c > c_m^*$

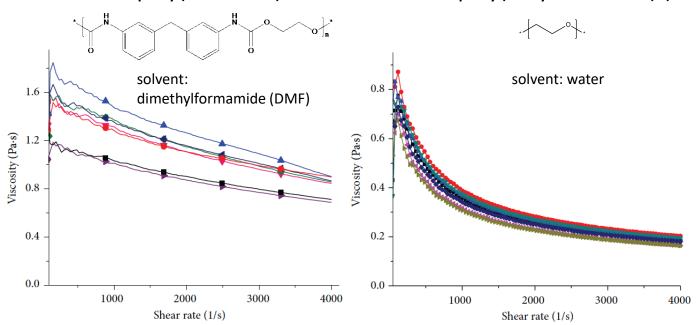
elastic

Viscosity of polymer solutions



poly(urethane) FIGURE 1: Diagram of the roller electrospinning system. poly(ethylene oxide) (400 kDa)

44

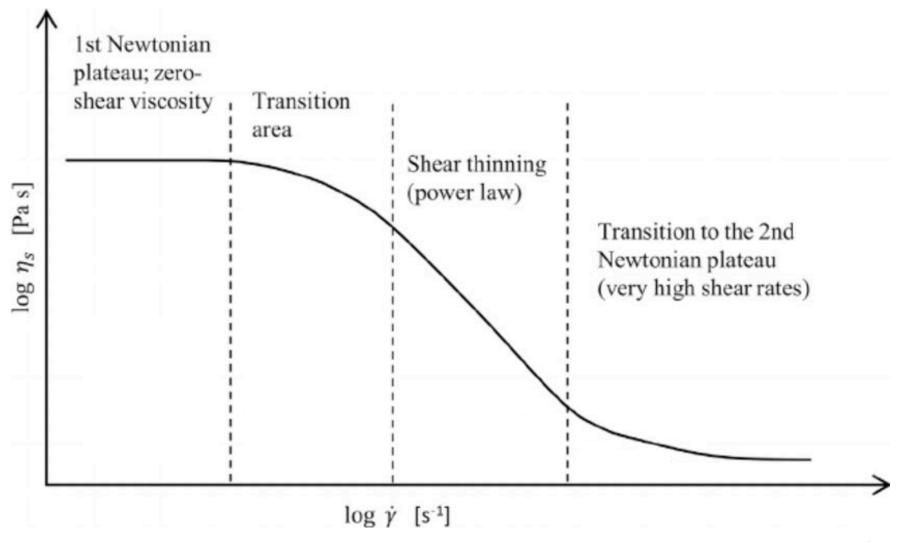


F. Yalcinkaya, B. Yalcinkaya and O. Jirsak, Journal of Nanomaterials, 2015

Outline

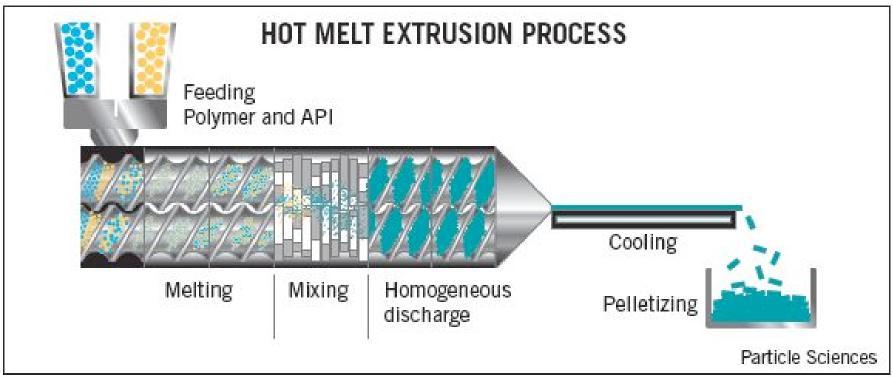
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Flow behavior of polymer melts



Polymer melts

Figure 1

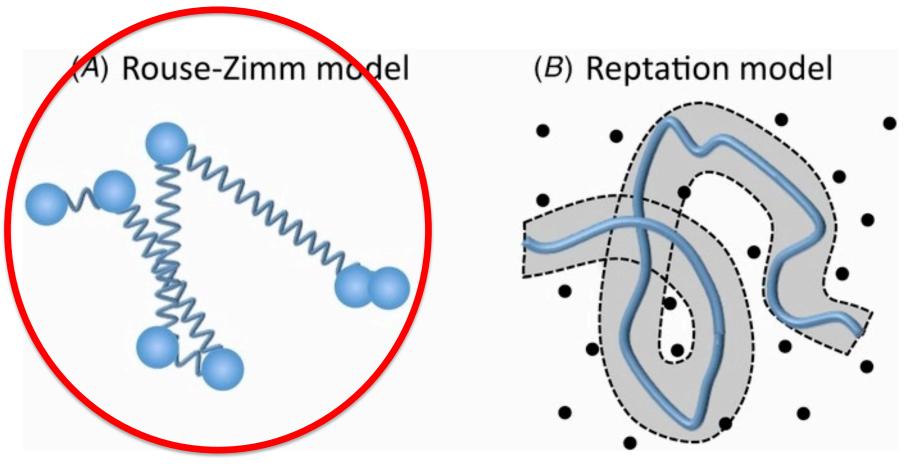


http://www.particlesciences.com/news/technical-briefs/2011/hot-melt-extrusion.html

Flow behavior of spaghetti



Flow of polymers in melts

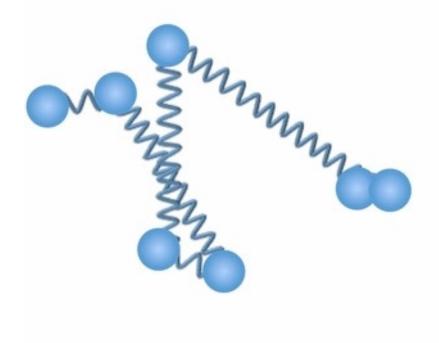


Flow of un-entangled polymers.

One chain moves through network of others.

Flow of polymers in melts

(A) Rouse-Zimm model



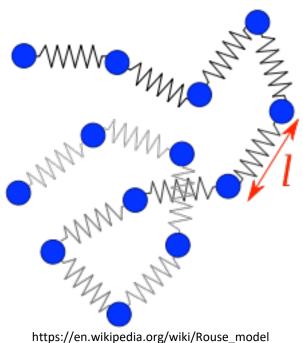




Flow of un-entangled polymers.

Rouse-Zimm Model

Description of conformational changes of an ideal chain.



- Segments are connected by springs
- Segment diffusion follows Brownian motion
- The diffusion coefficient scales with $D \propto \frac{1}{N^{\nu}}$

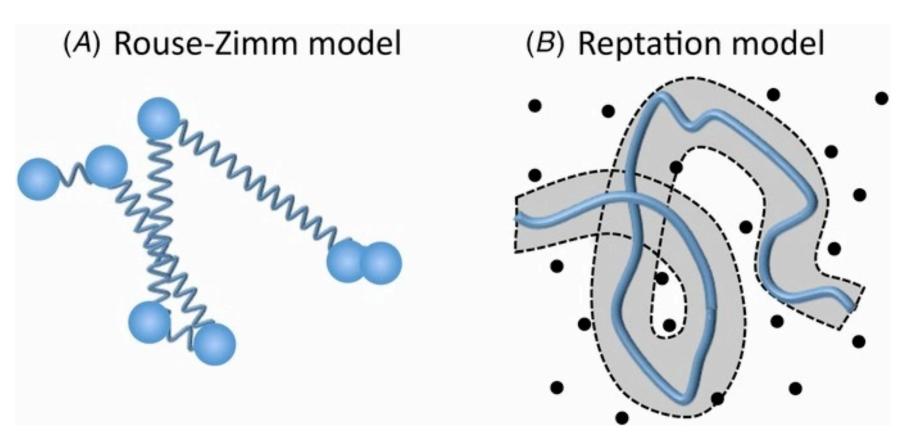
D: Diffusion coefficient [m²/s]

N: Number of segments [-]

v: Flory exponent [-]

But: It is only a good model for polymers that are not entangled.

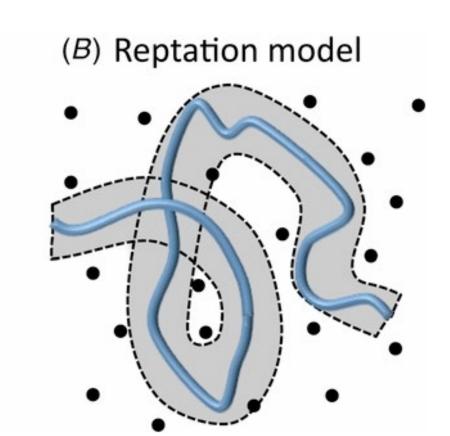
Flow of polymers in melts



Flow of un-entangled polymers.

One chain moves through network of others.

Flow of polymers in melts



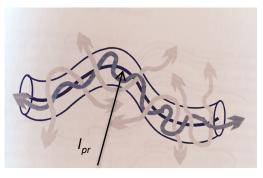


One chain moves through network of others.

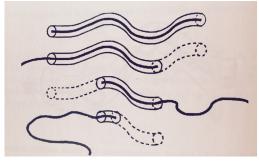
Reptation model

$$\langle r^2 \rangle = R_0^2 = N_R a_R^2 = l_{pr} \ a_{pr}$$

$$\widehat{D} = \frac{k_B T}{\zeta_p}$$



chain motion



G. Strobl, *The Physics of Polymers*. (Springer-Verlag, Berlin Heidelberg, 1997)

 $\langle r^2 \rangle$: average end-to-end distance [m²] I_{pr} : contour length of primitive path [m] a_{pr} : associated sequence length of primitive path [m]

D: associated curvilinear diffusion coefficient [m²/s]

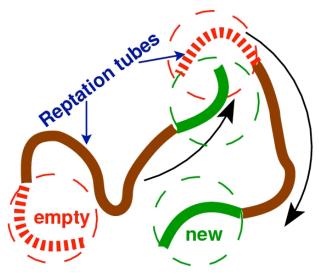
 ζ_{p} friction coefficient of the chain [Ns/m] τ_{d} : time required to disentangle [s]

Because there are no entanglements within the tube: $\zeta_P = N_R \zeta_R \Rightarrow \widehat{D} = \frac{k_B T}{N_R \zeta_R}$

The time required to lead to complete disentanglement: $\tau_d = \frac{l_{pr}^2}{\widehat{D}} \implies \tau_d \propto \zeta_R N_R^3$

Reptation

How does the polymer molecular weight influence the diffusion coefficient of polymers in melts?



 $\langle r^2 \rangle$: average end-to-end distance [m²] I_{pr} : contour length of primitive path [m] a_{pr} : associated sequence length of primitive path [m] \hat{D} : associated curvilinear diffusion coefficient [m²/s]

 $\zeta_{\vec{p}}$ friction coefficient of the chain [Ns/m] $\tau_{\vec{q}}$: time required to disentangle [s]

http://www-ics.u-strasbg.fr/etsp//research/poly/high snakedyn.php

For non-entangled polymer chains:

$$D = \frac{k_B T}{N_R^{\nu} \zeta_R} \propto \frac{1}{M_W^{\nu}}$$

For entangled polymer chains:

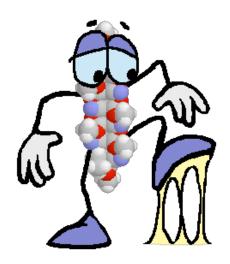
$$D = \frac{\langle r^2 \rangle}{6\Delta t}$$

$$D \propto \frac{l_{pr}a_{pr}}{\tau_d} \propto \frac{N_R}{N_R^3} \propto \frac{1}{M_W^2}$$

In which case does the viscosity increase most strongly if the molecular weight of the polymer/monomer is increased?

- A. of non-entangled polymers
- B. For melts composed of entangled polymers
- C. For melts composed of monomers

Viscosity of polymer melts



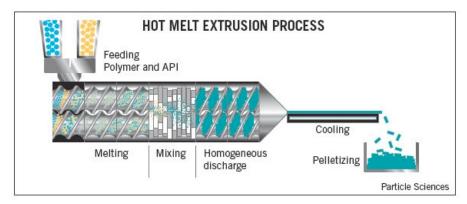
low M_{w} , no entanglements:

$$\eta_0 \propto M_w$$

high $M_{w'}$ entanglements:

$$\eta_0 \propto M_{\rm w}^{3.4}$$

Why does the viscosity of polymer melts matter?





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 - Elastomers
- Recycling of polymers
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- Application

Polymers

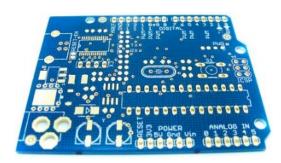
Thermoplasts

Can be molten and when cooled form an amorphous or glassy structure



Thermosets

Harden when the temperature is increased → crosslinking



Elastomer

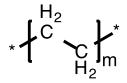
Can be stretched to many times their original dimension and recover if stress is released



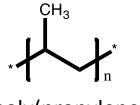
Thermoplasts



Examples:



poly(ethylene)

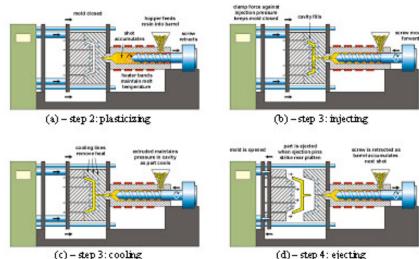


poly(propylene)

Analogy:



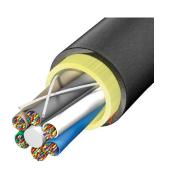
Processing:



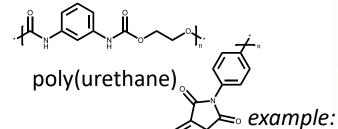
http://www.idsa-mp.org/proc/plastic/injection/injection process.htm

Polymers contained in thermoplasts are physically crosslinked, they are entangled.

Any change caused by heating is reversible, as long as the temperature is kept below the thermal decomposition temperature.



Examples:



poly(imide)

Analogies:

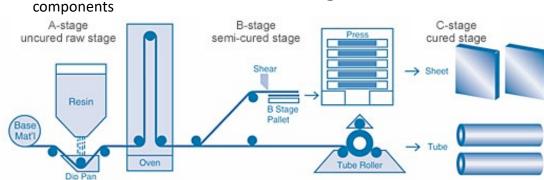




Thermosets

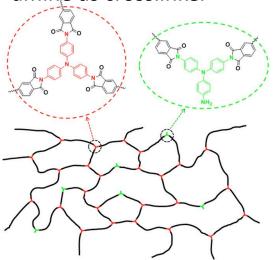


1. mix 2. cure **Processing:**



http://www.acculam.com/processes.html

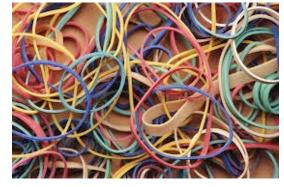
use of tris(4-aminophenyl) amine as crosslinker



X. Xiao, D. Kong, X. Qiu, W. Zhang, Y. Liu, S. Zhang, F. Zhang, Y. Hu and J. Leng, *Scientific Reports*, 2015, **5**, 14137

Polymers contained in thermosets are covalently crosslinked.

Elastomers



Examples:

poly(butadiene)

synthetic poly(isoprene)

Processing:

Elastomers are formed by crosslinking linear chains.

If disulfide bonds are used to crosslink linear chains, this process is called vulcanization.

Vulcanization of Natural rubber (Polyisoprene)

Polyisoprene chains with Sulfur Bridges

http://scinote.tumblr.com/

Analogy:

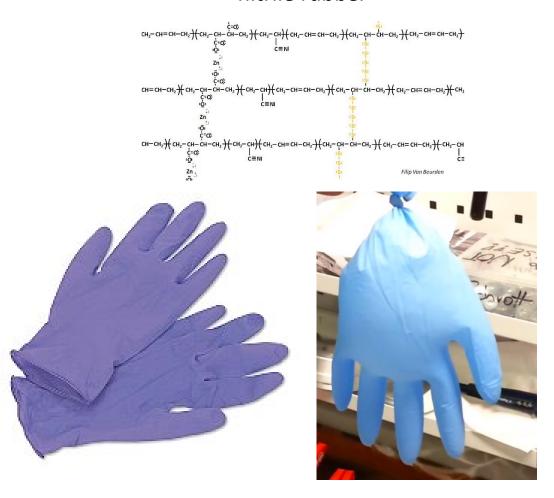


Elastomers are usually covalently crosslinked.

Elastomers have a low crosslink density, which makes them flexible.

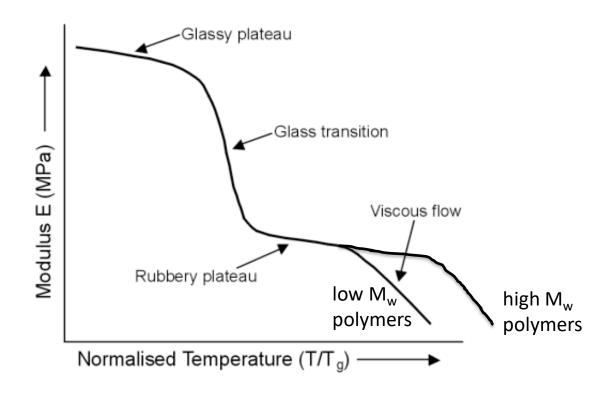
Elastomers

nitrile rubber



$$T_g \approx -15 \, ^{\circ}\text{C} - 30 \, ^{\circ}\text{C}$$

Elastic properties of polymers

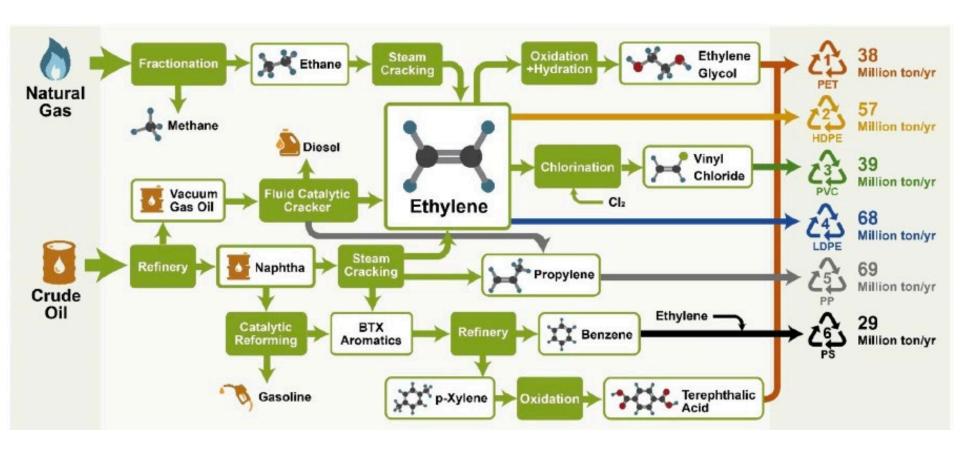


At T_g :
The change in the polymer conformation is small.
The change in the chain mobility is large.

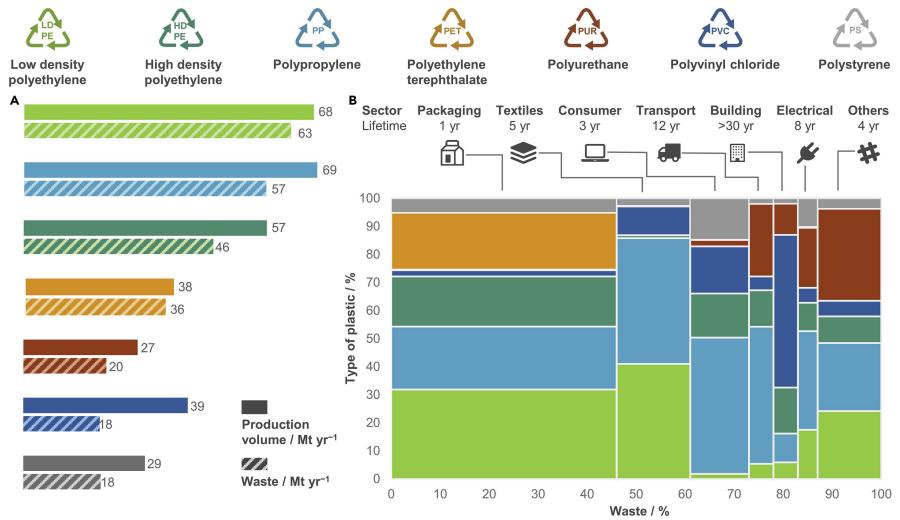
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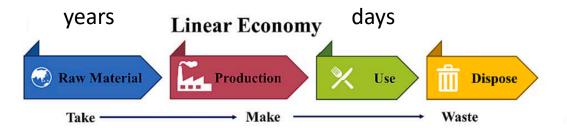
Plastic production



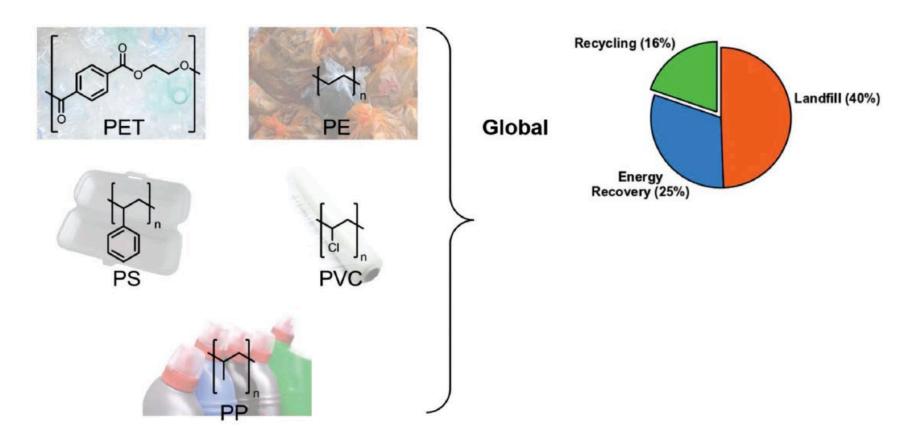
Plastic types



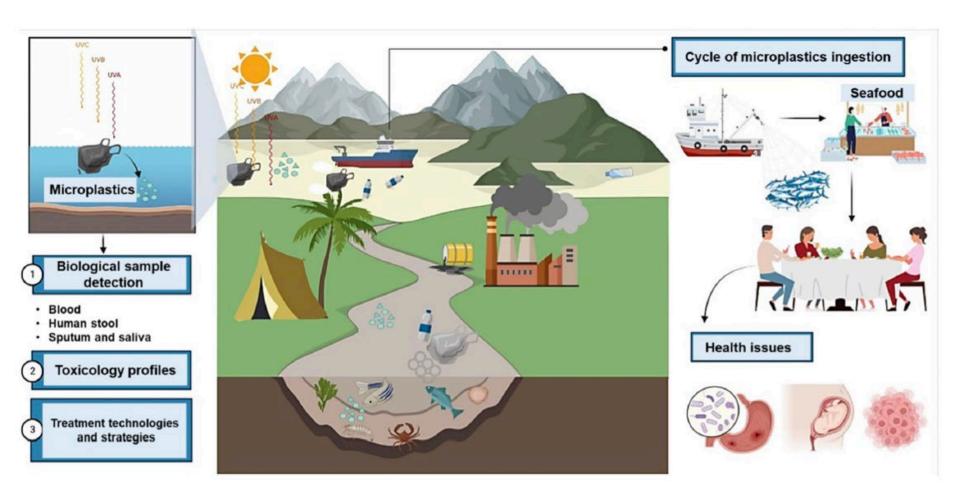
What is the problem with plastics?



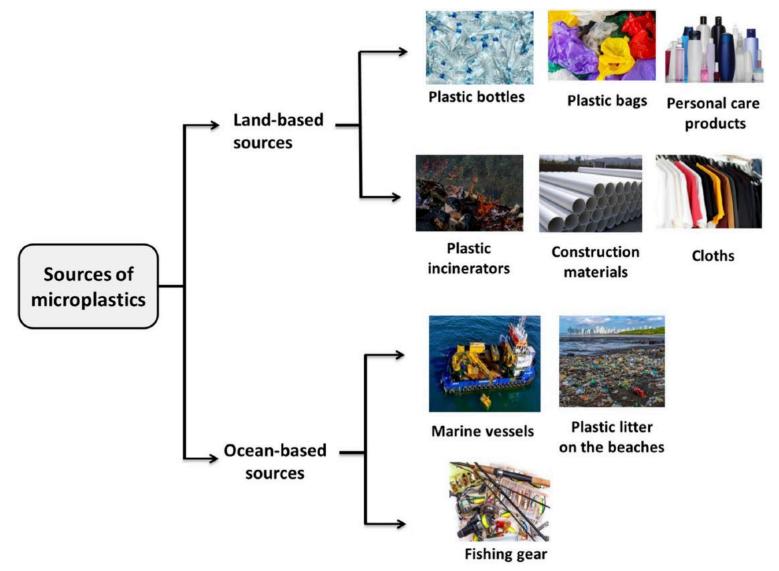
Recycling of polymers



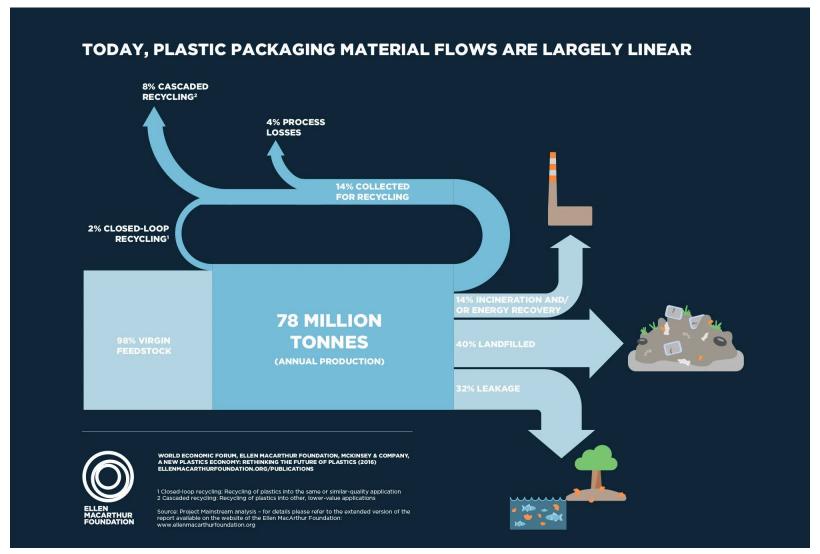
Problematic of landfill



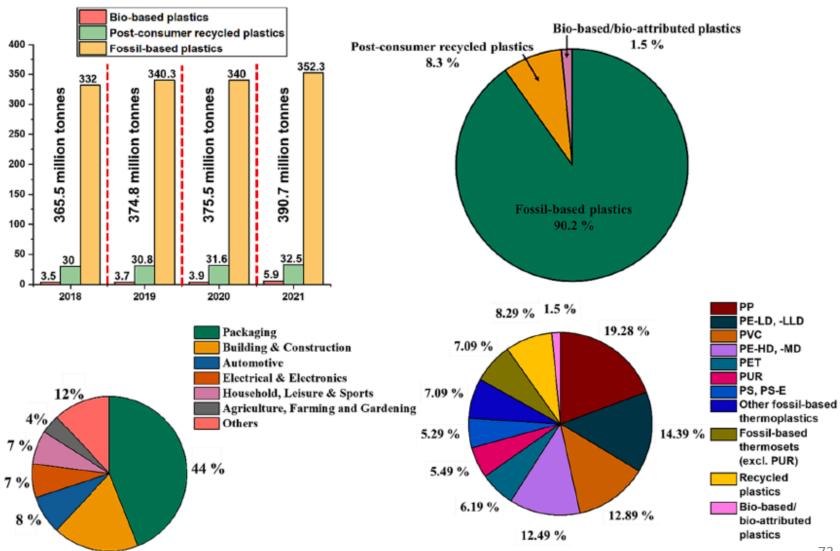
Sources of microplastics



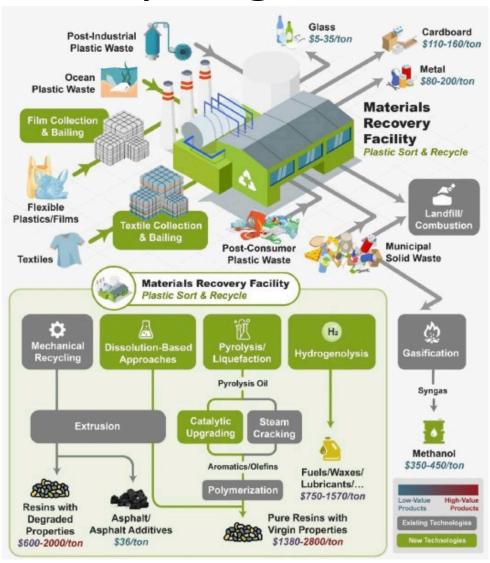
Recycling of plastic packaging



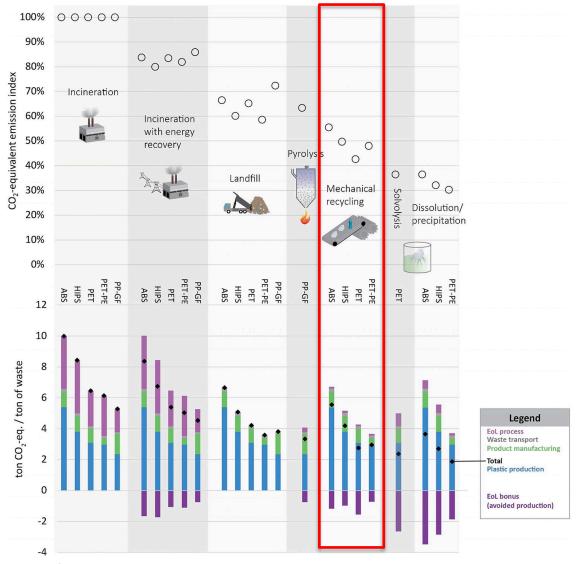
Recycling of polymers



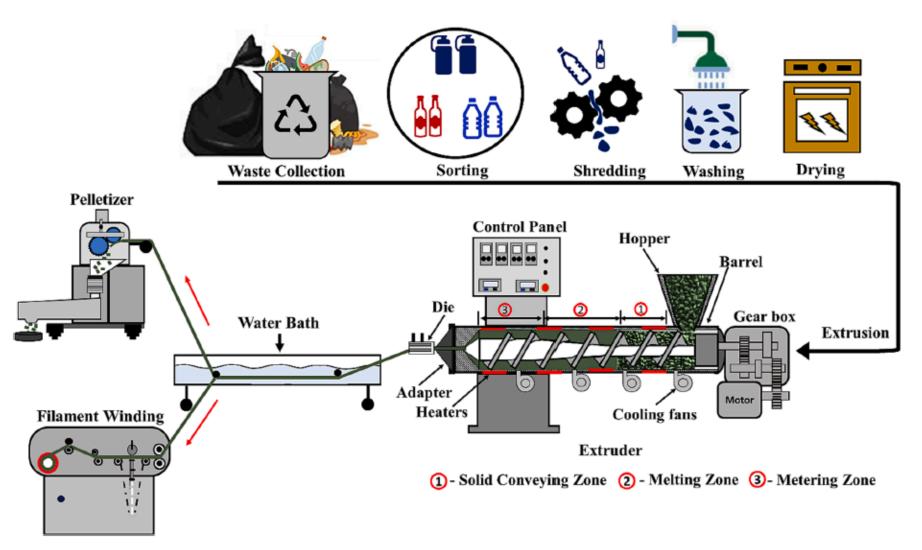
Recycling routes



Different recycling routes



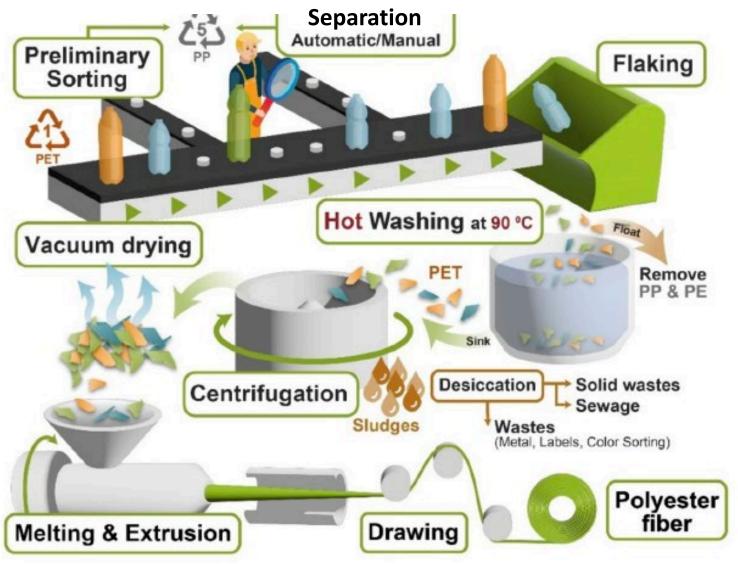
Mechanical recycling



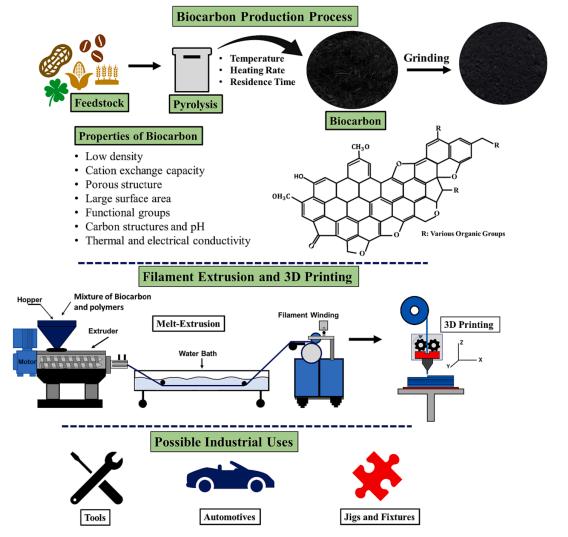
Mechanical recycling



Mechanical recycling



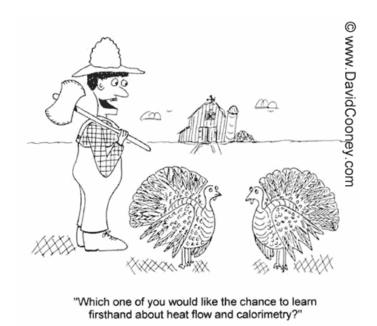
Transforming waste biomass into functional polymers



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Characterization of polymers





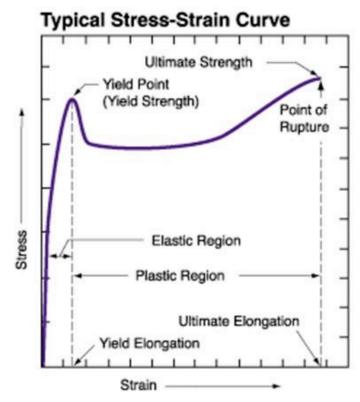
http://www.greatdreams.com/political/non-orwell-zebra.htm

Elastic materials

Tensile test



Thermoplastic polymer



http://www.kazuli.com/UW/4A/ME534/lexan2.htm

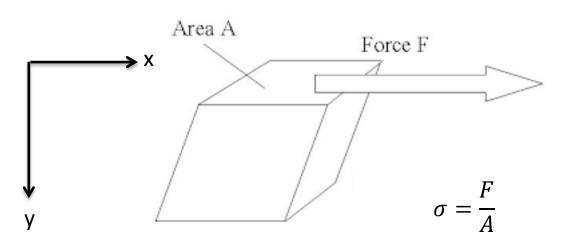
Viscous or viscoelastic materials





Rheology

The shear and storage moduli of polymers can be characterized with rheology.

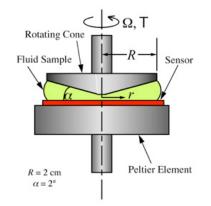


$$\tau = \dot{\gamma}\eta$$

$$\dot{\gamma} = \frac{dv_x}{dy}$$

$$\tau = \eta \frac{dv_z}{dz}$$

τ: shear stress [Pa]
η: viscosity [Pas]
ν: velocity [m/s]
F: force [N]
A: area [m²]
γ: shear rate [s⁻¹]



1. mode of operation: The top geometry of the rheometer rotates at a defined $\dot{\gamma}$ and you measure the stress required to reach this value.

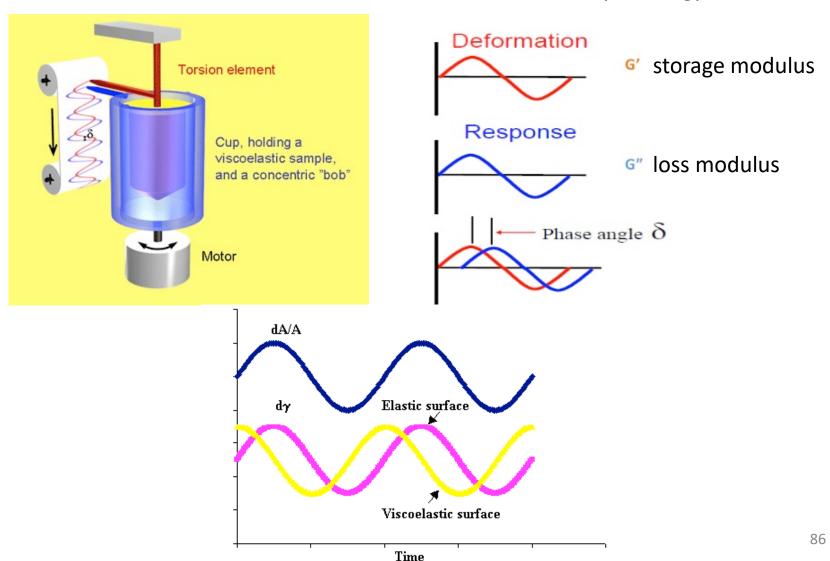
2. mode of operation: The top geometry of the rheometer rotates with a defined τ and you measure the resulting $\dot{\gamma}$.

Viscous or viscoelastic materials



Oscillatory Rheology

Viscoelastic materials are often characterized with oscillatory rheology:



Oscillatory Rheology

Viscoelastic materials are often characterized with oscillatory rheology:

 $\tan \delta = \frac{G''}{G'}$



$$\gamma = \gamma_0 \sin(\omega t)$$

δ: loss tangent, measures energy loss per cycle [rad]

$$\sigma = \sigma_0^I \sin(\omega t) + \sigma_0^{II} \cos(\omega t)$$

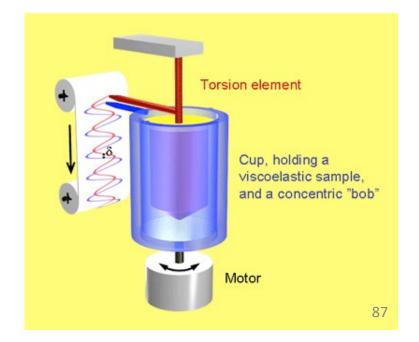
$$\sigma = \sigma_0' \sin(\omega t) + \sigma_0'' \cos(\omega t) \qquad \sigma = G' \gamma_0 \sin(\omega t) + G'' \gamma_0 \cos(\omega t)$$

storage modulus

$$G' = \frac{\sigma_0'}{\gamma_0}$$

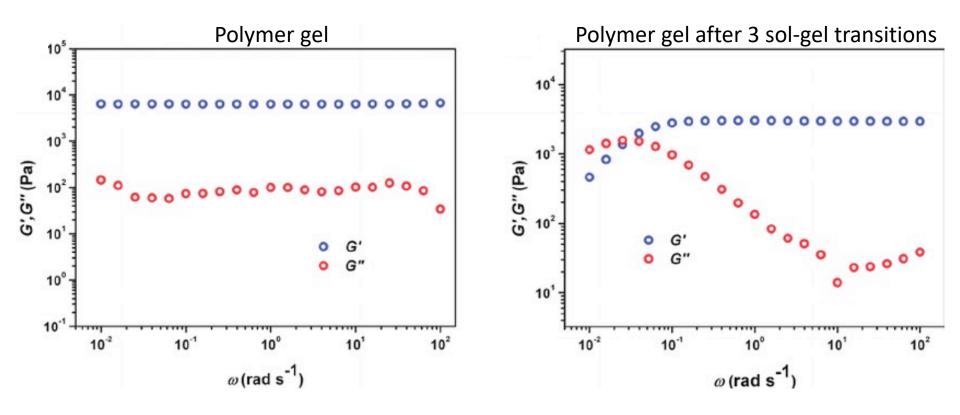
loss modulus

$$G'' = \frac{\sigma_0''}{\gamma_0}$$



Rheology on polymer gels

bis-acylhydrazine functionalized poly(ethylene oxide)



G. Deng, C. Tang, F. Li, H. Jiang and Y. Chen, *Macromolecules*, 2010, 43, 1191-1194

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3D printing of polymer solutions

