ChE 430

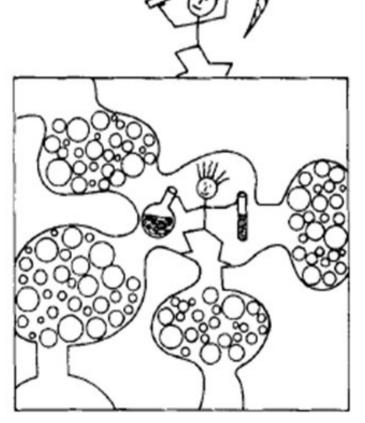
Colloidal synthesis of nanoparticles and their energy applications

MODULE 3: Nucleation and Growth

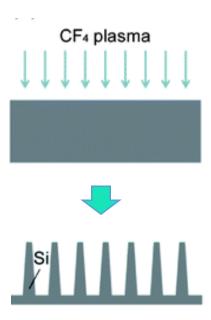
- 3.1. Introduction
- 3.2. The Classical Nucleation and Growth Theory
- 3.3. The Non-Classical Nucleation and Growth Theory
- 3.4. Chemical reaction mechanisms
- 3.5. In-situ studies

Top-down and bottom-up

Bottom-up approaches (chemical vapor deposition, laser pyrolysis, chemical synthesis)

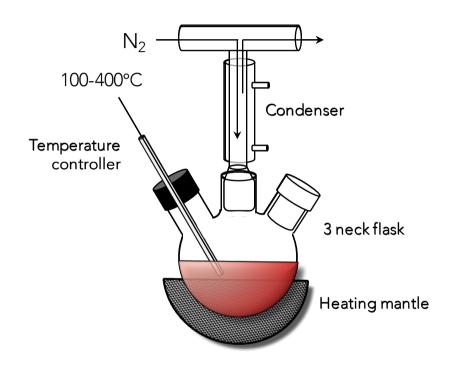


Top-down approaches (sputtering, lithography, plasma etching)



Colloidal synthesis

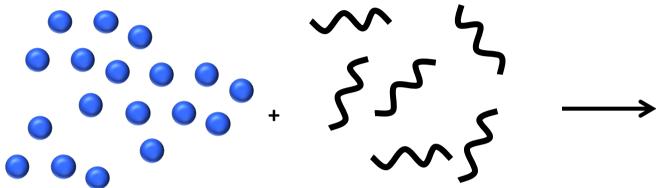


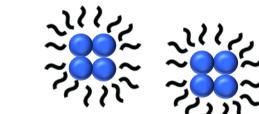


Reaction parameters:

- Ligands/Precursors reactivity
- Reaction temperature
- Reaction time
- Concentration

Building nanocrystals from atoms





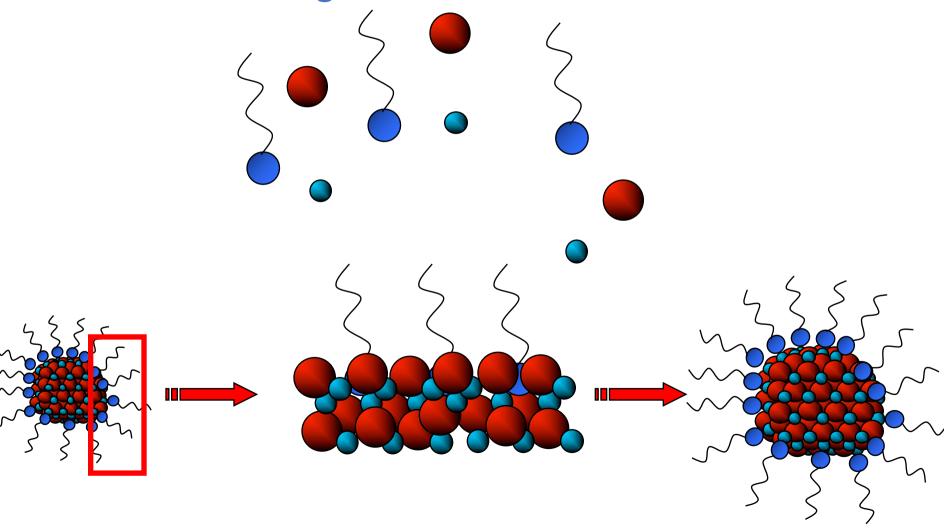
Precursors

- salts
- organometallic complexes
- elemental compounds

Ligands

- carboxylic acids
- amines
- thiols
- phosphonic acids
- phosphines

Ligands are in dynamic equilibrium between NC surface and the growth solution



"..the current state of nanocrystal synthesis is comparable to that of organic synthesis a century ago"

(Buhro et al. Chem. Mater. 2014)

"While organic reactions that underpin molecular total synthesis can be applied rationally in a by-design manner, the rules and guidelines for constructing high-order hybrid nanoparticles are poorly understood and far from mature"

(Schaak et al. J. Am. Chem. Soc. 2015)

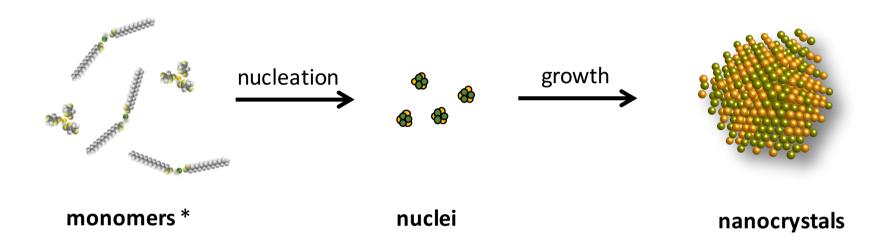
Nucleation: The process whereby nuclei (seeds) act as templates for crystal growth.

Homogeneous Nucleation: it occurs when nuclei form uniformly throughout the parent phase

Heterogeneous Nucleation: it occurs when nuclei form on a pre-existing surface (container surface, impurities, nanocrystal seeds)

2.2. The Classical Nucleation and Growth Theory

2.3. The Non-classical Nucleation and Growth Theory



The growth proceeds through addition of monomers to the nuclei

^{*}Monomers might be different from the precursors injected in the flask

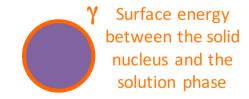
Free energy of nuclei formation

$$\Delta G_N = 4\pi r^2 \gamma + \frac{4}{3}\pi r^3 \Delta G_V$$

$$\Delta G_V = \frac{-k_B T ln(S)}{V_m}$$

 k_B is the Boltzmann's constant T is the temperature S is the supersaturation of the solution V_m is the molar volume

Surface energy: energetic penalty to pay in order to establish an interface between the new solid phase and the pre-existing solution phase.



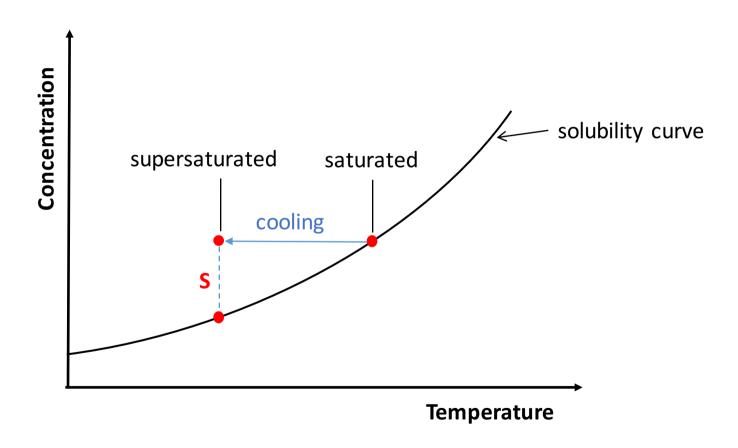
Bulk energy: energetic gain from the formation of a new spherical nucleus

QUESTION: are you familiar with the definition of unsaturated, saturated and supersaturated solution?

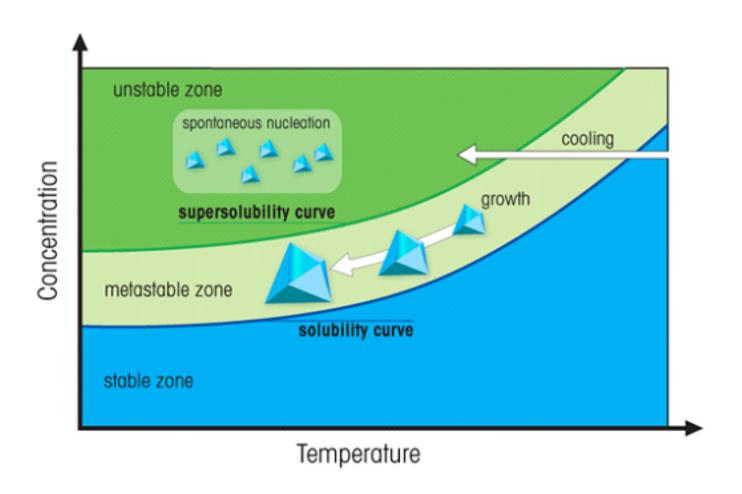
What does supersaturation mean?

How do we get out of supersaturation?

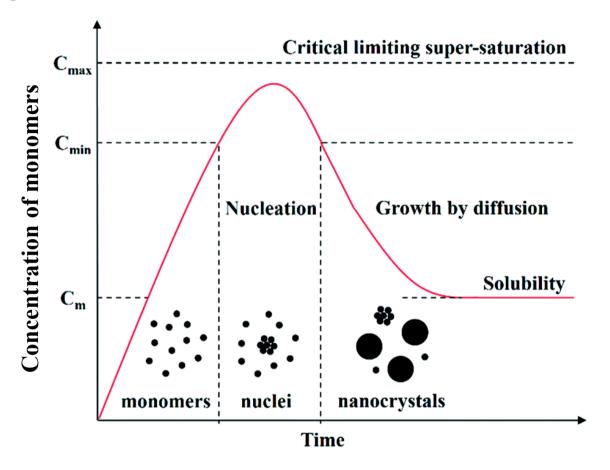
Supersaturation (S): The difference between the actual concentration and the solubility concentration at a given temperature



Supersaturation: The difference between the actual concentration and the solubility concentration at a given temperature

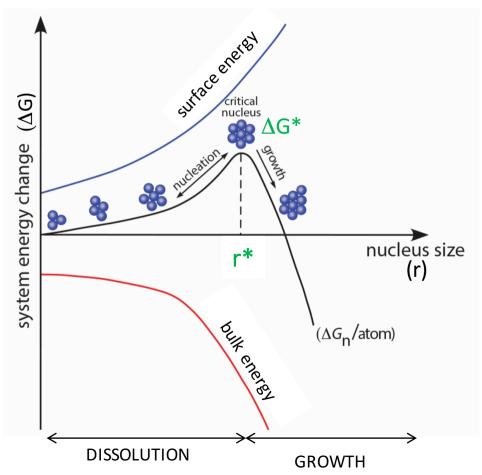


La Mer diagram



The goal is the achieve supersaturation which will lead to one brief outburst of nucleation. Above the critical limiting super-saturation, only spontaneous nucleation and no growth will occur.

$$\Delta G_N = 4\pi r^2 \gamma - \frac{4}{3}\pi r^3 \frac{k_B T \ln(S)}{V_m}$$



$$\Delta G^* = \frac{4}{3}\pi\gamma(r^*)^3 \quad \text{from } \frac{d\Delta G}{dr} = 0$$

$$\mathbf{r^*} = \frac{2\gamma V_m}{RT \ln S} \qquad \text{from } \Delta G = \Delta G^*$$

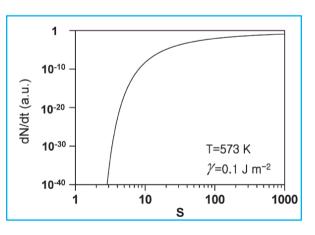
r < r* nuclei form but redissolve (Gibbs-Thomson effect)

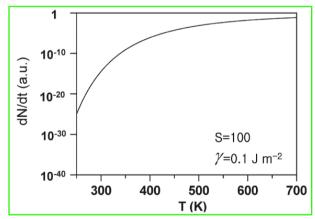
r > r* nuclei form and grow

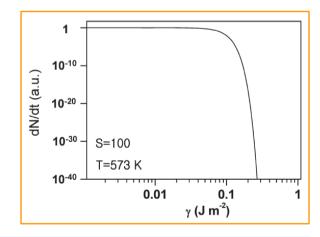
(The **critical-nucleus size** is in the range of tens to hundreds of atoms, 1-3 nm)

The rate of nucleation of N particles during time t can be described using an Arrhenius type equation:

Experimentally controllable parameters influencing the nucleation rate:



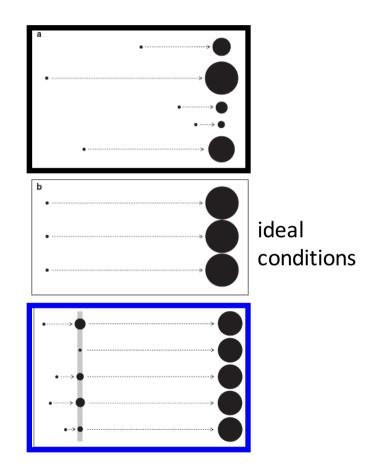




As the level of supersaturation is increased from S = 2 to 4, the nucleation rate is increased by $\approx 10^{70}$ times

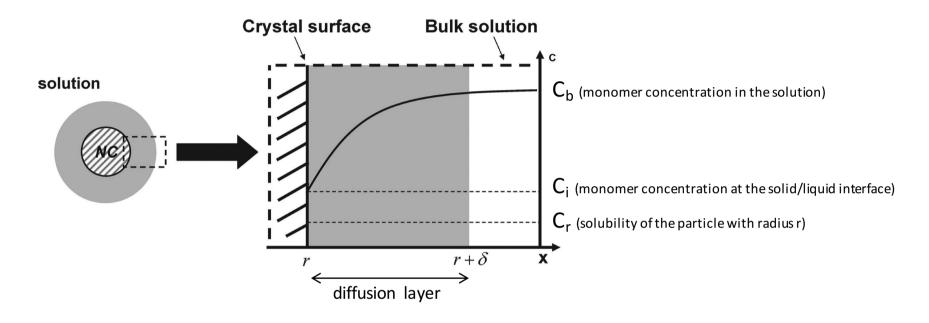
Size distribution is largely dependent on the time over which nuclei are formed:

- A narrow time window for nucleation leads to monodisperse nanocrystals (assuming we are in a focusing regime)
- A wider time window for nucleation leads to polydisperse nanocrystals



Diffusion of monomer to the surface
 Surface reaction
 Diffusion
 Dissolution

We can model the growth using the Fick's Law (ChE-204)



ASSUMPTION: steady-state (*J = const*)

$$J = 4\pi x^2 D \frac{\mathrm{d}C}{\mathrm{d}x}$$

J is the total flux of monomers
x is the radius of the spherical plane
D is the diffusion coefficient
C is the concentration at distance x

By integrating C(x) between r and $r + \delta$, we obtain:

$$J = \frac{4\pi Dr(r+\delta)}{\delta} (C_{\rm b} - C_{\rm i})$$

The consumption rate of monomers at the surface is: (ChE-204, exercize 4.6)

$$J = 4\pi r^2 k_{\rm g}(C_{\rm i} - C_{\rm r})$$

K is the rate constant of a simple first order deposition reaction

We can write the growth rate as

$$\frac{dr}{dt} = \frac{JV_m}{4\pi r^2}$$

$$\frac{dV}{dt} = \frac{d}{dt} \left(\frac{4}{3} \pi r^3 \right) = \frac{4}{3} \pi \frac{d}{dt} (r^3) = \frac{4}{3} \pi (3r^2) \frac{dr}{dt} = 4\pi r^2 \frac{dr}{dt}$$

$$\frac{dV}{dt} = JV_m \qquad V_m \text{ is molar volume}$$

$$\frac{\mathrm{d}r}{\mathrm{d}t} = \frac{\frac{D}{r} \left(1 + \frac{r}{\delta}\right) V_{\mathrm{m}} (c_{\mathrm{b}} - c_{r})}{1 + \frac{D}{k_{\mathrm{d}} r \left(1 + \frac{r}{\delta}\right)}}$$

From Gibbs-Thompson equation:

$$c_b = c_\infty \, \exp\!\left(rac{2\sigma
m V_m}{r^* R T}
ight) pprox c_\infty \left(1 + rac{2\sigma
m V_m}{r^* R T}
ight)$$
 $\stackrel{V_m}{}$ is the molar volume R is the universal gas constant

 C_{∞} is the solubility of the bulk material σ is the interfacial energy V_m is the molar volume R is the universal gas constant

Since the diffusion layer δ is usually of the orders of microns, we can assume $r << \delta$ and derive:

$$\frac{\mathrm{d}r}{\mathrm{d}t} = \frac{2\sigma V_{\mathrm{m}}^2 c_{\infty}}{RT(1/D + 1/k_{\mathrm{d}}r)} \frac{(1/r^* - 1/r)}{r}$$

Lifshitz-Slyozov-Wagner (LSW) Theory (1950)

If the growth is DIFFUSION-LIMITED, $D \ll K_D$

$$\frac{dr}{dt} = \frac{2\sigma D V_{\rm m}^2 c_{\infty}}{RT} \frac{(r/r^* - 1)}{r^2} = K_{\rm D}(r/r^* - 1)/r^2$$

ASSUMPTIONS:

- Spherical particles in a supersaturated solution
- Processes such as nucleation and aggregation that introduce new particles are negligible

If the growth is DIFFUSION-LIMITED,

$$\frac{dr}{dt} = \frac{K_D}{r} \left(\frac{1}{r^*} - \frac{1}{r} \right) \longrightarrow$$

$$\frac{\bar{r}}{r^*} \ge 2 \qquad \frac{d(\Delta r)}{dt} \le 0$$

The size distribution will be self-sharpening over time

$$\frac{\bar{r}}{r^*} < 2 \qquad \frac{d(\Delta r)}{dt} > 0$$

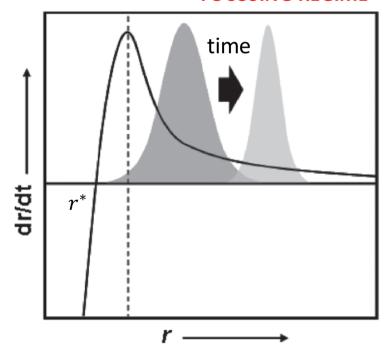
The size distribution will become broader over time

Uncommon because in the earlier growth stages S is high, thus r* is small

size distribution

$$\frac{d(\Delta r)}{dt} = \frac{K_D \Delta r}{\bar{r}^2} \left(\frac{2}{\bar{r}} - \frac{1}{r^*}\right)$$

FOCUSING REGIME



If the growth is REACTION-LIMITED, $D\gg K_D$

$$\frac{\mathrm{d}r}{\mathrm{d}t} = \mathsf{K}_\mathsf{D} \left(\frac{1}{r^*} - \frac{1}{r} \right) \qquad \qquad \frac{\mathrm{d}(\Delta r)}{\mathrm{d}t} = \frac{\mathsf{K}_\mathsf{D} \Delta r}{\overline{r}^2}$$

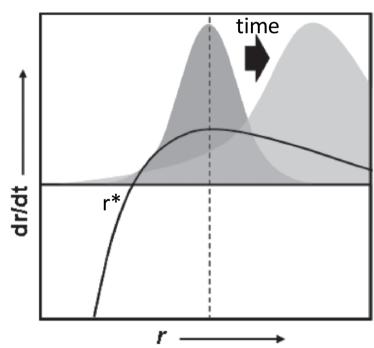
$$r = r^* \qquad \frac{dr}{dt} = 0$$

$$r < r^*$$
 $\frac{dr}{dt} < 0$ The particle will dissolve

$$r > r^*$$
 $\frac{dr}{dt} > 0$ The particle will grow

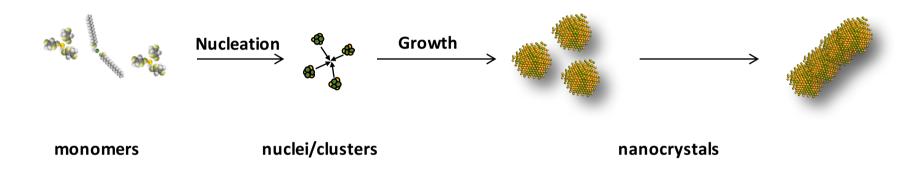
For any
$$\bar{r} = \frac{d(\Delta r)}{dt} > 0$$

DEFOCUSING REGIME

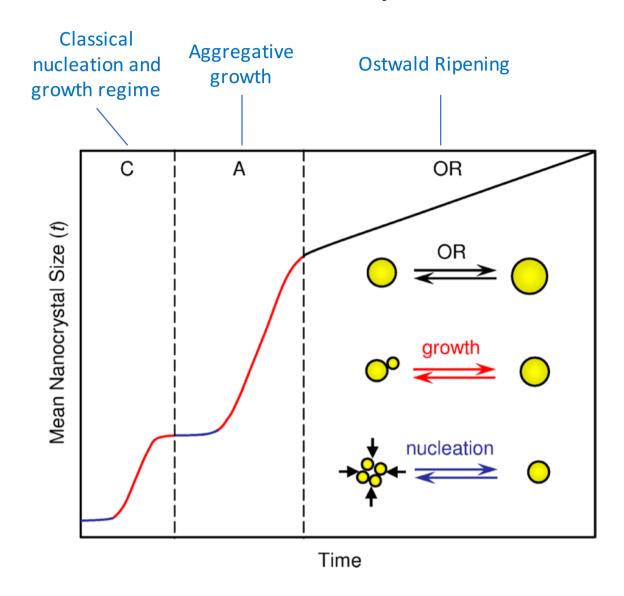


Bigger particles will grow faster than smaller particles (Ostwald Ripening)

3.3. The Non-Classical Nucleation Theory (or Aggregative Nanocrystal Growth)



The growth proceeds by coalescence of nuclei



DRIVING FORCES FOR COALESCENCE:

- Van der Waals forces
- Hydrophobic attractions
- Charge-charge interactions
- Dipolar interactions

DRIVING FORCE FOR ALIGNMENT:

The thermodynamic driving force for the reduction of the defect-free energy becomes significant at the nanoscale, where it induces the rotational motion of the primary particles and the lattice reconstruction.

3.4.1 Metal Oxides (MO)

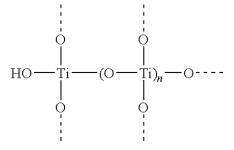
Sol-gel synthesis of MO thin films: Hydrolysis-Condensation Reaction

$$(H_3C)_2HCO \xrightarrow{\text{Ti}} OCH(CH_3)_2 + 4 H_2O$$

$$OCH(CH_3)_2$$

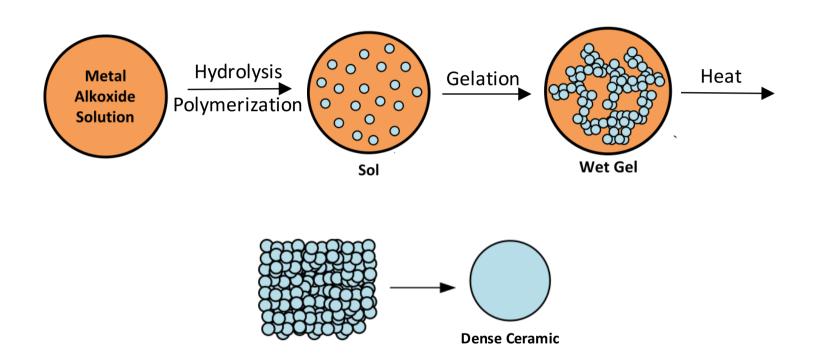
Titanium (IV) Isopropoxide

Titanium Hydroxide



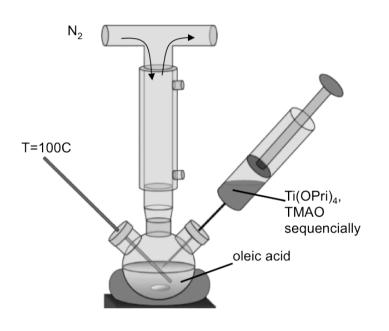
3.4.1 Metal Oxides (MO)

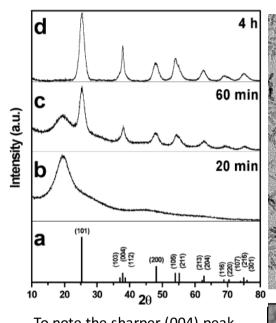
Sol-gel synthesis of MO thin films: Hydrolysis-Condensation Reaction

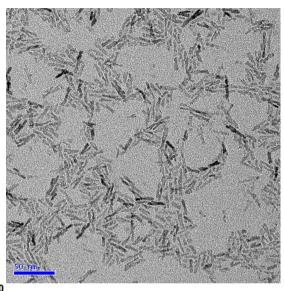


3.4.1 Metal Oxides (MO)

Synthesis of anatase TiO₂ nanorods







To note the sharper (004) peak

REAGENTS

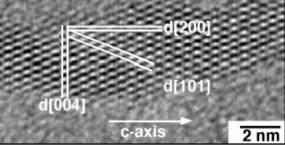
Titanium tetraisopropoxide: Ti precursor

Trimethylamino-N-oxide (TMAO: base to catalyze hydrolysis)

Oleic acid: ligand

Ethanol: non-solvent for clean-up

Hexane: solvent



3.4.1 Metal Oxides (MO)

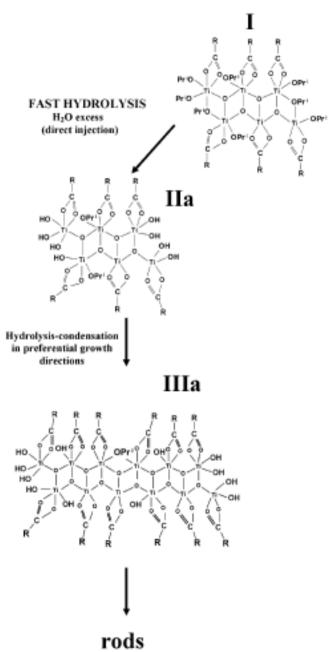
Hydrolysis/condensation with Oleic Acid as ligand

$$Ti(OR)_4 + xH_2O \rightarrow Ti(OH)_x(OR)_{4-x} + xROH$$
 (1)

$$(OR)_{4-x}(OH)_{x-1}Ti-OH + XO-Ti(OH)_x(OR)_{3-x} \rightarrow$$

 $(OR)_{4-x}(OH)_{x-1}Ti-O-Ti(OR)_{3-x}(OH)_x + XOH$ (2)

where $R = -CH(CH_3)_2$; X = H, R.

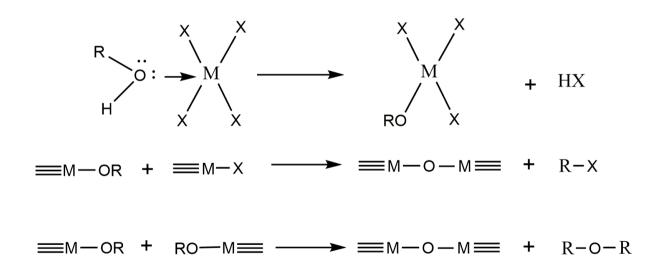


3.4.1 Metal Oxides (MO)

Some typical reactions for the synthesis of metal oxide nanocrystals

> Alcoholysis:

Metal halides and alcohols



3.4.1 Metal Oxides (MO)

Some typical reactions for the synthesis of metal oxide nanocrystals

> Alcoholysis:

• Metal acetylacetonates and alcohols

3.4.1 Metal Oxides (MO)

Some typical reactions for the synthesis of metal oxide nanocrystals

> Aminolysis:

Metal acetylacetonates and amines

3.4.1 Metal Oxides (MO)

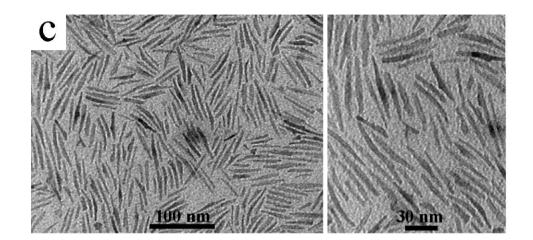
Some typical reactions for the synthesis of metal oxide nanocrystals

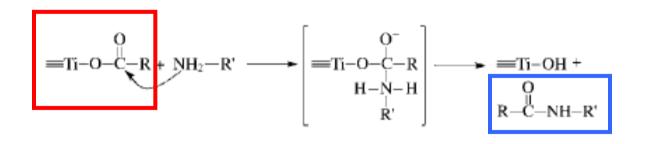
> Ester Elimination:

3.4.1 Metal Oxides (MO)

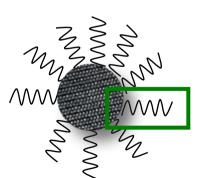
Synthesis of brookite TiO₂ nanorods

REAGENTS





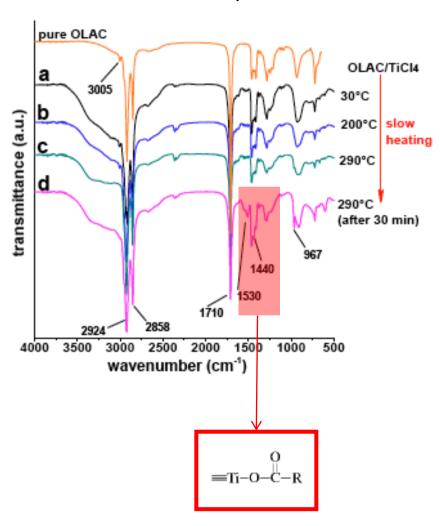
aminolysis of Ti-oleate complex at 280C



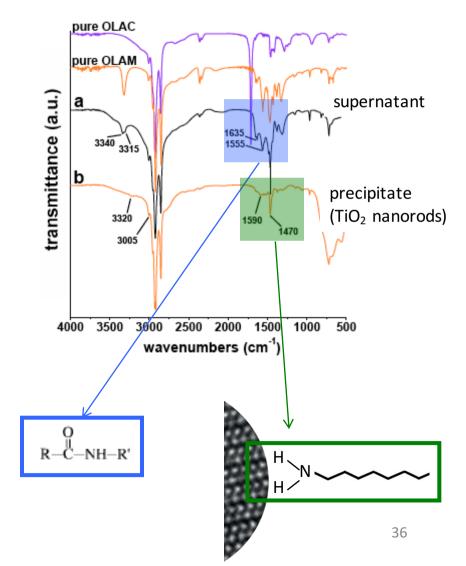
3.4.1 Metal Oxides (MO)

Synthesis of brookite TiO₂ nanorods

FTIR on RCOOH+TiCl₄ mixture



FTIR on RCOOH+RNH₂+TiCl₄ mixture



Buonsanti et al. JACS (2008)

3.4.1 Metal Oxides (MO)

Some typical reactions for the synthesis of metal oxide nanocrystals

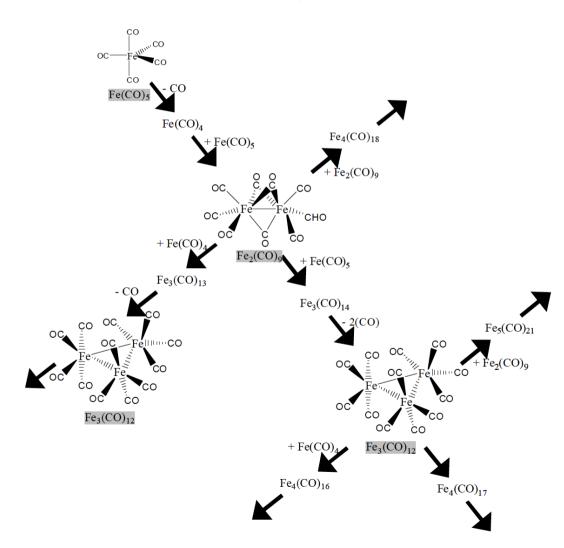
> Thermal Decomposition:

Organo-metallic precursor:

 $Fe(CO)_5 \longrightarrow Fe + 5CO$

Typical solvent: octadecene

Typical ligand: oleic acid



3.4.2 Semiconductors (II-VI, III-V, IV-VI)

It usually occurs by thermal decomposition of highly reactive precursors

Typical synthesis for CdSe quantum dots:

3.4.2 Semiconductors (II-VI, III-V, IV-VI)

Other examples of synthesis for CdSe QDs

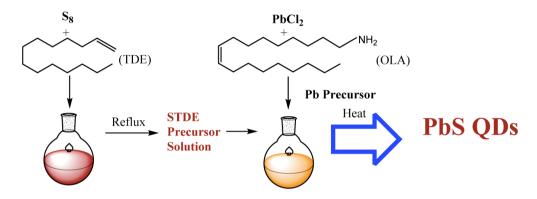
Method 1: ^[23, 117] Dimethylcadmium^a (CdMe₂) and selenium powder were co-dissolved in a tri-alkyl phosphine solvent (-butyl or -octyl) in a glove box (stock solution: 20 g TBP or TOP, 0.197 g Se, 0.534 g CdMe₂). This solution (2 ml) was injected in one shot using a glass syringe into hot (340-360°C) TOPO (6 g) under Argon flow (previously evacuated for 1 hour at 100°C and few mbars). The temperature dropped to 270°C. Nucleation occurred rapidly followed by growth at 280-300°C. The reaction time depended on the size required and can vary from 1 second to several hours. At the end of the reaction, the solution was quickly cooled down by removing the heating mantle and blowing air on the flask, and the NCs were precipitated from solution by adding methanol, isolated by centrifugation and redispersed in toluene, chloroform or hexane.

TBP = tributylphosphine

TOP = trioctylphosphine

TOPO = trioctlyphosphine oxide

3.4.2 Semiconductors (II-VI, III-V, IV-VI) Synthesis of PbS QDs



Noninjection Synthesis

Table 1. Synthesis Conditions and Results of the Noninjection Syntheses of PbS QDs^a

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0 1	40	60	80	100	120	140	160	180	200	220	240
Growth Temperature (°C)											
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Sample No.	Heating Rate (°C/min)	Growth Temperature (°C)	Sample Size (Diameter) (nm)	σ	Band Gap (eV)
QD-1	3.0	45	2.1	8.6%	1.83
QD-2	14.2	90	4.0	6.9%	1.06
QD-3	13.5	130	7.0	6.9%	0.72
QD-4	16.5	160	8.2	7.1%	0.66
QD-5	16.3	185	8.9	6.8%	0.63
QD-6	18.5	200	14.8	7.3%	0.51
QD-7	19.3	220	16.5	7.0%	0.50

^aPb-to-S feed mole ratio is 1:2, and OLA-to-Pb mole ratio is 26:1.

3.4.3 Metals

Representative chemical reduction schematics for metallic nanoparticles

Pathway 1: reduction followed by bonding of atoms

$$M^{x+} + xe \rightarrow M^0$$

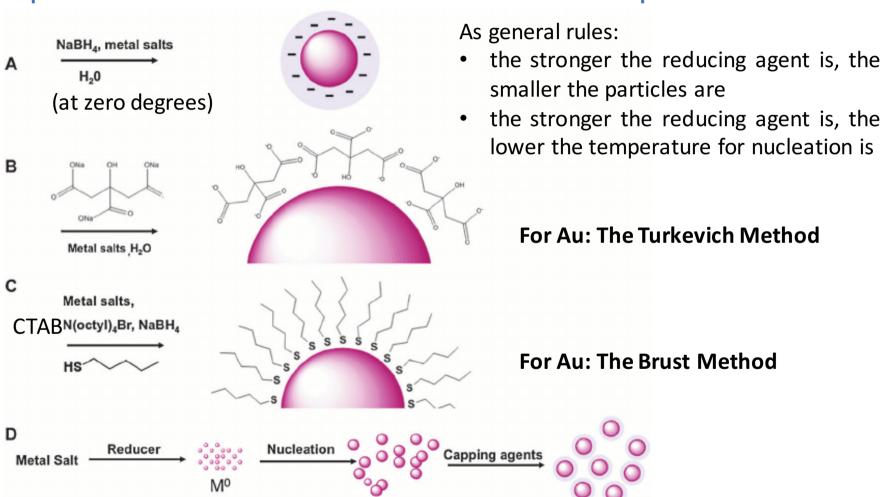
$$M^0 + M_n^0 \to M_{n+1}^0$$

Pathway 2: bonding of atoms as ions followed by reduction

$$M^{x+} + e \rightarrow M^{(x-1)+}$$
 $M^{(x-1)+} + M^{(x-1)+} \rightarrow M_2^{(2x-2)+}$
 $M_2^{(2x-2)+} + xe \rightarrow M_x$

3.4.3 Metals

Representative chemical reduction schematics for metallic nanoparticles



3.4.3 Metals Synthesis of gold nanoparticles

> the Turkevich Method (1951, reduction of tetrachloroaurate by trisodium citrate)

$$2\text{AuCl}_{4}^{-} + \text{Ctr}^{3-} + 2\text{H}_{2}\text{O} \xrightarrow{100 \text{ °C}} 2\text{Au} + 3\text{CH}_{2}\text{O} + 3\text{CO}_{2} \uparrow + 8\text{Cl}^{-} + 3\text{H}^{+}$$

> Reduction of gold chloride by oleylamine

some reduced N-species

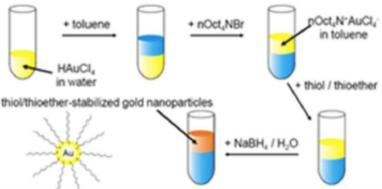
$$AuCl_3 + oleylamine \xrightarrow{80^{\circ}C} Au + NH = CH - R$$

in toluene

3.4.3 Metals Synthesis of gold nanoparticles

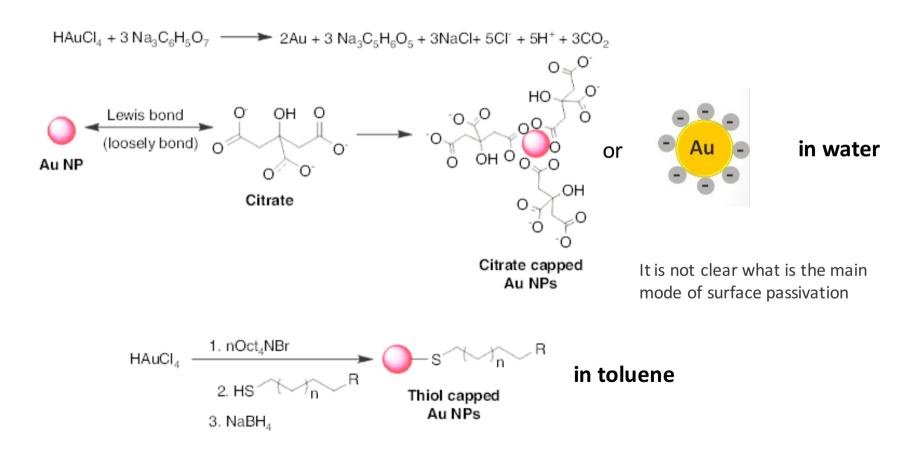
Brust method

- This method was discovered by Brust and Schiffrin in early 1990s, and can be used to produce gold nanoparticles in organic liquids that are normally not miscible with water (like toluene). It involves the reaction of a chlorauric acid solution with tetraoctylammonium bromide (TOAB) solution in toluene and sodium borohydride as an anti-coagulant and a reducing agent, respectively.
- Here, the gold nanoparticles will be 2 to 6 nm in diameter. NaBH₄ is the reducing agent, and TOAB is both the phase transfer catalyst and the stabilizing agent.



3.4.3 Metals Synthesis of gold nanoparticles

Thiols have a stronger interaction with the surface of gold, so the colloidal stability is better



3.4.3 Metals Synthesis of copper nanoparticles

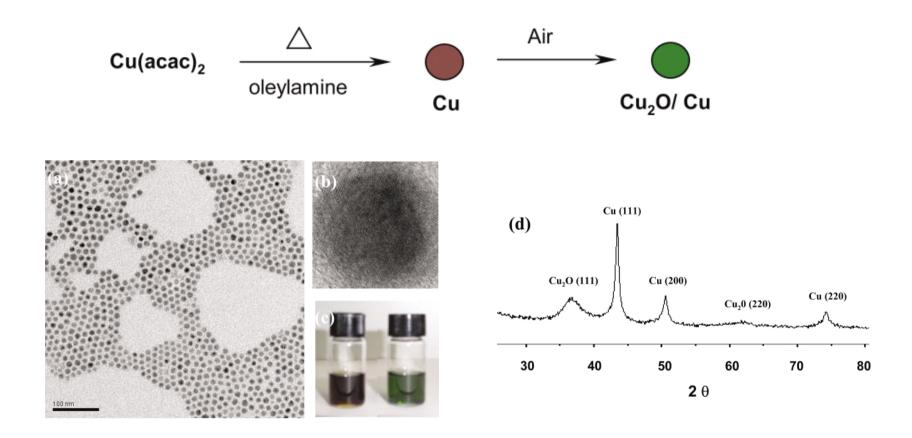
$$Cu^{+} + :NH_{2}-CH_{2}-R$$
 Starting precursors
$$\Delta \downarrow 140 \, ^{\circ}C$$

$$Cu^{+} - :NH_{2}-CH_{2}-R$$
 Complex formation
$$\Delta \downarrow 270 \, ^{\circ}C$$

$$(Cu^{0})_{n} + NH=CH-R \\ N=C-R$$

$$R-H_{2}C-H_{2}N- \rightarrow NH=CH-R \\ R-C=N- \rightarrow N=C-R$$

3.4.3 Metals Synthesis of copper nanoparticles

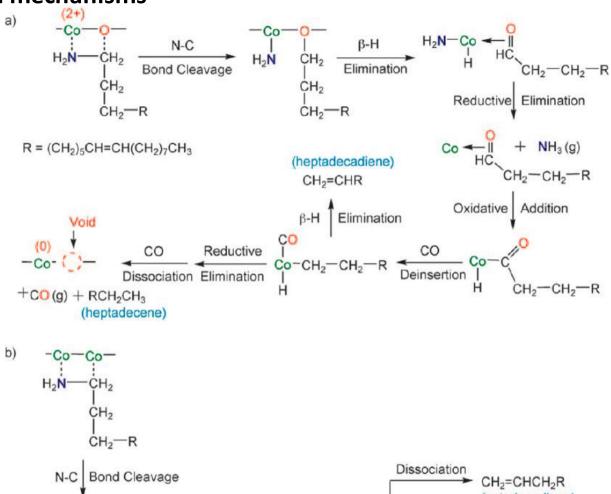


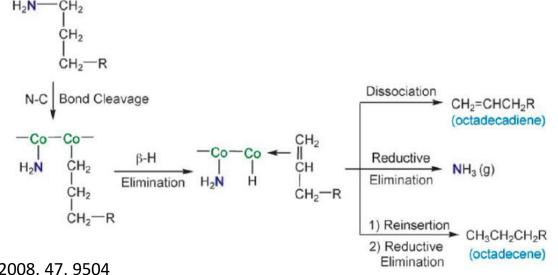
3.4.3 Metals

Some examples

COBALT

CoO reduction by oleylamine





CLASSICAL Nucleation monomers on the surface) NON-CLASSICAL Nucleation Nucleation Growth (by reaction of monomers on the surface) nanocrystals NON-CLASSICAL

coalescence)



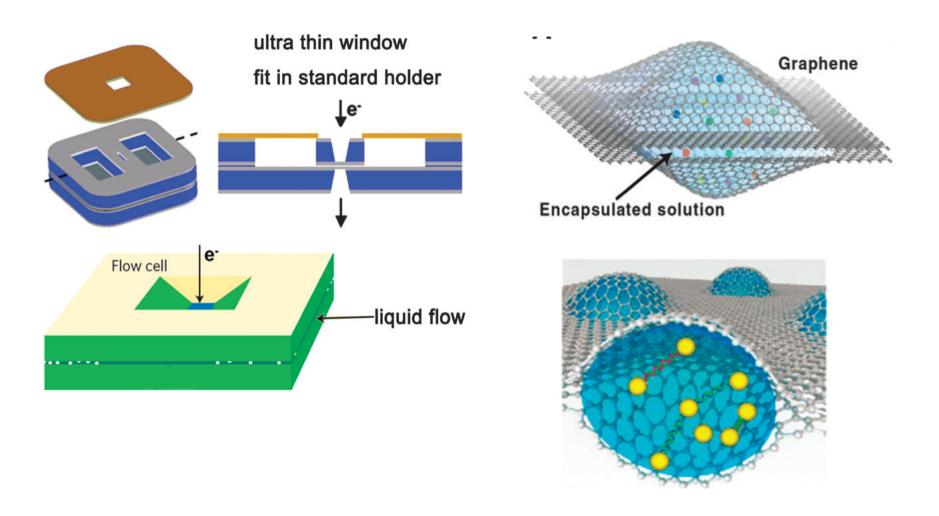
Chemical reactions

- ✓ FT-IR
- ✓ NMR
- √ X-Ray Absorption

Nucleation and growth

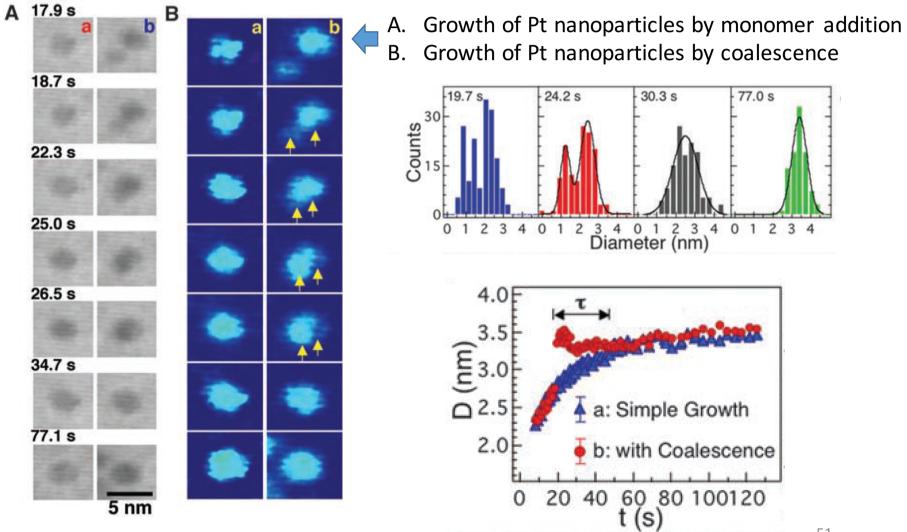
- ✓ UV-Vis
- **✓** TEM
- ✓ XRD
- ✓ X-Ray Absorption

3.5.1. In-situ transmission electron microscopy



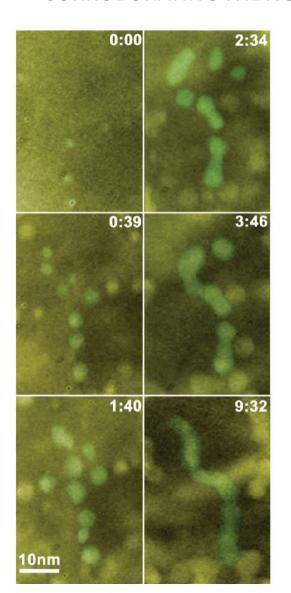
3.5.1. In-situ transmission electron microscopy

CORROBORATING THE NON-CLASSICAL GROWTH

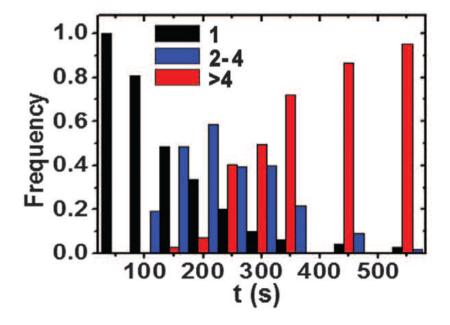


3.5.1. In-situ transmission electron microscopy

CORROBORATING THE NON-CLASSICAL GROWTH

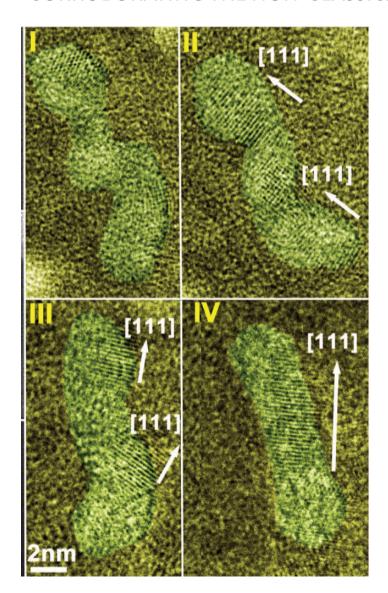


Pt₃Fe nanorods grow by coalescence of primary nanoparticles



3.5.1. In-situ transmission electron microscopy

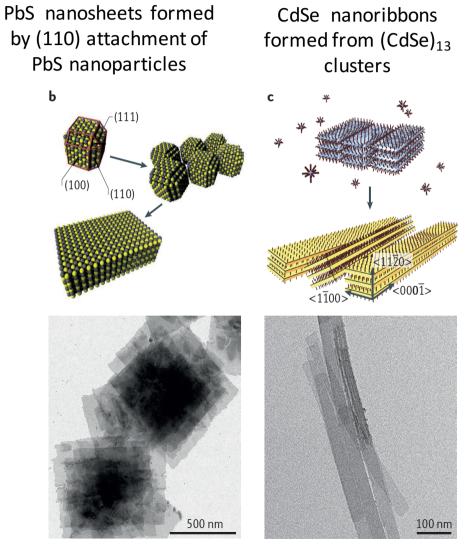
CORROBORATING THE NON-CLASSICAL GROWTH



Pt₃Fe nanorods turn into single crystals and orient along the [111] direction upon crystal orientation (ORIENTED ATTACHMENT)

3.5.1. In-situ transmission electron microscopy

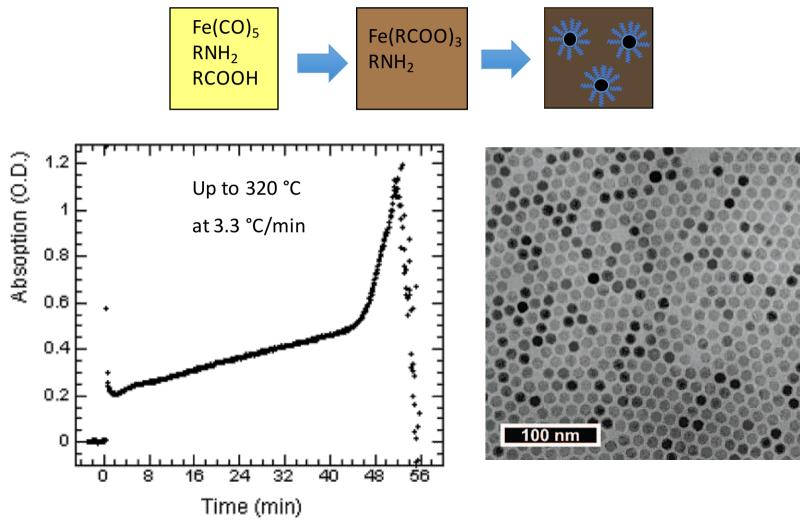
CORROBORATING THE NON-CLASSICAL GROWTH



Hyeon et al. Nature Reviews Materials (2016)

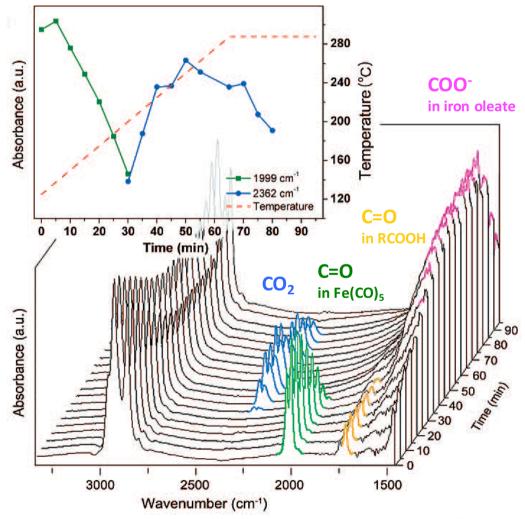
3.5.2. In-situ infrared spectroscopy

Formation and decomposition of iron oleate

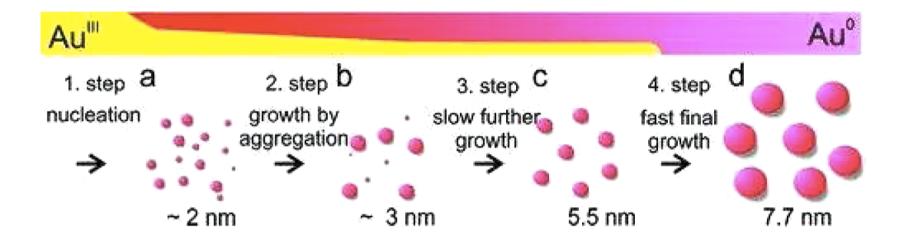


3.5.2. In-situ infrared spectroscopy

Formation and decomposition of iron oleate



3.5.3. In-situ X-Ray Absorption Studies



Teacher: Dr. Michal Strach this afternoon ©

Sorry, not for this course;)

Main references for this module:

Buhro et al. *Chem. Mater.* **2014**, 26, 5

Hyeon et al. Nature Review 2016, 1, 1

Niederberger et al. Chem. Eur. J. 2017, 23, 8542

Mahiddine et al. Chem Rev. 2014, 114, 7610