

Solid-Solid Interfaces

Lesson 8

MSE 304

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Key Topics in the Previous Class

$$\gamma_{\alpha\beta} = \gamma_{\alpha} + \gamma_{\beta} - W_{\alpha\beta}$$

fluid - fluid $\rightarrow$ diffuse interfaces

Reading for this Class

Chapter 13 Howe book

L: very little

L: all of it

Questions to be answered in Class

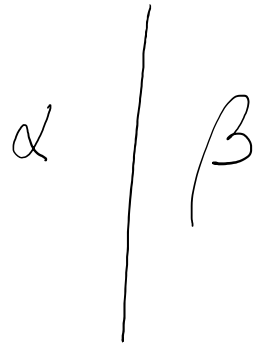
1) Is the TLK model compatible with a simple Langmuir model?



i molecule can form a monolayer

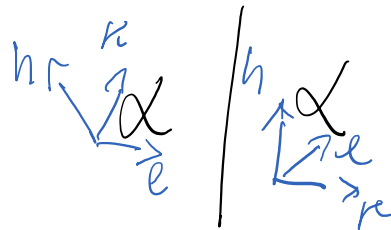
2)  does it always form micelle in a fluid phase & if the concentration exceeds the CMC?

Solid-Solid Interfaces



both α and β are solid

special case

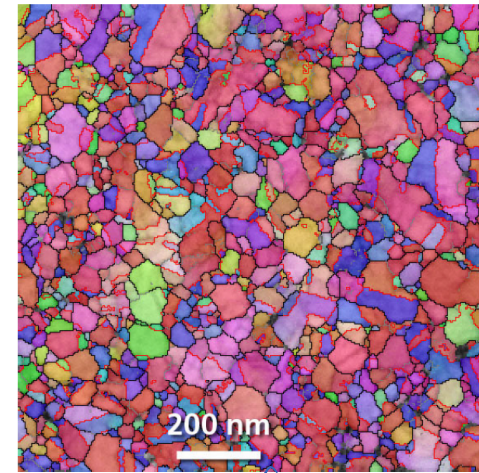
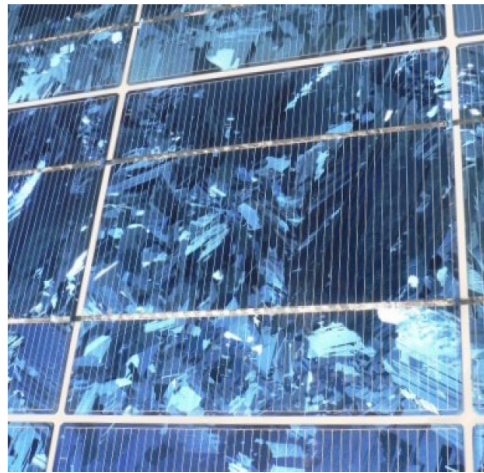


can be
 $\alpha \rightarrow$ solid crystalline phase

any pure material that is in a single phase but in a polycrystalline solid will have plenty of α - α interfaces

Solid-Solid Interfaces

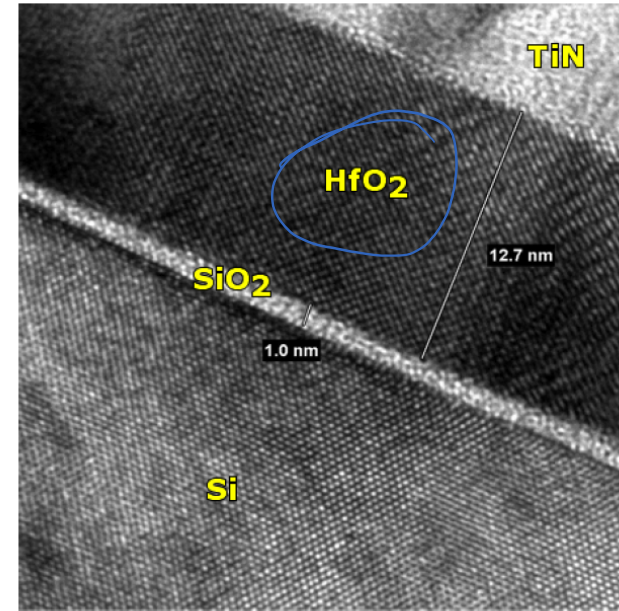
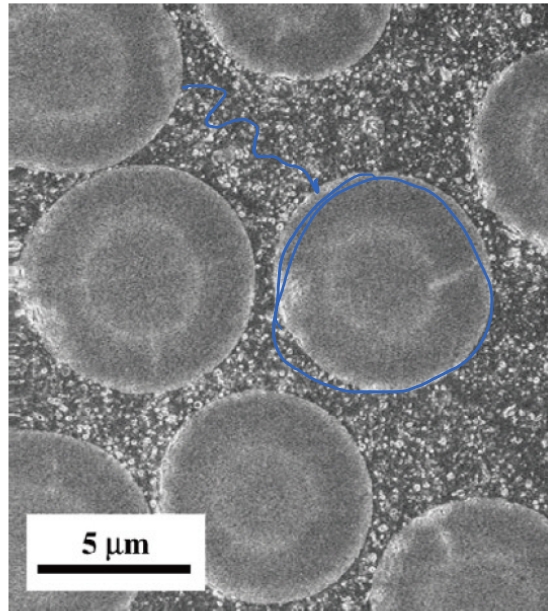
- Most bulk materials are polycrystalline, composed of many grains with different orientation
 - Grain boundaries affect mechanical or electronic properties
 - The story gets even more complex when it comes to interfaces between different phases (heterophase interfaces)



Grain
pure to
Au

Grain boundary is a solid-solid interface

Solid-Solid Interfaces



Grain boundary engineering is key to determine mechanical and electronic properties

Homophase and ~~Eteropahse~~ Solid-Solid Interfaces

Heterophase



↙
either α or β or both
can be amorphous
Heterophase

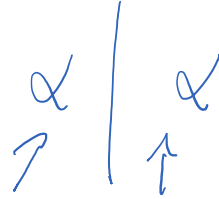


Homophase
interface

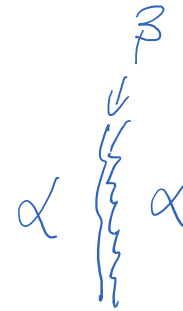
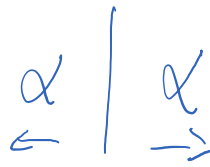
Homophase and Heterophase Solid-Solid Interfaces



clear distinction

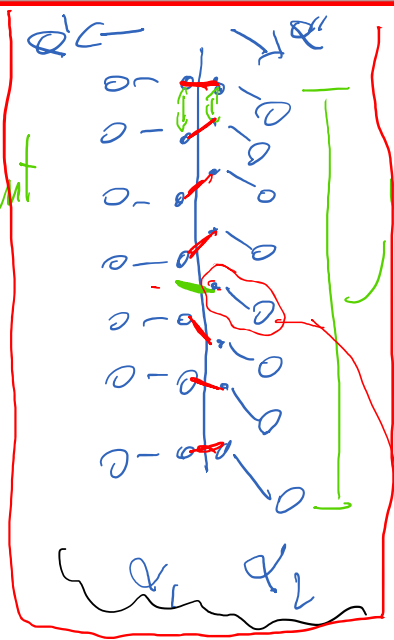


it depends



Homophase Solid-Solid Interfaces

Why
b is different
for
 α_1 and
 α_2

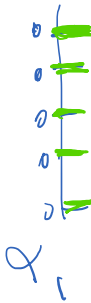


interface

$$\gamma_{\alpha_1, \alpha_2} = \gamma_{\alpha_1} + \gamma_{\alpha_2} - W_{\alpha_1, \alpha_2}$$

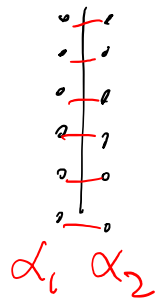
break break reform

energy of a single bond known
plus the elastic energy
to stretch all the red
reformed bonds past their
equilibrium length



Solid-Solid Interfaces

$$\lim_{\sigma \rightarrow 0} \gamma_{\alpha_1 \alpha_2} = \gamma_{\alpha_1} + \gamma_{\alpha_2} - W_{\alpha_1 \alpha_2} = 0$$



$$W_{\alpha_1 \alpha_1} = 2 \gamma_{\alpha_1}$$

(lim)
 $\sigma \rightarrow 0$

a few dangling bond +
↑ plus elastic energy

homophase solid surfaces $\rightarrow W_{\alpha_1 \alpha_2} \approx \gamma_{\alpha_1} + \gamma_{\alpha_2}$

Solid-Solid Interfaces

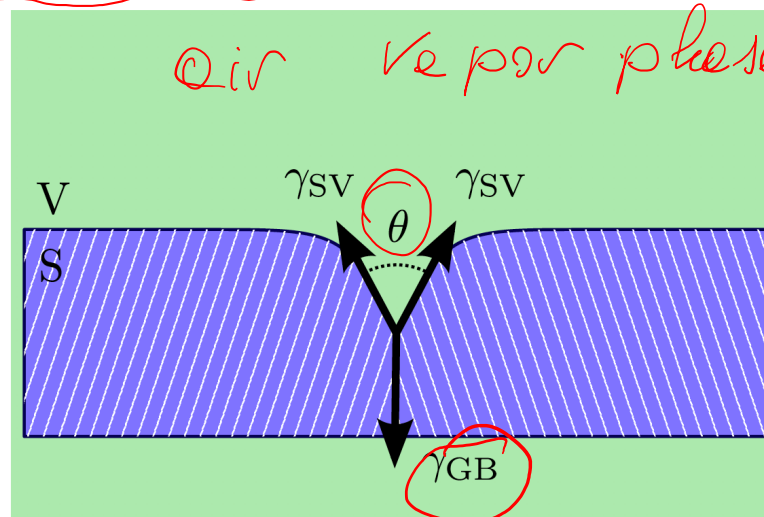
- One can associate a surface-energy term γ_{GB} to the presence of GBs

$$\gamma_{GB} = 2\gamma_{SV} - W_{GB}$$

- The GB energy can be estimated by thermal grooving
- Balance between surface energy γ_{SV} and γ_{GB} (typically $\gamma_{SV} \approx 3\gamma_{GB}$)

$$2\gamma_{SV} \cos \frac{\theta}{2} = \gamma_{GB}$$

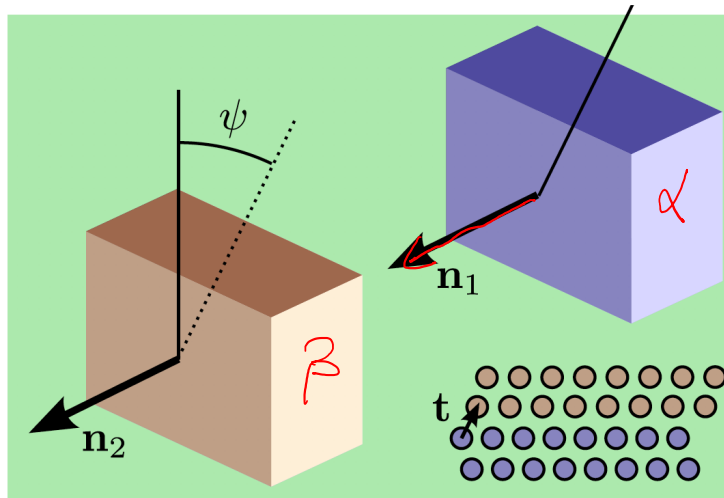
- What happens if $\gamma_{GB} > 2\gamma_{SL}$?



Solid-Solid Interfaces

for
your
consideration

- How many parameters we need to describe the geometry of a GB?
 - Orientation of the two surfaces + relative tilt = 5 degrees of freedom
 - Atomic-scale register between the surfaces
- Simpler geometry - symmetric tilt boundaries



β could
be α_2
5 degrees
of freedom
between
 $\vec{n}_1$ and $\vec{n}_2$

Conclusions

$$\gamma_{\alpha\beta} = \underbrace{\gamma_{\alpha} + \gamma_{\beta} - W_{\alpha\beta}}_{\text{count interfacial dislocations}}$$

$$W_{\alpha\beta} \approx \gamma_{\alpha} + \gamma_{\beta}$$

or
(elastic) compute interfacial stress
or
both